# Assessing the autogenous shrinkage cracking propensity of concrete by means of the restrained ring test

Die Bewertung der autogenen Schwindrissneigung von Beton mit Hilfe des Ring-Tests

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#### Vorwort

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# **Summary**

The autogenous shrinkage due to self-desiccation of high- and ultra-high performance concretes with very low water-cement ratio in case of restraint leads to considerable stresses starting from very early age. The resultant risk of cracking presently cannot be adequately investigated. Parameters that are particularly difficult to capture experimentally are the concrete temperature and the viscoelasticity.

The primary objective of this work was to assess as precise as possible the autogenous shrinkage cracking propensity of representative concretes at strong restraint and constant room temperature. Test methods needed to be chosen and enhanced in a way that preferably allowed for the efficient and precise investigation of all relevant factors in the future. Ideally, a method suitable for a complete empirical modeling was provided.

First the methodological requirements and the advantages and disadvantages of existing test methods were discussed. Based on this, optimized test methods were proposed. Their suitability was verified using the example of ultra-high strength concrete. The choice of concrete compositions considered the essential measures for reducing shrinkage (internal curing, shrinkage-reducing admixtures, reduction of the fraction of Portland cement in the binder).

The autogenous shrinkage was measured with the shrinkage cone method. This new test method was validated by investigations of the repeatability and reproducibility and proved efficient and precise. It allows for measurements under non-isothermal conditions; no established test method exists for that purpose to date. The autogenous shrinkage of the ultra-high strength concretes at the age of 24 h, investigated under quasi-isothermal conditions (20 °C), was between 0,25 mm/m and 0,70 mm/m. It was particularly low when a shrinkage-reducing admixture was added and when superabsorbent polymers were used.

The stresses due to restraint were determined with the restrained ring test. A large part of the stresses to be expected according to Hooke's Law was eliminated by creep and relaxation. The relaxation capacity being very pronounced at very early age was the main reason that no visible cracking occurred, not even with the concretes with high autogenous shrinkage.

The development of the autogenous shrinkage cracking propensity was described as ratio of restraint stress and splitting tensile strength. By means of modified ring tests, used to determine the maximum tensile stress, it could be shown that the ratio of stress to strength is an appropriate failure criterion. However, the cracking propensity can be calculated correctly only if the strongly age-dependent ratio of uniaxial to splitting tensile strength is accounted for. Besides, it needs to be considered that at very early age a plastic stress redistribution may occur in restrained ring tests.

The reference concrete showed a high cracking propensity of up to 0.68. The fact that shrink-age-reducing measures led to significantly lower values reveals their relevance for the safe application of ultra-high strength concrete. However, the investigations carried out here at 20 °C do not allow for a final assessment of the cracking propensity under typical on-site conditions. To empirically model the autogenous shrinkage cracking propensity as a function of temperature and stress level in the future, an analytical stress solution for non-isothermal restrained ring tests and a new approach for investigating the residual stress and relaxation capacity by means of non-passive restrained ring tests was suggested.

# **Kurzfassung**

Das durch Selbstaustrocknung verursachte autogene Schwinden von besonders leistungsfähigen Betonen mit sehr niedrigem Wasserzementwert führt bei Dehnungsbehinderung bereits in sehr frühem Alter zu erheblichen Zwangsspannungen. Die Gefahr der Rissbildung, die sich daraus ergibt, lässt sich bislang nur unzureichend untersuchen. Experimentell besonders schwer zu erfassende Faktoren sind die Betontemperatur und die Viskoelastizität.

Das vorrangige Ziel der Arbeit war die möglichst genaue Ermittlung der autogenen Schwindrissneigung repräsentativer Betone bei starker Dehnungsbehinderung und konstanter Raumtemperatur. Dabei waren die Prüfverfahren möglichst so zu wählen und weiterzuentwickeln, dass sich zukünftig alle relevanten Faktoren effizient und genau untersuchen lassen. Im Idealfall sollte eine Methode entstehen, die eine vollständige empirische Modellierung erlaubt.

Zunächst wurden die methodischen Anforderungen und die Vor- und Nachteile existierender Prüfverfahren diskutiert. Darauf aufbauend wurden optimierte Verfahren vorgeschlagen. Ihre Eignung wurde an ultrahochfestem Beton überprüft. Bei der Auswahl der Betone wurden die wesentlichen Maßnahmen zur Schwindreduzierung berücksichtigt (innere Nachbehandlung, schwindreduzierende Zusatzmittel, Verringerung des Portlandzementanteils am Bindemittel).

Das autogene Schwinden wurde mit dem Schwindkegelverfahren gemessen. Das neue Verfahren wurde durch Untersuchungen zur Wiederhol- und Vergleichsgenauigkeit validiert und erwies sich als effizient und genau. Es ermöglicht Messungen unter nicht-isothermen Bedingungen; hierfür existiert bisher kein etabliertes Verfahren. Das autogene Schwinden der untersuchten ultrahochfesten Betone unter quasi-isothermen Bedingungen (20 °C) betrug im Alter von 24 h zwischen 0,25 mm/m und 0,70 mm/m. Besonders gering war es bei Zugabe eines schwindreduzierenden Zusatzmittels bzw. Verwendung superabsorbierender Polymere.

Mit dem Ring-Test wurden die bei Dehnungsbehinderung entstehenden Spannungen ermittelt. Ein großer Teil der gemäß Hooke'schem Gesetz zu erwartenden Spannungen wurde durch Kriechen und Relaxation abgebaut. Die im sehr frühen Alter stark ausgeprägte Relaxationsfähigkeit war der wesentliche Grund dafür, dass es selbst bei Betonen mit hohem autogenen Schwinden zu keiner erkennbaren Rissbildung kam.

Die Entwicklung der autogenen Schwindrissneigung wurde als Verhältnis von Zwangsspannung und Spaltzugfestigkeit beschrieben. Durch modifizierte Ring-Tests, mit deren Hilfe die maximale Zugspannung ermittelt wurde, konnte gezeigt werden, dass das Verhältnis von Spannung und Festigkeit als Versagenskriterium geeignet ist. Die Rissneigung lässt sich aber nur dann korrekt berechnen, wenn das stark altersabhängige Verhältnis von einaxialer Zugfestigkeit und Spaltzugfestigkeit berücksichtigt wird. Außerdem ist zu beachten, dass es im sehr frühen Alter zu einer plastischen Spannungsumlagerung in Ring-Tests kommen kann.

Der Referenzbeton wies eine hohe Rissneigung von bis zu 0,68 auf. Dass die schwindreduzierenden Maßnahmen zu deutlich geringeren Werten führten, zeigt deren Bedeutung für den sicheren Einsatz von ultrahochfestem Beton. Die hier bei 20 °C durchgeführten Untersuchungen erlauben allerdings keine abschließende Bewertung der Rissneigung unter baustellentypischen Bedingungen. Um die autogene Schwindrissneigung zukünftig als Funktion der Temperatur und des Lastniveaus empirisch modellieren zu können, wurden eine analytische Spannungslösung für nicht-isotherme Ring-Tests und ein neuer Ansatz zur Untersuchung der Resttrag- und Relaxationsfähigkeit mit Hilfe nicht-passiver Ring-Tests vorgeschlagen.

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#### 1 Introduction

Autogenous shrinkage is the major shrinkage component of concretes that contain much less water than would be required for complete hydration. This mainly applies to ultra-high strength concrete and, to a lesser extent, to high strength concrete. Both have particularly low water-cement ratios. The relative surplus of cement leads to an internal drying, irrespective of whether the concrete dries out to the ambient air or not. This process of so called self-desiccation is associated with autogenous shrinkage which, if restrained, can lead to cracks, potentially impairing the in many respects outstanding durability of these kinds of concrete. Hence, to fully benefit from the advantages of high and ultra-high strength concrete, it is essential to minimize the risk of autogenous shrinkage cracking. Attempts to do so, however, require a reliable method for assessing this risk. Presently, there is no such method.

Cracks are the result of relatively complex processes, in particular at early age as concrete properties change rapidly. A dependable assessment of the cracking risk requires comprehensive testing and a thorough understanding of the interacting parameters. Early age cracking in cementitious systems is not a new problem; cracking due to restrained drying shrinkage and thermal contraction has been examined at length. However, the investigation and prediction of stresses and cracks due to autogenous shrinkage brings about new challenges. The essential issue is the onset of stresses at *very* early age. This greatly increases the influence of creep and relaxation. Especially at stress levels close to failure this influence is highly non-linear and difficult to quantify, experimentally as well as mathematically.

Another challenge is the fact that temperature strongly influences the autogenous shrinkage and, presumably, the cracking risk as well. From isothermal tests at different temperatures it appears that this influence cannot be accounted for by formulas conventionally used to describe the temperature dependency of cement hydration. The lack of clarity in this regard in part is a consequence of a series of methodological issues, most importantly the large number of different test methods and the difficulties in defining the onset of the autogenous shrinkage. The measurement of autogenous shrinkage, yet error-prone at constant temperatures, becomes particularly demanding at realistic temperature histories. The thermal deformations that inevitably superimpose the shrinkage strains are difficult to compensate for. At present there is no general agreement on how to measure the autogenous shrinkage under non-isothermal conditions.

In brief, the current knowledge about the influence of creep and temperature on autogenous shrinkage, restraint stress and cracking is insufficient. Obviously the experimental methods need to be improved in order to overcome the existing deficiencies. The main aim of this study therefore is to contribute to this improvement. The experimental focus is put on tests of the autogenous shrinkage and on restrained ring tests. The common stress-strength failure criterion is used to assess the risk of cracking due to restrained autogenous shrinkage, or as it will be called herein, the 'autogenous shrinkage cracking propensity'. The strength is determined mainly by splitting tension tests. Special restrained ring tests are carried out to further examine the applicability of the chosen failure criterion. The potential of the comparably simple restrained ring test in quantifying creep is investigated. While the principal tests are conducted under quasi-isothermal conditions, the methodological analysis takes into account that tests of shrinkage and stresses under non-isothermal conditions will be required as well.

Non-reinforced fine-grained ultra-high strength concrete is used as material. Steel fibers are disregarded to obtain a more undisturbed view on essential phenomena, and large aggregates are omitted to reduce the test efforts as well as to control thermal effects more easily. The autogenous shrinkage of the chosen reference concrete can be expected to be very high, making it particularly suitable. To provide for a wide range of results, concrete compositions include superabsorbent polymers, shrinkage-reducing admixture and ground-granulated blast furnace slag, representing common measures for reducing the autogenous shrinkage.

The following chapters report the individual aspects and results of this study. In Chapter 2, following an introduction into the autogenous shrinkage and the primary research approaches, the major issues regarding its measurement are described. The present knowledge as to the influencing parameters is briefly depicted. The concretes used in the own investigations are presented in Chapter 3, including mechanical and some other properties. Development and validation of the new shrinkage cone method for measuring the autogenous shrinkage are comprehensively described in Chapter 4. The autogenous shrinkage of the concrete compositions investigated is given. Furthermore, the method's suitability for tests under non-isothermal conditions is outlined. The subsequent Chapter 5 summarizes the basics of stress development and cracking due to restrained autogenous shrinkage and provides explanations of the terms 'autogenous shrinkage cracking propensity' and 'very early age'. In Chapter 6 it is first explained why an available temperature-stress testing machine was not used in this study. Then the restrained ring test is analyzed as to its suitability for investigating the autogenous shrinkage cracking propensity. The methodological foundations of this method are given and the own investigations with the restrained ring test are presented. The work is summarized and conclusions are drawn in Chapter 7. The utilized literature can be found in Chapter 8 and additional data are annexed in Chapter 9.

#### 2 Autogenous shrinkage

# 2.1 Shrinkage and hydration

Shrinkage and swelling of cement-based materials are load-independent volume changes caused by drying and wetting [Loc 00] or, in other words, by changes of the supply and state of water ("Wasserhaushalt") [Gru 91], [Gru 03]. These volume changes originate mainly from the cement paste, therefore shrinkage and swelling of concrete depend on the cement paste content. Commonly, shrinkage denominates a volume decrease and swelling a volume increase, both terms referring to the external volume of a sample or a specimen, including its voids. Experimentally, either shrinkage or swelling is observed, however, the respective underlying mechanisms may proceed simultaneously. Wetting and drying of concrete and the related volume changes usually cannot be fully prevented under field conditions. Especially shrinkage is a concern, mainly because of the cracks it can cause. Cracks are disadvantageous in several respects (cf. 5.3).

According to the mechanisms and the period it takes place, shrinkage can be subdivided into four types, viz. chemical, plastic, drying and carbonation shrinkage [Gru 91]. The autogenous shrinkage generally is considered part of the chemical shrinkage which comprises all non-thermal volume changes due to hydraulic reactions in cementitious systems [Dav 40], [Taz 99]. Autogenous shrinkage occurs irrespective of whether moisture is released to the surroundings or not, whereas plastic and drying shrinkage mainly take place due to evaporation from the surface during the fresh and the hardened state, respectively. The loss of water that leads to plastic shrinkage can be largely prevented by external curing measures, for example by water ponding. Drying shrinkage to a large extent can only be postponed by such measures. Generally, shrinkage due to evaporation from surfaces is not uniform. The resultant Eigenstresses can lead to cracks even if there is no external restraint (cf. 5.2). Such cracks usually begin at the surface but may affect deeper regions of the cross section later on.

Contrary to plastic and drying shrinkage, autogenous shrinkage is caused by an internal lack of water and therefore largely uniform. In most cases it cannot be minimized efficiently by an external supply of water, because it mainly occurs in particularly dense concretes with few capillary pores able to transport water. It usually leads to cracks only if restrained, mainly by aggregates, reinforcement, formwork or adjacent structures (cf. 5.2). The fourth type of shrinkage, carbonation shrinkage, is a volume reduction that begins at and usually is limited to the surface area, potentially leading to fine surface cracks. It is caused by the gradual carbonation of calcium hydroxide and therefore is more pronounced with less dense concrete surfaces which allow more of the required carbon dioxide to be absorbed from the ambient air. Consequently, carbonation shrinkage usually is negligible whenever autogenous shrinkage is not, in particular in dense or very dense high and ultra-high strength concretes.

It is noted that the term 'ultra-high strength concrete' in Germany is used for concrete with a compressive strength of minimum 150 MPa. Sometimes the term 'ultra-high *performance* concrete' is preferred in order to express that the compressive strength is not the only outstanding property. The development of ultra-high strength concrete goes back to the works of Bache [Bac 80], [Bac 81], [Bac 87]; substantial contributions were later made by Richard and Cheyrezy [Ric 95]. Historically, the addition of silica fume to concrete was a decisive step. The positive influence of silica fume on density and strength was recognized in the 1970ies [Sel 86]; a Swiss patent dates back to 1971 [Sik 76].

Shrinkage is a complex process. One factor fundamental for all types of shrinkage, except for plastic shrinkage, is the structure of the hardening cement paste. This structure is a result of hydration, a process of numerous chemical reactions, primarily of cement and water. Today, the principles of hydration of Portland cement are largely understood [Loc 00], [VDZ 08]. The main compounds of Portland cement clinker are tricalcium silicate (3CaO·SiO<sub>2</sub>, 'C<sub>3</sub>S' as per cement chemist notation), dicalcium silicate (2CaO·SiO<sub>2</sub>, 'C<sub>2</sub>S'), tricalcium aluminate (3CaO·Al<sub>2</sub>O<sub>3</sub>, 'C<sub>3</sub>A') and tetracalcium aluminoferrite (4CaO·Al<sub>2</sub>O<sub>3</sub>·Fe<sub>2</sub>O<sub>3</sub>, 'C<sub>4</sub>AF'). Calcium silicates (C<sub>3</sub>S, C<sub>2</sub>S) make up the major fraction. The most important group of reactions is the formation of poorly crystalline calcium silicate hydrates and crystalline calcium hydroxide from calcium silicates and water. The hydrates of calcium silicates, usually abbreviated as 'C-S-H phases', are extremely small needles that grow on the cement grains, increasing the surface area by a factor of more than 100 to 1000 [Cze 77]. Due to their size these needles are largely X-ray amorphous and one also speaks of 'C-S-H gel' [Tay 86]. A recent investigation evidenced the crystalline structure of C-S-H phases of C<sub>3</sub>S hydration [Röß 06].

Hardened Portland cement paste mainly consists of C-S-H phases. The development of strength and most other properties of the hardened cement paste is governed by the growth of these phases and the concurrent increase of Van der Waals forces [Loc 00]. In addition, calcium hydroxide is formed, mainly because the ratio of calcium to silicium on average is higher in C<sub>3</sub>S than in the formed C-S-H phases. The high alkalinity that is provided for by the calcium hydroxide is essential for reinforced concrete because it protects reinforcing steel from corrosion. The other compounds, i.e. tricalcium aluminate (C<sub>3</sub>A) and tetracalcium aluminoferrite (C<sub>4</sub>AF), react in the presence of sulfates to calcium aluminate hydrates and calcium ferrite hydrates. The required sulfates stem from calcium sulfate and alkali sulfates. The amount of available sulfates is important for order, type and rate of reactions and typically is controlled by the addition of calcium sulfates to the clinker during the grinding process, mainly in the form of gypsum and anhydrite. The most frequent reaction products of the hydration of C<sub>3</sub>A and C<sub>4</sub>AF are ettringite and monosulfate, i.e. crystalline calcium sulfoaluminate hydrates with different amounts of water and sulfate. Especially ettringite binds large amounts of water. Pronounced autogenous shrinkage was observed with C<sub>3</sub>A-rich cements (cf. 2.5.1).

The given description of hydration is very much simplifying, and hydration comprises many other reactions and processes which partly have not been fully clarified, yet. Also the bonding forces acting within the C-S-H phases are the object of a continuous discussion [Pow 58], [Wit 77], [Set 08]. Refined experimental techniques for investigations at micro- and nanometer scale continuously allow for new insights and improved understanding. For instance, C-S-H gel today is assumed to consist of areas with low and high density [Jen 00]. Nevertheless, the properties and the behavior of hardening Portland cement paste cannot yet be predicted by models in a general and precise way [Jen 08]. Notably, hydration can be even more complex with cement pastes or concretes that contain cements other than Portland cement. Ultra-high strength concrete commonly contains additional latent hydraulic or pozzolanic materials as well as chemical admixtures that may significantly alter the hydration process [Kor 09].

The chronological course of Portland cement hydration can be divided into three periods, viz. pre-induction, induction and acceleration period [VDZ 08]. These periods are characterized by a more or less rapid transformation of cement compounds into hydrate phases. Generally, C<sub>3</sub>S and C<sub>3</sub>A hydrate more rapidly than C<sub>2</sub>S and C<sub>4</sub>AF. During the short initial pre-induction

period mainly the C<sub>3</sub>A is involved in very rapid and exothermic reactions. A large part of the sulfates is dissolved as soon as cement and water come into contact, and short crystals of ettringite form on the C<sub>3</sub>A surfaces. After less than an hour the induction period begins, which is also called dormant period because reactions appear to be virtually halted. It is still not fully clear whether reactions at this early stage are slowed down because the first hydration products coat the cement grains and act as a barrier, or because of the dissolution and precipitation processes involved [Jui 09]. In any case, the induction period is of great practical importance because it allows for a sufficient time span for casting a concrete.

After one to three hours the formation of C-S-H phases restarts at more notable rates, gradually leading to the setting of the cement paste; as will be outlined below, this phase of solidification plays an important role for the onset of autogenous shrinkage (cf. 2.2 and 2.3.2). The concentration of ions (mainly Ca<sup>2+</sup>, OH<sup>-</sup>, K<sup>+</sup>, Na<sup>+</sup>, SO<sub>4</sub><sup>2-</sup>) increases, complete solidification is reached (final set), and approximately four hours after mixing the acceleration period begins, characterized by rapid hydration of calcium silicates and strong hardening of the cement paste. This final period of early hydration lasts until approximately 12 to 24 h. Afterwards, hydration proceeds at lower and constantly decreasing rates, occasionally considered a fourth period of hydration.

The hydration progress in general and the setting and hardening in particular notably depend on temperature. The acceleration that is caused by higher temperatures can be described quite precisely by the so called maturity approach. It is based on the Arrhenius equation which relates temperature to the rate of a chemical reaction. Temperature also influences the autogenous shrinkage, yet in a way that is less well understood (cf. 2.5.2). Hydration as such is an exothermic process. Especially in large concrete elements hydration may significantly increase the concrete temperature. The associated thermal strains complicate the investigation of autogenous shrinkage (cf. 2.3.3).

The hydration process furthermore is influenced by the water-cement ratio. It is known that the complete hydration of Portland cement theoretically requires a water-cement ratio of approximately 0.40 [Loc 00]. Notably, even then only 60 to 65 % of the water contained in the paste are chemically bound in hydrate phases. The other 35 to 40 % are bound physically in a way that prevents the water from participating in further chemical reactions. Powers and Brownyard comprehensively investigated this aspect and introduced the terms 'evaporable' and 'non-evaporable' water [Pow 46a]. The evaporable water can be quantified by drying at 105 °C, whereas the non-evaporable water, i.e. the water that is chemically bound in hydrates, requires temperatures of up to 1000 °C to be removed from the hardened cement paste. Some hydrates, in particular ettringite and monosulfoaluminate, may start to dehydrate below 105 °C [Loc 00].

The pioneering work of Powers and Brownyard revealed, among many other things, that the compressive strength proportionally increases with increasing ratio of non-evaporable water to overall water [Pow 48]. This finding also explains essential properties of ultra-high strength concretes which have water-cement ratios of about 0.20 to 0.28. Due to the relative surplus of cement, the ratio of non-evaporable to overall water converges to its theoretical maximum in the course of hydration. Hence, the condition for maximum strength as per Powers and Brownyard is practically fulfilled.

At such low water-cement ratios a significant part of the cement does not hydrate. From an investigation of the phase development in ultra-high strength concrete it actually was found that 13 % of the particularly reactive main compound C<sub>3</sub>S did not hydrate [Kor 09]. Yet, the fraction of grains that remain completely anhydrous may be smaller, because most of the C<sub>3</sub>S can be assumed to hydrate at least at the very surface (cf. [Mös 08]). Thus, even if the overall water allows for the hydration of less than for instance 50 % of the cement, most cement grains probably become interlocked by the C-S-H phases and do not function as a mere filler. This can be seen as another reason for the enormous compressive strength of ultra-high strength concretes.

Whatever the constituents and the boundary conditions of hydration are, the structure of the hardening and hardened cement paste to a considerable extent consists of pores of a widely varying size [Cze 77]. Grossly assuming these pores to be spherical and neglecting entrained air pores and larger voids due to incomplete compaction, the pore radius ranges from several micrometers to less than a nanometer. Generally, the terms 'micropores' (below 2 nm), 'mesopores' (2 nm to 50 nm) and 'macropores' (above 50 nm) may be used for the subdivision of pores [Loc 00].

Often pores are classified as gel pores and capillary pores. However, the range of pore radii these terms were attributed to greatly varies. In literature, the radius that separates the smaller gel pores from the larger capillary pores ranges from two nanometer (2 nm in [Koe 97], 2.5 nm in [Iga 05]) to some tens of nanometer (30 nm in [Set 94], 50 nm in [Gru 03]), making this distinction somewhat arbitrary.

Nevertheless this classification is important because it is based on the origin and the effects of pores of different size. If there is more water than can be bound by hydration, i.e. beginning from water-cement ratios of approximately 0.40, the capillary porosity disproportionally increases with increasing water-cement ratio. Contrary to that, the gel porosity is largely independent of the water-cement ratio. Gel pores primarily develop within the C-S-H gel and between the various hydrates. The essential reason for gel porosity is the fact that the products of hydration have a higher density than the reactants. Thus, the space that is freed in the course of hydration is gradually replaced by pores initially filled with water [Pow 46b], [Loc 00].

Drying of hardened cement paste and of concrete begins from the larger capillary pores. Generally, capillary forces that act in a moist capillary network transport water to less saturated areas. When a pore is partly emptied due to capillary forces, the residual water forms a meniscus which separates the saturated fluid phase from the unsaturated gaseous phase. Menisci can develop in a range of relative humidity of approximately 40 to 90 % [Loc 00]. The relative humidity at which a water-filled pore starts to dry out depends on its size. At pore radii of 1  $\mu$ m, 0.1  $\mu$ m, 10 nm and 1 nm the respective thresholds of relative humidity are 99.9 %, 99.0 %, 89.9 % and 34.8 % [Cze 77]. These values can be calculated by means of the Kelvin equation which relates the pore radius to relative humidity and temperature (Equation 2–1, as per [Lur 03a]). Accordingly, a circular pore with a radius of 1 nm at constant temperature will not be emptied by capillary forces as long as the relative humidity does not fall below 35 %. No capillary forces act in completely dry pores.

$$RH_K = \exp(-\frac{2 \cdot \gamma \cdot M \cdot \cos \theta}{\rho \cdot r \cdot R \cdot T})$$

 $\gamma$ : surface tension, M: molar weight of fluid,  $\theta$ : contact angle between fluid and solid,  $\rho$ : density of fluid, r: radius of meniscus, R: ideal gas constant, T: absolute temperature

It is understood that the Kelvin equation also accounts for the surface tension of the liquid, being the source of capillary forces. Notably, pores in cement paste are not filled with pure water but with pore solution which has a surface tension different from water. The effect of shrinkage-reducing admixtures to a considerable extent can be attributed to the reduction of the surface tension of the pore solution [Raj 08]. Also dissolution processes influence the relative humidity. A respective term which accounts for the drop of relative humidity due to the dissolution of salts was added to the Kelvin equation by Lura et al. [Lur 03a].

The capillary pressure that acts on a meniscus can be calculated from surface tension and pore radius by means of the Young-Laplace equation (Equation 2–2). The pressure increases with decreasing pore radius. The capillary depression within the pores compresses the solid part of the hardened cement paste structure and thereby leads to shrinkage. Hence, shrinkage also depends on the stiffness and viscoelasticity of the walls of the pores.

$$\sigma_{cap} = \frac{2 \cdot \gamma}{r}$$
 Equation 2–2

Pores smaller than the ones that at given boundary conditions can be emptied by capillary forces may experience a decrease of relative humidity by way of diffusion, which is a very slow process, ending only when external and internal relative humidity have leveled out. The forces that develop due to changing relative humidity in the area of such extremely small pores are difficult to investigate. The principal forces under discussion are those due to changes of the surface energy and the so called disjoining pressure [Mue 09], [Lur 03a]. The surface energy within the C-S-H phases is highest in the dry state and decreases when water is adsorbed. Thus, adsorption of water would lead to swelling, and desorption to shrinkage [Lur 03a], [Cze 77]. This effect may be limited though to ranges of comparably low relative humidity as the surface energy is notably changed only by the first three layers of water molecules [Lur 03a]. The disjoining pressure is assumed to act in areas in which no adsorption is possible, i.e. where the solid surfaces are so close to each other that water molecules cannot go in between [Fer 87], [Lur 03a]. It is assumed to be the result of attractive Van-der-Waals forces and two repulsive forces (double layer repulsion, structural force) [Der 74], [Fer 87]. It may be effective also at high relative humidity.

Ultra-high strength concrete, compared to concrete of normal strength, has extremely few capillary pores. Drying shrinkage is dominated by diffusion and proceeds at very low rates. However, a significant internal desiccation takes place and may reduce the relative humidity to less than 85 %. Although the micro-mechanisms that lead to autogenous shrinkage are supposed to be the same that cause drying shrinkage [Mue 09], [Wit 09], the autogenous shrinkage originates from and is part of the chemical shrinkage [Dav 40], [Taz 95], [Taz 99]. As mentioned above, the chemical shrinkage is caused by the fact that reaction products of hydration have a smaller absolute (non-porous) volume than anhydrous cement and water [Pow

46b], [Loc 00]. The capillary forces which are induced by chemical shrinkage in the initial viscous state lead to a macroscopic contraction that proceeds without major resistance of the cement paste. When the material starts to behave as a solid body, chemical shrinkage leads to the formation of an increasingly stiff pore structure which counteracts the material's tendency to contract [JCI 96]. In the further course of hydration, pore water continues to be incorporated into hydration products. In case there is considerably less water available than required for the complete hydration of all reactive constituents, the relative humidity in the pore system starts to decrease. This internal drying of the solidifying cement paste, commonly called self-desiccation, generates the aforementioned forces and leads to a virtually uniform volume reduction. This part of the chemical shrinkage is called 'autogenous shrinkage' [Taz 99]. Another term frequently used is 'self-desiccation shrinkage' [Jen 01a].

## 2.2 Definitions and research approaches

Most aspects of autogenous shrinkage have been known for more than fifty years and are largely undisputed, yet there still is no exact univocal definition. Furthermore, some terminological discrepancies persist. For instance, autogenous shrinkage in some publications of French origin apparently comprises the complete chemical shrinkage [Bar 99], [Bar 05], [Mou 06b], [Bou 08a]. A comprehensive overview of terminology is given in [Jen 01a].

The autogenous shrinkage of most concretes of normal strength is negligible as no self-desiccation takes place. The interest in autogenous shrinkage significantly rose in the 1990's when high strength concrete with relatively low water-cement ratios came into wider use. Despite appropriate curing, early age cracks were observed which could be explained only by restrained autogenous shrinkage [Bui 80], [Zie 82], [Taz 92], [Sel 95]. Subsequent to several years of research the Japan Concrete Institute (JCI) suggested the following definition [JCI 96], published in English in 1999 [Taz 99]: "Autogenous shrinkage is the macroscopic volume reduction of cementitious materials when cement hydrates after initial setting. Autogenous shrinkage does not include volume change due to loss or ingress of substances, temperature variation, application of an external force and restraint."

The definition explicitly excludes thermal deformations and deformations due to loss of water. The initial setting is defined as the beginning of autogenous shrinkage (cf. Figure 2–1). Hence, the definition is of the 'time-zero'-type: the onset of autogenous shrinkage is not directly linked to self-desiccation or underlying micro-mechanisms but to a somehow related point in time that is determined experimentally. This point generally is called 'time-zero' and the subsequent shrinkage is termed 'autogenous' (cf. 2.3.2). According to the JCI's definition, time-zero is to be determined by needle penetration.

The authors' motivation is explained as follows: "Since autogenous shrinkage is generally used for prediction of cracking, the strain generated in a period when cementitious material is fresh is excluded. Therefore, the time of initial setting of cement is specified as the start point of autogenous shrinkage measurement." This emphasizes the functional character of the definition: it is assumed that strains from the moment of initial setting are relevant for the prediction of cracking.

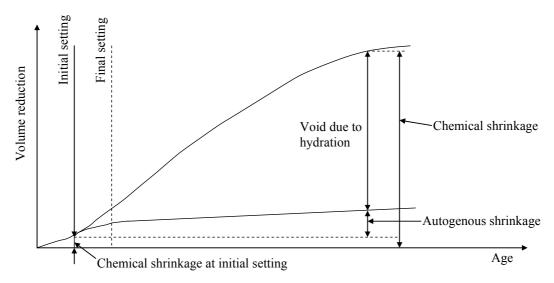


Figure 2–1 Autogenous shrinkage as defined by the Japan Concrete Institute (from [JCI 96])

A second definition, suggested by Jensen and Hansen a few years later, differentiates between "autogenous deformation" and "self-desiccation shrinkage" [Jen 01a]. "Autogenous deformation" is defined as the "bulk deformation of a closed, isothermal, cementitious material not subjected to external forces", whereas "self-desiccation shrinkage" is the "autogenous deformation of a set cementitious material system caused by chemical shrinkage".

This differentiation accounts for the underlying mechanisms, in particular for the fact that the present knowledge in this regard is incomplete. The meaning of the term 'autogenous' is extended to all isothermal bulk deformations that occur without external influence, whatever the mechanisms are. The term 'deformation' comprises negative as well as positive strains. This part of the definition is sufficiently general not to be compromised by future findings on the mechanisms. On the other hand the fraction of autogenous deformation that is known to be caused by self-desiccation is precisely named as such. "Self-desiccation shrinkage" is supposed to occur in a "set system". Rather than to define a starting point, this recalls the solidifying pore structure as a precondition to self-desiccation.

Notably, neither definition fully accounts for the influence of temperature on autogenous shrinkage. Autogenous deformations as defined by Jensen and Hansen are deformations of "isothermal material". A strict interpretation would have the consequence that autogenous deformations only occur at constant temperatures, which, of course, is not the case. The JCI's definition excludes thermal deformations, however, it is not specified at which temperature the autogenous shrinkage is to be measured. The vague consideration of temperature partly can be attributed to the methodological problems associated with measuring its effects (2.3.3, 2.5.2).

The definition suggested by the JCI is more pragmatic and committed to the practical problem of cracking, whereas the definition proposed by Jensen and Hansen is more fundamental because it renounces to simplifying assumptions. The JCI defines the initial setting as the beginning of autogenous shrinkage. Jensen and Hansen do not clearly specify a starting point, even though such a point is required for evaluation. Thus the outlined definitions reflect two different research approaches: a phenomenological and a micro-analytical approach. Both are briefly described in the next paragraphs.

Phenomenological research mainly focuses on the measurement of the autogenous shrinkage strain and the consequences it brings about if restrained, i.e. stresses and cracking (cf. Figure 2–2). In view of the numerous difficulties associated with such measurements, this is a comprehensive and challenging task (2.3, 5). Sometimes relative humidity [Per 97], [Lou 99], [Per 01], [Jia 05] or capillary pressure [Aly 09], [Suz 09], [Rad 98], [Hol 05], [Liu 06], [Mia 07] are examined as well and related to the measured shrinkage strains. Provided that measurement methods are precise and efficient, phenomenological parameter studies can contribute to the clarification of essential relations, for instance the influence of single concrete constituents on autogenous shrinkage. However, phenomenological research usually cannot provide thorough explanations of observed behavior.

Approach	Scale	Investigated Property	Investigated Object	Significance
Pheno-	Macro	Autogenous shrinkage strain	Concrete element / structure	Experimental determination
menological	Meso	Relative humidity (Self-desiccation) Capillary pressure	Concrete / mortar specimen	Validation
	Micro	Structure and micro-mechanisms, e.g.  Hydration products	Cement paste incl. admixtures	Prediction
Micro- analytical	<b>↑</b> Nano	<ul> <li>Porosity and transport processes</li> <li>Disjoining pressure</li> <li>Tension of capillary water</li> <li>Dissolution and precipitation</li> </ul>	Pure cement paste  Cement paste structure, pore system	Physico-chemical explanation and modelling

Figure 2–2 Autogenous shrinkage: schematic comparison of phenomenological and micro-analytical approach

Contrary to that, micro-analytical research is dedicated to the structure and the micro-mechanisms and forces that cause autogenous shrinkage, including their quantitative contribution. Apart from theoretical considerations, experiments typically include various aspects of hydration, the state of water, the pore system, chemical shrinkage and other properties, often observable only at nano- or micro-scale. To reduce the complexity, micro-mechanical models usually are formulated not for concrete but for more simple combinations of constituents, for instance pure cement paste or, more rarely, for cement paste with silica fume and superplasticizer. As a consequence, the autogenous shrinkage strain typically is measured on cement paste for validating a model. Often good agreement of predicted and measured values is achieved, usually limited though to isothermal conditions [Lur 03a].

Jensen and Hansen explicitly criticized that most research on autogenous shrinkage is phenomenological. They called for more physical, chemical and thermodynamic studies aiming at a more complete understanding of underlying mechanisms and the prediction of autogenous shrinkage [Jen 01a]. Lura, Jensen and van Breugel in 2003 reviewed the knowledge about autogenous shrinkage mechanisms [Lur 03a]. They described the close correlation of autoge-

nous shrinkage and internal relative humidity, found in a number of studies [Lou 99], [Per 02], [Naw 04], [Jia 05]. The underlying micro-mechanisms, however, were referred to as unknown. Three potential mechanisms were mentioned: changes in the surface tension of solid gel particles, disjoining pressure and surface tension in capillary water (cf. 2.1). Müller listed the same micro-mechanisms, however, he referred to drying *and* autogenous shrinkage [Mue 09]. Obviously the micro mechanisms of drying and autogenous shrinkage can be considered the same [Wit 09], [Taz 95]. Wittmann recently pointed out that general quantitative solutions based on a micro-analytical approach are not at hand, although drying shrinkage is being researched since more than 50 years [Wit 09].

To predict the micro-mechanisms' quantitative contribution to autogenous shrinkage from one of the current models, several hydration-related parameters are required: in the most simple case the pore volume, the pore size distribution, the state of water in the pores and the stiffness of the cement paste [Lur 03a], [Lur 03c]. Such parameters are difficult to investigate and change rapidly at early age. An efficient solution would be to model these parameters as well. However, this appears to be very difficult. Jennings states [Jen 08]: "Despite the fact that precise characterization of the pore system in cement paste has been the subject of research for more than half a century, we still do not have [...] a full geometric description of the pore system. Therefore, models [...] are still incomplete, and none can adequately predict performance over a wide range of mineral and chemical formulations." Or, as Bažant put it, "A fully rational prediction of concrete creep and shrinkage properties from its composition is a formidable problem, not yet solved satisfactorily" [Baz 01].

An additional obstacle to the practical application of micro-analytical models is the fact that results of cement paste investigations for several reasons cannot be directly transferred to concrete [Bjø 04], [Lur 03c]. For instance, a cement paste may bleed, while a concrete made from that cement paste may not; the reabsorption of bleeding water alters the autogenous shrinkage. Alternatively the water-cement ratio of the cement paste may be modified to avoid bleeding. However, then a direct comparison would be hampered by the different pore systems. Furthermore, concrete usually contains aggregates which restrain deformations, absorb and release water and modify the mechanical properties [Bjø 04]. These influences can cause a different development of relevant micro-mechanical parameters in hardening cement paste and concrete, respectively, limiting the compatibility of results.

Then again, if concrete is investigated instead of cement paste, apart from the major complexity there is another problem: in some cases the influence of certain parameters simply cannot be examined independently. For instance, ultra-high strength concrete typically contains silica fume. In the fresh mix it fills the space in between the cement particles and thereby allows for a reduction of the water content. If the silica fume is left out, the gaps need to be filled largely by water. This can result in the complete loss of workability. Hence, it may be impossible to investigate the influence of cement and silica fume independently. Likewise, superplasticizers in most cases are indispensable for producing ultra-high strength concrete. As they typically modify hydration, results obtained from tests on pure cement paste may not be comparable.

The micro-analytical approach can be considered the key to a more profound understanding of autogenous shrinkage and shrinkage in general. It may even turn restraint stress and cracking into phenomena that can be calculated from the properties of the concrete constituents and the prevailing boundary conditions. However, this will require large resources and take a consid-

erable amount of time. Presently there is no efficient way to precisely predict the autogenous shrinkage for a wide range of concrete compositions and boundary conditions, let alone the stresses and the cracks it may cause. One goal of this study was to improve the assessment of the autogenous shrinkage cracking propensity in order to allow for the efficient investigation and short-term elimination of a major obstacle to the broader application of ultra-high strength concrete. Therefore the phenomenological approach was chosen, and, as a consequence of the numerous challenges, micro-analytical aspects had to be disregarded.

## 2.3 Metrological issues

#### 2.3.1 Multitude of test methods

Since the autogenous shrinkage of high and ultra-high strength concrete usually begins during or shortly after setting, the measurement period necessarily encompasses the transition from viscous suspension to porous solid. Hence, paste, mortar or concrete need to be placed in some kind of container, at least at the beginning of the measurement. A multitude of different setups and test methods has been developed for this purpose, e.g. [Jen 95a] (2.4), [Taz 99] (Figure 2–3), [Gag 99], [Lou 00], [Ham 03], [Naw 04], [Bou 06], [San 06], [Cra 06], [Ong 06]. At the 2006 Conference on Volume Changes of Hardening Concrete held in Denmark, 20 contributions provided results of autogenous shrinkage [RIL 06]. These results were obtained with 15 different test setups; either the measuring technique was different or there were significant differences in the specimen geometry [Epp 08b]. Most setups can be supposed to be in use only in a small number of institutions. The precision of test methods in many cases is unclear.

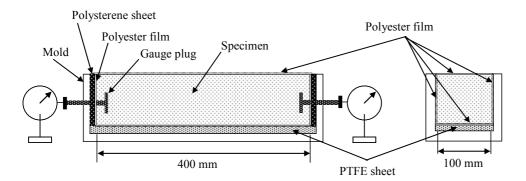
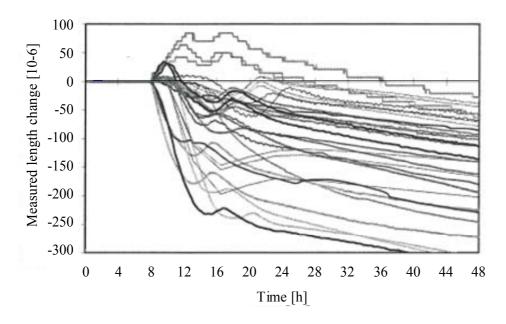


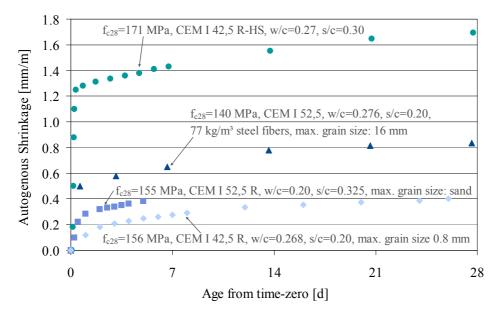
Figure 2–3 Setup for the measurement of autogenous shrinkage of concrete (from [Taz 99])

The large number of test methods represents a serious problem since the measurement of the autogenous shrinkage is difficult, and results are partly inconsistent. A recent round robin test with different test methods and a relatively simple mortar revealed a large scatter (Figure 2–4) [Bjø 06]. According to the statistical evaluation of these tests, at least eight individual measurements of the autogenous shrinkage would have been required to achieve a reliable result [Kra 06]. The results of another round robin test with rather unfavorable outcome are presented in chapter 6.2 dealing with the suitability of temperature-stress testing machines [IPACS].



**Figure 2–4** Result of a round robin test (RILEM Technical Committee 195-DTD), 30 tests of the free autogenous shrinkage of a mortar, constant set temperature (20 °C), 10 institutions involved [Bjø 06]

Widely differing values also were reported for the autogenous shrinkage of ultra-high strength concretes. As the concretes had similar compositions, the discrepancies indicate a significant influence of the test method employed in the respective institution (Figure 2–5) [Epp 08a].



**Figure 2–5** Autogenous shrinkage under quasi-isothermal conditions of similarly composed ultra-high strength concretes, measured with different setups; data (in the order of magnitude of autogenous shrinkage) from [Sch 02] (highest values), [Bjø 03], [Sta 04] and [Ma 03] (lowest values)

Considering the multitude of test methods on the one hand and the observed inconsistencies on the other, the comparability of a large part of the results obtained in the past is in doubt. It was argued that research should focus on reliability of measurement methods before further conclusions are drawn [Bjø 04].

#### 2.3.2 Time-zero

Chemical shrinkage virtually begins with the contact of cement and water and proceeds as long as hydration continues. Autogenous shrinkage is part of the chemical shrinkage and is caused by the decrease of relative humidity in a pore structure that has become sufficiently stiff to partly withstand the internal forces (2.2). Solidification, the decrease of relative humidity as well as the concurrent change of capillary pressure start very slowly, and it is practically impossible to determine the precise onset of these processes. Therefore, many authors attempted to find other criteria that are suitable to define and determine a threshold as a starting point of the autogenous shrinkage. This point in time commonly is called time-zero (cf. 2.2). Earlier and later time-zero result in higher and lower autogenous shrinkage, respectively.

The application of a time-zero criterion in most cases is indispensable for the evaluation of autogenous shrinkage measurements. From a practical point of view the suitability of a criterion mainly depends on its universal ability to delimit the crack-relevant shrinkage as accurately as possible (2.2). However, it would require extremely laborious and sophisticated experiments to examine to which degree a criterion actually fulfills this requirement. No such investigation is known. Alternatively it can be investigated when deformations start to generate restraint stress and how closely the time-zero determined by a given criterion coincides with this onset of stress. However, the development of stresses is a gradual process, too, and its onset depends so much on the sensitivity of the test method, that reliable and precise results would be very difficult to obtain. In conclusion, it is hardly feasible to thoroughly examine the suitability of a time-zero criterion as to the abovementioned requirement.

Partly as a consequence of these difficulties a second requirement gains importance: the application of a criterion should be as simple as possible. The most simple approach is to rely on tests of the solidification process. Two of the most common criteria for determining a threshold within the transition from viscous fluid to porous solid are the initial and final setting time, determined by Vicat needle penetration tests. As mentioned above, the definition of autogenous shrinkage given by the JCI stipulates the initial setting as time-zero (2.2), other authors recommended or used the final setting time, e.g. [Ham 06], [San 06]. The evaluation of shrinkage data from initial setting largely avoids skipping any of the crack-relevant shrinkage. However, the autogenous shrinkage may be considerably overestimated. To apply the final setting time may have the opposite consequence, i.e. an underestimation of autogenous shrinkage.

The significance of time-zero and the impact that the use of different criteria can have on results is illustrated in **Figure 2–6**. The dashed and the solid curve of autogenous shrinkage are calculated from initial and final setting, respectively. The figure is based on own tests with an ultra-high strength concrete. At the age of 18 h the autogenous shrinkage differs by approximately 350  $\mu$ m/m, corresponding to more than 50 % of the shrinkage strain from final setting. Hence, it depends essentially on the choice of the time-zero criterion whether results of different investigations can be reasonably compared or not.

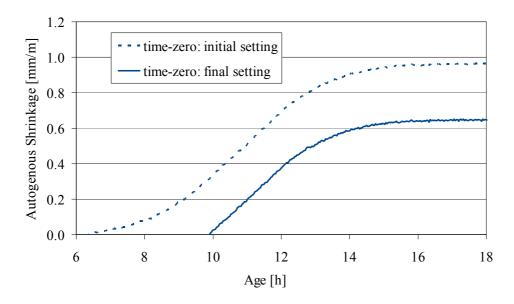


Figure 2–6 Autogenous shrinkage of an ultra-high strength concrete: influence of time-zero

Notably, different standards exist for the determination of setting times by needle penetration tests. For instance, the needle diameter according to ASTM C191-08 is approximately 10 % smaller than the one given in EN 196-3 (Table 2–1). Also, the penetration depth for initial setting differs considerably. Furthermore, it is important to consider the precision of needle penetration tests. According to ASTM C191 the repeatability for initial and final setting is 34 min and 56 min, respectively. This data is in line with experience gathered at German Cement Works Association. The respective values for the reproducibility are 45 min and 122 min. In other words: results for final setting when determined in different laboratories may differ by 2 h and still would have to be considered precise. Moreover, the values apply when final setting occurs within the first 312 min. If it occurs much later, as it is the case with some ultra-high strength concretes, precision tends to be worse.

**Table 2–1** Vicat needle penetration tests: differences between ASTM C191-08 and EN 196-3

	ASTM C 191-08	EN 196-3
Needle diameter	1 mm	1.13 mm
Initial setting <sup>1)</sup>	penetration ≤ 25 mm (sample height: 40 mm)	penetration ≤ 37 mm (sample height: 40 mm)
Climate	23 °C / 95 % RH	20 °C / under water

<sup>1)</sup> Final setting in both standards: needle does not leave a complete impression in the surface anymore

Another disadvantage of needle penetration tests is the disturbing influence of changing consistencies. For the regular control of cement quality by Vicat tests the consistency of the cement paste is adjusted to a standardized value. To achieve this so called normal consistency, the cement's water demand is determined prior to the preparation of the mix for the penetration test. Yet, if the Vicat test is used to determine time-zero, different viscosities of the investigated pastes or mortars are unavoidable. Adding water to adjust the consistency of course would change the shrinkage behavior and yield inadequate results.

Finally, the determination of the final setting time according to the mentioned standards requires manual operation. This may represent an obstacle in case the final setting occurs only after many hours, because it may require tests during night times. Alternatively, an apparatus for the automatic determination of initial setting can be employed and the quasi-continuous needle penetration data it provides may be used to extrapolate final setting times [Epp 08b]. Yet, the precision of such a procedure is unknown.

Several other time-zero criteria have been suggested and applied, for example a point of maximum or minimum shrinkage rate [San 06b], [Fon 06], the increase of the rate of heat release after the induction period as well as changes in the electrical conductivity, in the transmission of ultrasonic waves and in the acoustic emission activity [Lee 06], [Wei 03], [Lur 09]. An overview of various criteria was given recently [San 09]. No criterion was proven to be more suitable and in particular more precise than penetration tests, yet. Especially those criteria that require sophisticated equipment and uncommon know-how may not become widely accepted unless the results for time-zero are more precise and generated largely automatically.

The determination of time-zero becomes a minor problem if parallel tests under restraint conditions are carried out. Then the onset of stress development, for instance in restrained ring tests, can be used as time-zero for the evaluation of free shrinkage tests. Any earlier shrinkage would be irrelevant for the further evaluation of stresses, anyway.

#### 2.3.3 Other metrological issues

The multitude of test methods and the uncertain comparability of results led to an intense discussion about the correct measurement and the possible sources of error. In particular, the fact that measurement needs to begin when concrete is still almost fluid can lead to errors. Artifacts tend to be the larger, the lower the stiffness of the concrete is [Tia 09]. The container itself may restrain the deformation of the concrete. Friction between the container's walls and the concrete may influence results. An underestimation of autogenous shrinkage may be caused by mechanically coupled sensors whose functioning depends on the stiffness of the specific concrete the coupling part is embedded in. Alternatively the autogenous shrinkage can be measured without contact between measuring unit and specimen, for instance with a laser [Epp 08b], [Alo 02], a digital image analysis [Ong 06] or eddy current sensors [Bou 08a].

A major problem lies in thermal strains, especially with large specimens. Concrete in practice is subjected to various temperature histories. As temperature is known to influence autogenous shrinkage, concrete temperature is an essential parameter that needs to be investigated [Jen 99], [Bjø 04]. On the other hand, temperature changes can be a significant source of error since thermal deformations which superimpose the autogenous shrinkage are difficult to account for [Ham 03], [Bjø 04]. A RILEM committee suggested that a maximum temperature difference of 2 K is acceptable for quasi-isothermal tests [Ham 06]. Cross sections of more than approximately 30 x 30 mm² to 50 x 50 mm² usually require an efficient external cooling system to provide for such a constant and uniform temperature. To account for thermal strains in non-isothermal tests, precise knowledge of the coefficient of thermal expansion as a function of time is required. The coefficient can vary considerably at early and very early age and its accurate determination is very difficult [Cus 06], [Sel 06], [Los 10]. Attempts were made to model its development, in particular for high-strength concrete [Oza 06]. Presently, there is

no widely accepted method for measuring the autogenous shrinkage under non-isothermal conditions.

Other metrological aspects frequently discussed are the effects of bleeding, linear vs. volumetric measurement and vertical vs. horizontal measurement. Bleeding is relevant mainly with cement pastes, as high- and ultra-high strength concretes usually do not bleed [Aït 02]. Reabsorbed bleeding water can lower the measured autogenous shrinkage [Bjø 04]. The constant rotation of samples may prevent bleeding effects [Bou 08a].

For geometric reasons volumetric methods should yield a volume decrease three times as high as linear methods. This was not always the case, mostly due to inconsistencies of at least one of the confronted methods [Bar 99], [Ham 03], [Bjø 04], [Lur 05], [San 06], [Lur 07], [Bou 08a]. To avoid artifacts, paraffin oil was recommended instead of water for measuring the volumetric autogenous shrinkage of an immersed sample by the buoyancy method, i.e. by hydrostatic weighing [Lur 07], [Bou 08a].

In some investigations horizontal and vertical linear measurement led to notable differences, however, the evaluation in all cases appears to have included the chemical shrinkage prior to self-desiccation [Bou 06], [Mou 06b], [Bou 08a]. It was shown that the linear autogenous shrinkage after final setting does not significantly depend on the direction of measurement [Tia 08]. If vertical measurement methods are used, the length of the container should be much greater than its width. Otherwise the horizontal shrinkage component may influence results before and around final setting.

Measurement errors may also result from insufficient sealing, especially at very early age as more water is not yet bound. Loss of moisture inevitably leads to drying shrinkage. A maximum mass loss of 0.01 % with respect to the specimen weight was reported in [Sta 04], corresponding to roughly 0.16 % of the contained water. Similar results were obtained in own investigations [Epp 08b]. The accurate measurement of such small weight changes requires a very good balance, especially if the sample is located in comparably heavy measurement setups. No general agreement on the allowable moisture loss does exist, yet.

## 2.4 Corrugated tube method

The corrugated tube method was introduced by Jensen and Hansen in 1995 [Jen 95a] and recently standardized by ASTM [ASTM 2]. It is the most established method for measuring the autogenous shrinkage of cement paste and mortar under quasi-isothermal conditions [Epp 08b]. Numerous investigations carried out over a period of more than 15 years have shown its suitability for this purpose [Jen 96], [Jen 99], [San 07], [Mia 09]; own investigations of the method's precision are described in section 4.4.3.2. The basic setup described in the ASTM standard C1698 consists of the following items: (a) a corrugated tube mold made of plastic with a length of 420 mm and an outer diameter of 29 mm (Figure 2–7); (b) a testing frame ('dilatometer bench') on which the filled corrugated mold is placed; (c) a measuring gauge with a minimum resolution of 2.5 µm/m and a measuring range of at least 10 mm.

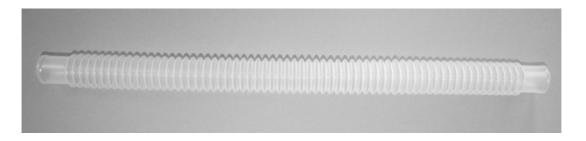


Figure 2–7 Corrugated tube mold, length: 420 mm, outer diameter: 29 mm

The corrugated mold's stiffness in longitudinal direction is very low. Thus the longitudinal deformation is virtually unrestrained. The recommended maximum grain size is 4.75 mm. The ASTM standard implies quasi-isothermal conditions. Due to the relatively small diameter of the tube, the temperature of a cementitious material inside the tube typically does not increase by more than 2 K. The method presently cannot be used for tests under non-isothermal conditions. It would be very difficult to precisely account for the influence of the thermal expansion of the tube since plastics have a much higher coefficient of thermal expansion than cement based materials. Notably, the standard recommends not to use water for controlling the temperature of the specimen since the mold may not be completely impermeable to water.

The test procedure is comparably simple. Only the process of filling can be critical. Due to the small inlet diameter of approximately 20 mm it can be difficult to fill the corrugated tubes if paste or mortar are stiff or sticky. Ultra-high strength concretes often are sticky. This can also lead to the inclusion of air bubbles (Figure 2–8). It is noted that larger corrugated tubes do exist as well, however they were not tested as comprehensively as the smaller ones, yet.



Figure 2–8 Incompletely filled corrugated tube

The standardized procedure is intended for discontinuous readings. The first reading is taken at final setting. Before further readings can be taken the tested material needs to become sufficiently hard so that the handling of the specimen does not significantly alter its length anymore. The precision of the discontinuous readings can be further improved by the use of a reference bar made of invar, which, together with the aforementioned items, is provided by the manufacturer of the test setup [Germann].

Continuous readings require a somewhat different setup. The testing frame should accommodate at least two or three molds at the same time. An identical number of measuring gauges is required, each of them connected to a data logger. Furthermore, one end of the mold needs to be fixed to the frame; alternatively the deformation can be measured at both ends of the mold.

The fact that the corrugated tube method was standardized by ASTM and adopted by many researchers will eventually lead to more reliable results and improved comparability, even though the time-zero issue (2.3.2) will continue limiting the precision. Other methods for measuring the autogenous shrinkage can be validated against the corrugated tube method (4); given that samples come from the same mix and that temperature does not differ notably, the time-zero issue can be disregarded in such comparative investigations.

# 2.5 Influencing parameters

## 2.5.1 Concrete composition

Water-cement and water-binder ratio govern the autogenous shrinkage of hardening cement and concrete. Significant autogenous shrinkage occurs if less water is available than would be required for the complete hydration of all cementitious constituents. This usually is the case below water-cement ratios of roughly 0.40 to 0.45. The autogenous shrinkage of pure cement paste with very low water-cement ratios between 0.15 and 0.3 can reach more than two thousandth, whereas the strain measured for ultra-high strength concrete with variable cement paste volume typically is less than one thousandth, cf. e.g. [Lur 03c] and [Epp 09]. As the absolute values can depend very much on the determination of time-zero (2.3.2) and since the measurement of the autogenous shrinkage in the past was carried out by numerous partly very different test methods with unknown precision (2.3.1, 2.3.3), here the further focus is not on absolute values but on the influence of the essential parameters.

The autogenous shrinkage increases with decreasing water-cement ratio (or water-binder ratio); it begins the earlier the lower this ratio is (**Figure 2–9**), cf. also [Taz 95], [Naw 04], [Sch 04]. A close correlation of water-cement ratio and autogenous shrinkage strain at the age of 28 days was found for a wide range of water-cement ratios [Bar 06].

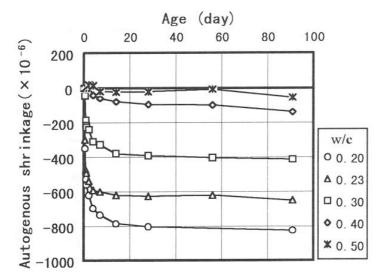
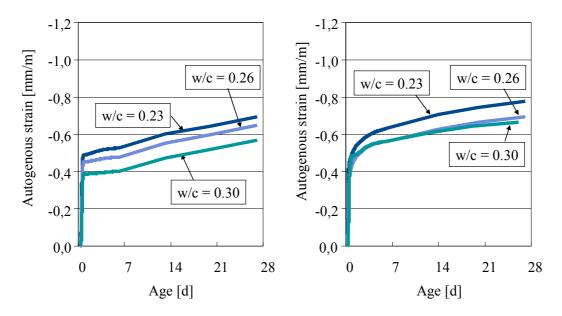


Figure 2–9 Autogenous shrinkage of cement pastes with different water-cement ratio [Taz 95]

Similar results were obtained for variable water-cement ratios of high and ultra-high strength concrete [Zha 03], [Epp 08a] (Figure 2–10). Notably, to compare the effect of different water-cement ratios on autogenous shrinkage of concrete, the cement content should be varied, not the water content. Otherwise the changing volume fraction of hydrating paste would have to be accounted for [Epp 08a].



**Figure 2–10** Autogenous shrinkage of ultra-high strength concretes with different water-cement ratio at constant water content; silica fume content: 16.2 % of cement; **left:** cement A (CEM I 52,5 R, 2.0 % C<sub>3</sub>A, 14.3 % C<sub>4</sub>AF), **right:** cement B (CEM I 52,5 R, 8.4 % C<sub>3</sub>A, 2.7 % C<sub>4</sub>AF); fineness of both cements approx. 4700 cm<sup>2</sup>/g [Epp 08a]

Type and fineness of cement significantly influence the autogenous shrinkage, too. In experiments with cement pastes made with different cements, the autogenous shrinkage increased with increasing content of C<sub>3</sub>A and C<sub>4</sub>AF [Taz 97], [Taz 97b]. Similar results as to the influence of C<sub>3</sub>A were found for ultra-high strength concrete [Epp 08a] (Figure 2–10). From thermodynamic calculations it was concluded that the hydration of C<sub>2</sub>S is slowed down at low relative humidity. Therefore high proportions of C<sub>2</sub>S may restrict autogenous shrinkage [Jen 95b]. Higher fineness of cement accelerates hydration and generally increases autogenous shrinkage [Taz 97], [Naw 04], [Ben 01b].

In many investigations the addition of silica fume increased the autogenous shrinkage, and higher proportions of silica fume led to higher autogenous shrinkage over a wide range of dosages [Jen 95c], [Jen 96], [Per 97], [Zha 03], [Epp 08a]. Substitution of Portland cement by ground granulated blast furnace slag tended to result in decreased autogenous shrinkage [Yan 07], [Epp 09]. Yet, opposite results, for instance for ground granulated blast furnace slag with high fineness, were reported as well [Taz 97b], [Jia 05], [Lee 06]. As hydration is slowed down by ground granulated blast furnace slag, autogenous shrinkage in part may be merely postponed. Fly ash and metakaolin reduced the autogenous shrinkage [Naw 04], [Sta 04], [Yan 07]. It was shown that supplementary cementitious materials alter the pore system and thereby influence shrinkage (cf. [Bui 87], [Iga 05], [Med 09] and 2.1).

Superplasticizers considerably retarded the hydration and influenced the autogenous shrinkage at early and very early age, whereas the autogenous shrinkage at later age did depend less on type and amount of superplasticizer [Taz 95], [Naw 04], [Epp 08a]. For ultra-high strength concrete with two different superplasticizers Schachinger measured autogenous shrinkage strains of 0.8 and 1.2 mm/m at the age of 1 d [Sch 07]. Air-entrainment by apposite chemical admixtures did not significantly influence autogenous shrinkage [Per 97].

# 2.5.2 Temperature

Temperature considerably influences the autogenous shrinkage [Lou 00]. However, no generally valid equations were found to quantify this influence, yet. It appears to be rather unsystematic (Figure 2–11) [Lur 03b] or at least not linear (Figure 2–12) [Jen 99]. As different temperature levels change the onset of autogenous shrinkage (time-zero), its determination is a potential source of errors in such investigations (cf. 2.3.2).

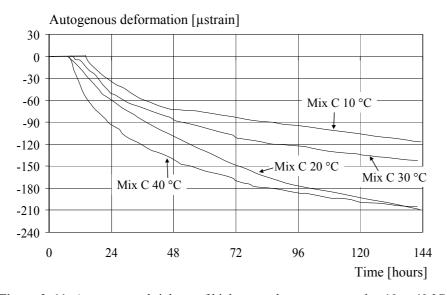


Figure 2–11 Autogenous shrinkage of high-strength concrete, cured at 10 to 40 °C (from [Lur 03b])

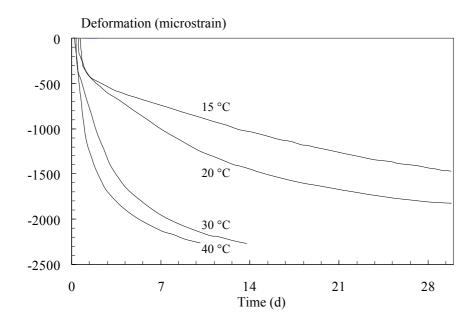


Figure 2–12 Autogenous deformation from final setting of cement past made of white Portland cement (w/c = 0.30, 20 % silica fume addition), (from [Jen 99])

The influence of temperature on the development of mechanical properties of normal strength concrete at early age can be modeled by so called maturity concepts. The maturity concept is based on the correlation of concrete properties and temperature-dependent hydration progress. The degree of hydration usually is given as a function of time and of the rate of cement hydration. The rate of cement hydration is a function of temperature. It is generally assumed that the influence of temperature on the rate of hydration does not depend on the degree of hydration and that it can be expressed by a constant activation energy [Jen 99]. The activation energy is a parameter within the Arrhenius equation which describes the influence of temperature on the rate of a chemical reaction.

It was shown that the conventional maturity concept as applied to compressive strength and other mechanical properties of normal strength concrete is not generally applicable to high-strength concrete [Jen 99], [Zha 05]. One general obstacle is that chemical reactions of frequently used supplementary cementitious materials proceed at a rate different from that of pure Portland cement hydration. Thus, no single activation energy may be sufficient to describe maturity [Jen 99], [Zha 05].

Furthermore, Zhang et al. observed that the activation energy depends on the property to be described. In their experiments with high-strength concrete the activation energy also changed with temperature [Zha 05]. A reverse maturity method was suggested according to which the activation energy is determined from different specimens which have achieved the same level of a given property under different curing temperatures [Zha 08].

Jensen and Hansen substantiated that the conventional maturity concept cannot be applied to autogenous shrinkage. The authors argued that the surface tension of water as well as the dissolution of salts does not only depend on the rate of reaction but also on the temperature range. Both parameters are known to influence self-desiccation (2.1). The authors' experiments confirmed these theoretical considerations [Jen 99].

Several other experimental investigations likewise revealed that the conventional maturity concept cannot be applied to autogenous shrinkage [Bjø 00], [Lur 03b], [Bar 04], [Per 05], [Mou 06], [Mou 06b]. Contrary to that, a good correlation between degree of hydration and autogenous shrinkage was found for a high-strength concrete [Sta 06]. Some affirmative results have also been reported for the case of plain cement paste: The maturity concept was successfully applied within a temperature range of 10 °C to 40 °C [Tur 02]. However, specimens were prepared in latex condoms and continuously weighed under water. Latex condoms are not sufficiently impermeable to water [Lur 05].

In another investigation the autogenous shrinkage was divided into two phases. A good correlation between maturity equivalent age and autogenous shrinkage of the cement paste was found for the second phase [Ter 08], [Ter 08b]. Elevated curing temperature may accelerate autogenous shrinkage in the beginning and lead to lower autogenous shrinkage at later ages [Kam 07]. The application of maturity concepts is continued, mainly because no better alternative has been established, yet [Miy 01] [Suz 05], [Yan 07].

### 2.5.3 Specific countermeasures

Several types of chemical admixtures were successfully used to counteract autogenous shrinkage, mainly shrinkage reducing admixtures, water-repellent treated powders and expansive admixtures [Taz 97b], [Sub 02], [Ich 05], [Suz 05], [San 07], [Wei 08]. The effect of shrinkage reducing admixtures sometimes was entirely attributed to reduced surface tension of pore water, however, other mechanisms may be involved as well [San 10], [Sch 08].

Furthermore, internal curing was found to be an effective method for mitigating self-desiccation and autogenous shrinkage. Internal curing comprises all measures which provide a reserve of water which is released when mixing water is gradually consumed by hydration. Different materials with the ability to store significant amounts of water have proven effective in this regard, mainly superabsorbent polymers [Jen 01b], [Jen 02], [Mec 06], [Mec 08] and light-weight aggregates [Ben 01a], [Zhu 02], [Cus 05], [Hen 09], furthermore recycled materials like porous ceramic waste aggregates [Suz 09]; cf. also [Ben 08a]. In 2006 a complete symposium was dedicated to the reduction of autogenous shrinkage by internal curing [RIL 06], followed by a conference on the use of superabsorbent polymers in 2010 [RIL 10].

## 2.6 Summary and conclusions with respect to the own work

Autogenous shrinkage is a consequence of hydration and consequently considered part of the chemical shrinkage. Self-desiccation occurs in concretes with very low water-cement ratio, especially in ultra-high strength concrete, leading to a uniform macroscopic volume decrease. While it is clear that a certain solidification of the pore structure is a precondition to that process, the precise onset of the autogenous shrinkage is difficult to determine. The most common criteria for defining such an onset are initial and final setting, leading to different results though. From a practical point of view it is essential to obtain the stress-relevant shrinkage. Parallel stress tests therefore are helpful in finding the required threshold, commonly called 'time-zero'. This practice was chosen for this study.

Micro-analytical models that would allow for predicting the autogenous shrinkage from the properties of the constituents and the concrete temperature do not exist, yet. Hydration in general and the development of the pore structure in particular are complex processes whose influence on autogenous shrinkage is only partly understood. Especially the influence of temperature has not been clarified, yet. Conventional maturity concepts do not seem to apply. Moreover, autogenous shrinkage, even if not externally restrained, depends on the viscoelastic behavior of the hardening pore structure being particularly difficult to examine.

In the framework of a more pragmatic phenomenological approach aiming at the assessment of the autogenous shrinkage cracking propensity, the free autogenous shrinkage is primarily an auxiliary quantity, its precise measurement being the most important issue. The autogenous shrinkage of very fluid pastes and mortars can be measured by means of the corrugated tube method. However, sticky or stiff mixes are difficult to fill into the tubes, and the method is limited to isothermal conditions. An alternative test method without these limitations would signify a substantial methodological improvement. However, a thorough validation would be required in order not to just add another test method to the multitude of existing methods largely generating hardly comparable data.

Several measures for reducing the autogenous shrinkage have been suggested, most importantly internal curing, shrinkage-reducing admixtures and a modification of the binder. It is one of the advantages of the chosen phenomenological approach that their effect on the cracking propensity can be directly investigated. Furthermore, such measures can provide for a wide range of results, potentially revealing whether the employed test methods are generally suitable or not. Therefore it was decided to also consider concrete compositions with superabsorbent polymers, with a shrinkage reducing-admixture and with a large amount of blast furnace slag.

#### 3 Concretes used in the own investigations

## 3.1 Preliminary remarks

This chapter comprises the description of the concrete compositions that were used in the own experiments as well as the most relevant mechanical properties. As will be discussed below, the options for evaluating restrained ring tests are considerably enhanced if additional concrete properties are known (cf. 6.3.2). The splitting tensile strength was tested as a measure for the resistance to cracking. The modulus of elasticity, tested by means of the resonance frequency method, was required to quantify the influence of creep and relaxation on stresses due to restrained autogenous shrinkage. The compressive strength was to reveal whether the concretes achieved the required strength to be considered of ultra-high strength. Furthermore, the coefficient of thermal expansion was tested as it was required for the evaluation of some additional tests under non-isothermal conditions (cf. 4.6, 6.4.7.2). Finally, results of isothermal calorimetry are provided, later used to analyze the period of maximum cracking propensity.

Most tests were conducted under quasi-isothermal conditions. If not otherwise mentioned, storage and set temperature during all tests was 20 °C. The concrete temperature in the specimens increased by not more than 2 K and typically by less than 1.5 K. Due to the small concrete specimens used, no additional measures for cooling were required.

## 3.2 Concrete compositions

The compositions of the ultra-high strength concretes used in this study are given in Table 3–1.

 Table 3–1
 Concrete compositions

Concrete composition 1)	Cement content (c)	Mixing water (w) <sup>2)</sup>	w/c	Silica fume (s) <sup>3)</sup>	w/(c+s)	Superplasticizer	Quartz sand <sup>6)</sup>	Quartz powder <sup>7)</sup>	SAP <sup>8)</sup>	SRA <sup>9)</sup>	GGBFS <sup>10)</sup>
Con	kg/m³	kg/m³	-	-	kg/m³	kg/m³	kg/m³	kg/m³	kg/m³	kg/m³	kg/m³
1A	800	168	0.230	130	0.198	24 4)	1019	220	-	-	-
1A SAP	800	168	0.230	130	0.198	24 4)	1019	220	2.4	-	-
1A SRA	800	168	0.230	130	0.198	24 4)	1000	220	-	7.56	-
1A BFS	200	176	-	130	-	13 4)	1019	220	-	-	600
1B	800	168	0.230	130	0.198	24 4)	1019	220	-	-	-
1A SP2	800	168	0.230	130	0.198	32 <sup>5)</sup>	1019	220	-	-	-

- 1) upper case 'A" and 'B" indicate the use of cement A and B, respectively (cf. 3.1.1)
- 2) de-ionized tap water
- 3) all concretes except for 1A BFS: 16.2 mass-% of c
- 4) superplasticizer 1 (all concretes except for 1A BFS: 3.0 mass-% of c)
- 5) superplasticizer 2 (4.0 mass-% of c)
- 6) particle size: 0.125 mm 0.5 mm
- 7) particle size: 0 mm 0.125 mm
- 8) superabsorbent polymers (0.3 mass-% of c)
- 9) shrinkage reducing admixture (4.5 mass-% of w)
- 10) ground-granulated blast furnace slag

Concretes had a maximum grain size of 0.5 mm (quartz aggregate) and did not contain any fibers. Fibers would have complicated the investigation of basic methodological issues. Larger aggregates would have necessitated larger specimens, thereby leading to more notable thermal effects and increasing the general test effort. Different measures for influencing the autogenous shrinkage were taken to provide for a wide range of shrinkage results. This was supposed to facilitate the clarification of methodological questions. Most investigations involved concrete 1A (reference), concrete 1A SAP (with superabsorbent polymers), concrete 1A SRA (with shrinkage reducing admixture) and concrete 1 BFS (with a high proportion of ground granulated blast furnace slag). Concretes 1A and 1B differed only in the cement that was used (cements A and B). Concrete 1A SP2 contained an alternative superplasticizer. All concretes contained the same amount of (de-ionized) water. The water content of the superplasticizer was accounted for in the calculation of the mixing water. The water-cement ratio of all concretes except for concrete 1A BFS was 0.23. The content of silica fume was 16.25 % with respect to the cement content, again with the exception of concrete 1A BFS. As was outlined above, hydration under these conditions remains incomplete. The theoretical maximum degree of hydration of reference concrete 1A was approximately 0.52 (calculated according to [Mil 66], silica fume considered as cement).

#### 3.3 Constituents

#### **3.3.1** Cement

Two Portland cements were used in the own experiments, both originating from German cement works and conforming with the European standard EN 197-1 (Cement – Part 1: Composition, specifications and conformity criteria for common cements). They will be referred to as cement 'A" and 'B". Cement A was a CEM I 52,5 R-HS/NA. The abbreviation 'NA" is specified in the German standard DIN 1164-10 (Special cement - Part 10: Composition, requirements and conformity evaluation for special common cement) and stands for a sodium oxide equivalent below 0.6 %. Cement B was a CEM I 52,5 R. It also went below the mentioned alkali limit, however was not specified accordingly. Cement A and B, being similar in many respects, notably differed in the content of aluminate phases. The C<sub>3</sub>A content of cement A and B was approximately 2.0 % and 8.0 %, respectively; in turn, the content of C<sub>4</sub>AF was higher with cement A. The water demand of cement B was higher than that of cement A. The specific Blaine surface of both cements was approximately 4.700 cm<sup>2</sup>/g. The cements' chemical composition as well as basic granulometric and physical parameters can be found in the Annex (Table 9-1). The differences among the individual batches of each cement were within common limits. If not indicated otherwise, comparisons and conclusions presented herein are based on test series throughout which the same batch of cement was used.

#### 3.3.2 Ground-granulated blast furnace slag

One type of ground-granulated blast furnace slag was used. It conformed with the European standard 15167 (Ground granulated blast furnace slag for use in concrete, mortar and grout). Its chemical composition was tested, results are given in the Annex (Table 9–1).

#### 3.3.3 Silica fume

One type of silica fume powder was used, conforming with the European standard EN 13263 (Silica fume for concrete). It had a specific BET surface of approx. 20 m²/g and a content of amorphous silicon dioxide of approx. 98 % (data provided by the manufacturer).

#### 3.3.4 Admixtures

Two superplasticizers, one shrinkage reducing admixture and one type of superabsorbent polymers were used. Both superplasticizers were polycarboxylate ether based and had a solids content of approximately 35.0 mass-%. The shrinkage reducing substance in the shrinkage reducing admixture was dipropylene glycol tertiary butyl ether. The superabsorbent polymers were of the cross-linked poly-acrylic acid type and more than 95 % of the particles were smaller than 63 µm. This information was provided by the manufacturers. In own experiments with the superabsorbent polymers a 22fold weight increase after 30 min immersion in synthetic pore solution was observed. The solids content of superplasticizer 1 used in most tests was determined to 35.0 mass-%, confirming the data provided by the manufacturer.

# 3.3.5 Aggregates

Quartz sand and quartz powder were used as aggregates. The nominal particle size of the quartz sand ranged from 0.125 mm to 0.5 mm, whereas the quartz powder contained particles up to a diameter of 0.125 mm. The aggregates were delivered and maintained in a dried state. Their density and coefficient of thermal expansion was 2.65 kg/dm³ and  $14\cdot10^{-6}*K^{-1}$ , respectively. The silicon dioxide content was above 99 % (data provided by the manufacturer).

## 3.4 Mixing

Materials were stored in airtight containers at a temperature of approximately 20 °C. Air temperature and relative humidity in the room the concrete was mixed in were 20 °C  $\pm$  2 °C and 65 %  $\pm$  5 %, respectively. A mortar mixer as per EN 196 with a nominal capacity of five liters and two rotation speeds (140 and 285 rotations/min) was used. The actual maximum concrete volume was approximately 1.5 liters. If more than 1.5 liters of concrete were required, two mixers were used simultaneously. Mixer type and mixing procedure are shown in **Figure 3–1**.



1A, 1B, 1A SP2	· · · · · · · · · · · · · · · · · · ·	1/3 SP, ment	2/3 SP	aggregates, silica fume		
1A SAP	ĺ í	1/3 SP, ment	2/3 SP	aggreg silica t	SAP + 70 g quartz sand	
1A SRA	· · · · · · · · · · · · · · · · · · ·	1/3 SP, nt, <b>SRA</b>	2/3 SP	aggregates, silica fume		
1A BFS	·	1/3 SP, GGBFS	2/3 SP	aggregates, silica fume		
Speed	low high		high	low high		high
Duration	1 min	1 min	1 min	1 min	1 min	1 min

Figure 3–1 Mortar mixer (left) and mixing procedure (right), (SP: superplasticizer)

Mixing in all cases took six minutes. First the mixing water and one third of the superplasticizer were premixed and filled into the mixer bin, followed by cement and, where applicable, other constituents. The partial addition of superplasticizer in preliminary tests was found to increase the workability of the concretes. The residual amount of superplasticizer was added after 2 minutes while mixing was continued. Then quartz sand, quartz powder and silica fume were added through the feeder while mixing proceeded at the lower speed for one minute. The

superabsorbent polymers were premixed with a small portion of the quartz sand to ensure a good distribution in the mix; no extra water was used in order not to alter the water-cement ratio. As a consequence the workability, which was not systematically tested in this study, notably decreased in the course of time. To maximize the period of good workability, the superabsorbent polymers were added one minute before the end of mixing.

## 3.5 Basic properties

## 3.5.1 Compressive strength

The compressive strength was tested on small cubes (40 x 40 mm³). Additionally, the compressive strength of concrete 1A was tested on cylinders with diameter and height of 150 mm and 300 mm, respectively, to investigate the influence of the specimen geometry. Molds were made of steel and treated with a release agent before being filled. The concrete was compacted on a vibrating table for at least one minute after partial (cylinders) and complete filling. The specimens were immediately covered with a plastic sheet and a glass plate to minimize the loss of moisture. Then the filled molds were stored at 20 °C and 95 % RH. Specimens were demolded after 24 h and stored in water (20 °C) until shortly before the test. The cylinders were ground to achieve plane-parallel surfaces. Before testing, the specimens' weight and dimensions were measured. The compressive strength tests were carried out on two class I testing machines with a capacity of 300 kN and 6 MN, respectively. The compressive stress was increased by 0.5 MPa/s until failure. A minimum of three specimens per age was tested. The results of all tests are given in the Annex (Table 9–2).

The ratio of the strength of cylinders (150 mm x 300 mm) to the strength of cubes (40 x 40 x 40 mm<sup>3</sup>), tested with concrete 1A, was 0.94. To account for size effects, all results obtained from cubes were multiplied by this value. The resultant compressive strength of the concretes 1A, 1A SAP, 1A SRA and 1A BFS at the age of 2, 7, 28 and 56 days is shown in **Figure 3–2**.

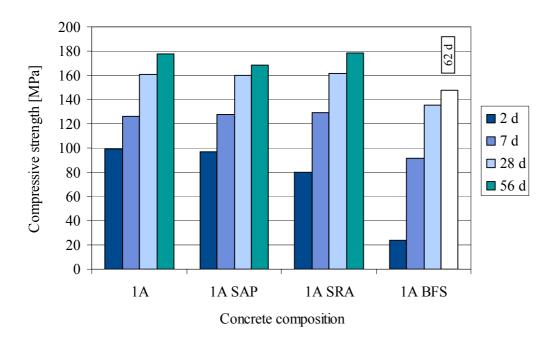


Figure 3–2 Compressive strength development, tested on cubes (40 x 40 x 40 mm<sup>3</sup>), values multiplied by 0.94

All concretes except for concrete 1A BFS (148 MPa at 62 days) achieved ultra-high strength as defined in Germany, i.e. 150 MPa or more. The compressive strength of the reference concrete 1A at 2, 28 and 56 days was 99, 161 and 178 MPa, respectively. In comparison to concrete 1A the strength development of concrete 1A BFS was much slower and final strength was significantly lower. The shrinkage reducing admixture led to a slightly slower hydration and strength development (1A SRA); however, final strength was as high as that of concrete 1A. The addition of superabsorbent polymers caused a slightly lower compressive strength at 56 days, while the earlier strength development was not significantly affected (1A SAP).

### 3.5.2 Splitting tensile strength

The splitting tensile strength was tested on specimens with diameter and height of 37 mm and 50 mm, respectively. Specimens were cast in cylindrical polypropylene molds with a wall thickness of 0.6 mm. The filled molds were tightly closed with a cap and stored at a temperature of 20 °C. Specimens were demolded and cut to length by means of a water-cooled saw shortly before testing. The specimens' weight and dimensions were measured. The diameter was callipered; no deviations from circularity were found. The load-controlled splitting tension tests were carried out on a class I testing machine with a maximum capacity of 50 kN. Photos of the specimen preparation and the testing procedure can be found in the Annex (Figure 9–1). Masonite load bearing strips with a thickness and width of 3 mm and 7 mm, respectively, were used for load distribution. A minimum of three specimens was tested. The rate of stress increase in the principal tests was 3 MPa/min as per EN 12390-6. In addition, a rate of stress increase of 0.2 MPa/min was used with concretes 1A and 1A BFS to investigate the influence of the rate of stress increase.

The splitting tensile strength of concretes 1A, 1A SAP, 1A SRA and 1A BFS until the age of approximately 40 h, tested at a rate of stress increase of 3.0 MPa/min, is shown in Figure 3–3. Bars indicate the range of the individual test results. The splitting tensile strength of the reference concrete 1A increased from 0.28 MPa at 9 h to 4.1 MPa at approximately 16.5 h and 9.8 MPa at 40 h. In comparison to concrete 1A the strength development was much slower with concrete 1A BFS and slightly slower with concrete 1A SRA. The splitting tensile strength of concretes 1A and 1A SAP developed similarly. Not shown in Figure 3–3, concrete 1A BFS reached a splitting tensile strength of 7.5 MPa at the age of 163 h. The results of all individual tests are given in the Annex (Table 9–3, Table 9–4). The mean coefficient of variation for all tests was 8.8 %.

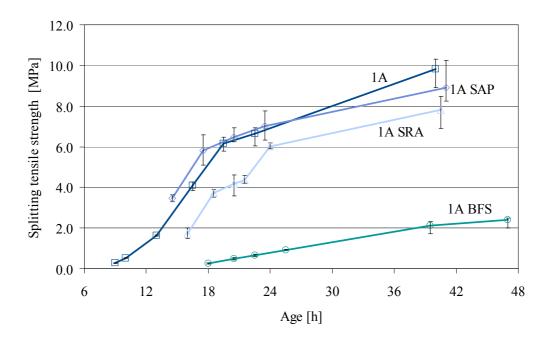


Figure 3–3 Splitting tensile strength, rate of stress increase: 3.0 MPa/min

As is shown in Figure 3–4 (1A) and Figure 3–5 (1A BFS), the splitting tensile strength was dependent on the rate of stress increase. Strength tended to result higher at the lower rate of stress increase. This tendency was confirmed by repeating some of the tests. At later age the influence of the stress rate on strength usually is vice versa, cf. e.g. [Car 94]. However, the splitting tensile strength is known to be sensitive to a number of test parameters [Roc 99], [Mal 07]. Possibly the failure mechanisms are different at very early age. Here, this important aspect could not be further looked into. Generally, the location of the crack initiation and the ratio of splitting to uniaxial tensile strength depend on concrete strength, specimen geometry and test setup [Mal 10]. This issue is discussed further in section 6.4.6.2.

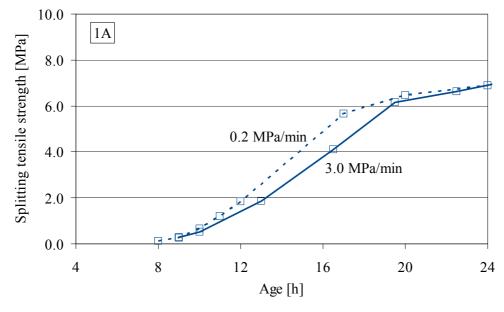


Figure 3–4 Splitting tensile strength of concrete 1A at different rates of stress increase

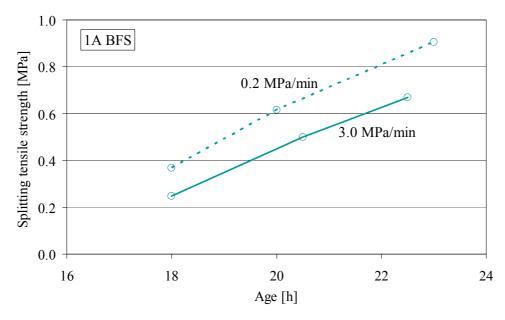


Figure 3–5 Splitting tensile strength of concrete 1A BFS at different rates of stress increase

# 3.5.3 Modulus of elasticity

The relative dynamic modulus of elasticity was determined by the resonance frequency method according to ASTM C 215. Two specimens (285 x 25 x 25 mm³) per concrete were tested. Because of the commonly very good repeatability, two specimens were considered sufficient. The differences between results of companion specimens in fact were negligible.

Additionally, the static modulus of elasticity of concrete 1A was tested according to DIN 1048-5. A class I testing machine was used. The lower compressive stress was 0.5 MPa and the upper stress corresponded to 1/3 of the expected compressive strength (Figure 3–6, right). The elastic modulus was calculated from the respective strains at the end of the third compression cycle. Strains were measured by means of two DD1 clip gauges (Figure 3–6, left). Tests were carried out on cylindrical specimens with diameter and height of 37 mm and 62 mm, respectively. Specimens were cast in polypropylene molds, cut to length by means of a water-cooled saw and grinded to achieve even and plane-parallel surfaces. The surface of the specimens was dry when the test was carried out. Three specimens per age were tested.



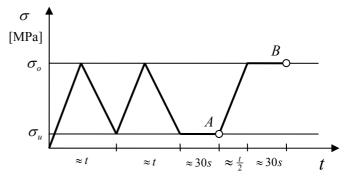


Figure 3-6 Testing of the modulus of elasticity: specimen with clip gauges (left) and stress scheme (right)

The development of the dynamic modulus of elasticity within the first 24 h is shown in **Figure 3–7**. The dynamic modulus of elasticity of reference concrete 1A and concrete 1A SAP was approximately 40 GPa at 24 h. In analogy to compressive and splitting tensile strength, the development was slower with concrete 1A SRA and much slower with concrete 1A BFS. Also, the values at 24 h were lower (1A SRA: 36 GPa, 1A BFS: 16 GPa). The development of the dynamic modulus of elasticity until approx. 300 h is given in the Annex (**Figure 9–2**).

Parallel tests of the static and dynamic modulus of elasticity of concrete 1A yielded a conversion factor of 0.84 (21 h) and 0.82 (24 h) (Figure 3–8). The stress analysis of restrained ring tests (cf. 6.4.5) was based on the results of the dynamic tests, multiplied by 0.83. This implies the assumption that the ratio of static to dynamic modulus during the first 24 h did neither depend on the age nor on the composition of the concrete.

As concrete in common restrained ring tests is under tension, it would have been more appropriate to use the tensile modulus of elasticity for the analysis. However, the tensile modulus is very difficult to determine at early and very early age. It possibly is higher than the compressive modulus, however, this is not yet clarified and unsure in particular for the very early age [Atr 03]. In recent investigations with a concrete very similar to the reference concrete 1A the tensile modulus was slightly higher at some test ages (roughly 5 to 10 %), however, there was no consistent relation and tests were carried out at 24 h or later [Bud 07].

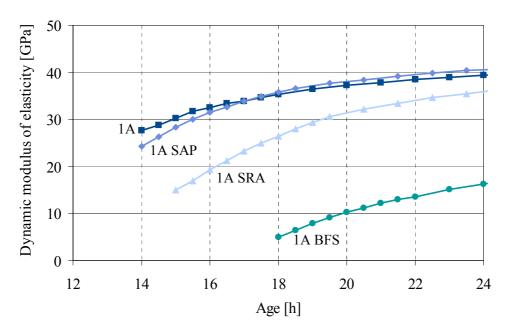


Figure 3–7 Relative dynamic modulus of elasticity until the age of 24 h

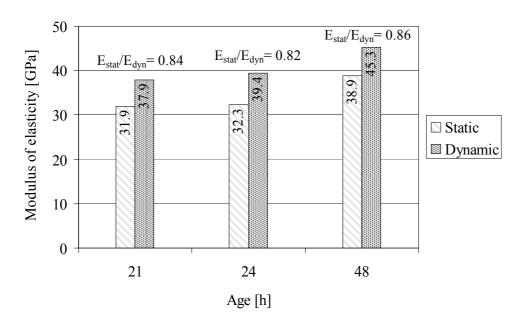


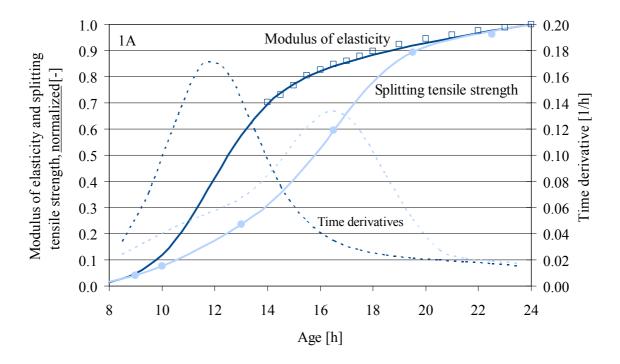
Figure 3–8 Static and dynamic modulus of elasticity of concrete 1A

#### 3.5.4 Analysis of mechanical properties

The mechanical properties are essential for the enhanced evaluation of restrained ring tests. The stress-strength ratio in this study is built on the splitting tensile strength, and the computation of theoretical-elastic stress and creep factor requires the modulus of elasticity (cf. 6.3.2). Since the less important compressive strength was tested only at later ages, it was disregarded at this point.

To facilitate the further computations, splitting tensile strength and modulus of elasticity were fitted to sigmoid functions. A common example of a sigmoid function is the logistic function, which has the shape of an S with two horizontal asymptotes. Sigmoid functions here were considered particularly suited for modelling the very early age development of mechanical properties since they can capture the initially very slow, then exponential and then again slow development. The required regression analysis was accomplished by means of the commercial software FindGraph.

The following figure shows the development of modulus of elasticity and splitting tensile strength of concrete 1A. To directly compare the two properties their development was normalized to the respective value at 24 h. The solid lines represent the fitting functions and the dashed lines the respective time derivative (Figure 3–9). As can be seen, the modulus of elasticity initially developed faster than the splitting tensile strength. From 14 h the splitting tensile strength rate was higher. By and large this was similar with the other concretes investigated. The respective figures for concretes 1A SAP, 1A SRA and 1A BFS can be found in the Annex (Figure 9–3, Figure 9–4, Figure 9–5).



**Figure 3–9** Relative dynamic modulus of elasticity and splitting tensile strength (tested at a rate of stress increase of 3.0 MPa/min), normalized to the 24 h-value, concrete 1A

At very early age the relatively low number of test results partly brought about considerable uncertainties in relation to the fitting curves. This concerned primarily concretes 1A SAP, 1A SRA and 1A BFS. Especially the input data for the period shortly after solidification needed to be chosen carefully in order to obtain reasonable curves; more test results would have been helpful. On the other hand it is critical to demold specimens at that point in time. One may easily damage them, possibly without noticing it.

In any case, the uncertainties in the fitting process did not compromise the further evaluation. The modulus of elasticity was not required for determining the cracking propensity, and the applicability of the splitting tensile strength, in particular at very early age, was comprehensively analyzed by additional measures (cf. 6.4.6). The evaluation of the influence of creep in restrained ring tests - here the modulus of elasticity is required - is still at a conceptual stage; potential inaccuracies in the fitting process represent a secondary issue. Nevertheless, more data of the very early development of mechanical properties should be gathered and used for developing dependable empirical models, if possible.

The mechanical properties of two ultra-high strength concretes were modelled in [Bud 07]. One of the concretes was very similar to the reference concrete 1A, however contained more superplasticizer and therefore hydrated more slowly [Bud 10]. The following figure shows the development of the uniaxial tensile strength as function of the test age (**Figure 3–10**). The strength was tested on cylinders with diameter and height of 80 mm and 300 mm, respectively. The specimens were sealed and stored at 20 °C until shortly before the test. Twenty-one tests were carried out between the age of 12 and 48 h. The uniaxial tensile strength started to develop at approximately 16 to 17 h. It developed rapidly between 20 and 40 h. Afterwards the development was much slower.

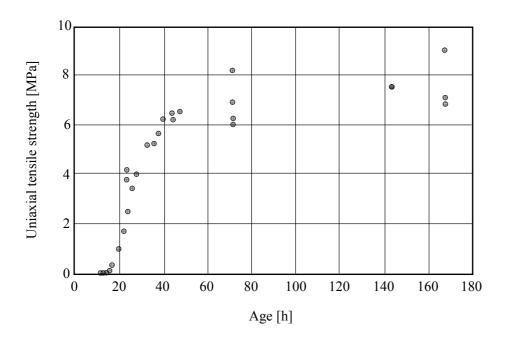
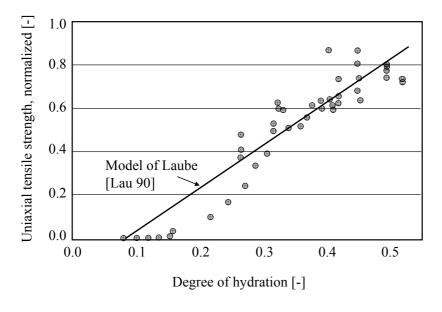


Figure 3–10 Uniaxial tensile strength of an ultra-high strength concrete similar to concrete 1A, from [Bud 07]

This data was evaluated by the authors according to the model of Laube which had been developed to describe restraint stress in mass concrete [Lau 90], cf. also [Gut 98]. To apply the model, the age was converted to a degree of hydration by means of formulas proposed by Jonasson [Jon 94]. To account for the incomplete hydration, a maximum degree of hydration of 0.55 was assumed. The model uses linear extrapolation of the tensile strength data to determine the degree of hydration at the onset of tensile strength development. The following figure shows the uniaxial tensile strength as function of the degree of hydration, normalized to the strength value at maximum degree of hydration which was approximately 9.5 MPa as per the linear model; the straight line represents the model, the single points are the measured strength values (Figure 3–11, [Mue 10a], [Bud 10]).



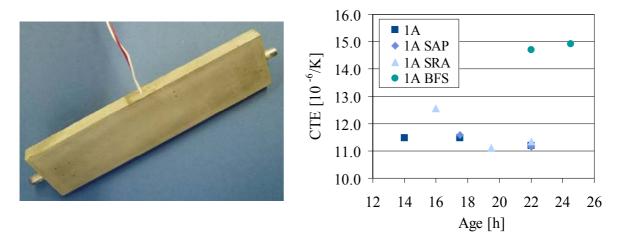
**Figure 3–11** Uniaxial tensile strength of an ultra-high strength concrete similar to concrete 1A, normalized to the value at maximum degree of hydration, as function of the degree of hydration, from [Bud 10]

The figure reveals that this linear approach may be suitable for describing the tensile strength at higher degrees of hydration but not for the very early age. One can speak of a reasonable agreement above a degree of hydration of approximately 0.3. According to the model, at this point the uniaxial tensile strength exceeded 6.0 MPa. As can be seen from the previous **Figure 3–10**, the corresponding age is roughly 40 h, i.e. much later than the period relevant for investigating the maximum cracking risk (cf. also 6.4.8). At a degree of hydration of 0.2 the model overestimated the actual uniaxial tensile strength by more than 100 %. Hence, this simple model was, as expected, not appropriate for this study. No other suitable model was found in literature, either. A specific study would be required to more comprehensively investigate and to empirically model the mechanical properties from the point of solidification.

# 3.5.5 Coefficient of thermal expansion

The coefficient of thermal expansion of concrete was tested on slender prisms (160 x 40 x 10 mm³, **Figure 3–12**, left). The coefficient was required only for the evaluation of some additional tests under non-isothermal conditions (cf. 4.6 and 6.4.7.2). Three specimens per age and concrete were heated up in a water bath with a temperature of approx. 85 °C from approx. 20 °C to approx. 80 °C in 4 min. The temperature inside the prisms was continuously measured via a resistance temperature detector (Pt 100). The length of the specimens was measured before and after heating with a dilatometer setup as per ASTM 157. Shrinkage within the period of heating was neglected. The effect of the gauge studs made of steel was accounted for by subtracting their expansion (calculated with a coefficient of thermal expansion of 16 µm/m/K) from the measured deformation. Each prism was used only once.

The coefficient of thermal expansion of reference concrete 1A amounted to approximately 11.5·10<sup>-6</sup>/K at 14 h and 17.5 h and to 11.2·10<sup>-6</sup>/K at 22 h (**Figure 3–12**, right). Results for concrete 1A SAP were very similar. The partly higher values observed for concretes 1A SRA and 1A BFS may be due to the slower hydration and the different state of water [Loc 00]. The mean coefficient of variation of this test series was 7.1 %.



**Figure 3–12** Left: Specimen (160 x 40 x 10 mm³) with embedded resistance temperature detector for testing the coefficient of thermal expansion, right: results

# 3.5.6 Isothermal calorimetry

The heat of hydration of concrete 1A and 1A SRA was measured by isothermal heat flux calorimetry (TAM Air), mainly to obtain additional information on the reference concrete and to examine whether such data is helpful in the further analysis of the autogenous shrinkage cracking propensity (cf. 6.4.8). Concretes were mixed as shown in **Figure 3–1** and then filled into the test jar. The following figure shows the rate of heat of hydration in the first 24 h (**Figure 3–13**). In comparison to reference concrete 1A, the maximum rate with concrete 1A SRA was lower and occurred later, clearly indicating the considerable retardation caused by the admixture.

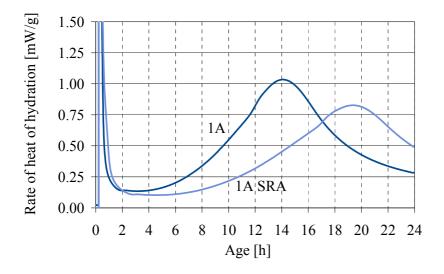


Figure 3–13 Rate of heat of hydration of concretes 1A and 1A SRA (isothermal heat flux calorimetry)

#### 3.6 Summary

Fine-grained, plain ultra-high strength concretes were tested as to their mechanical properties. Investigations focused on four concrete compositions, three of which being modifications of the reference concrete 1A with only Portland cement as binder and superplasticizer being the only admixture. Concrete 1A SAP additionally contained superabsorbent polymers, concrete 1A SRA a shrinkage-reducing admixture and concrete 1A BFS a large amount of ground granulated blast furnace slag, replacing 3/4 of the Portland cement. The concretes' composition was to provide for a wide range of results in order to facilitate the examination of test methods.

The influence of the specific constituents on the development of the mechanical properties was as expected. The overall development was retarded by the high amount of superplasticizer. The shrinkage-reducing admixture increased the retardation effect. The very slow development of mechanical properties of concrete 1A BFS was due to the large amount of ground granulated blast furnace slag. The superabsorbent polymers, which were added to the reference concrete without any extra water, did not significantly alter the development of the mechanical properties. For instance, the splitting tensile strength of concretes 1A and 1A SAP increased from virtually nil to 6.0 MPa between the age of 8 h and 18 h and thereafter to 7.0 MPa at 24 h. In comparison, the splitting tensile strength of concretes 1A SRA and 1A

BFS was 5.0 and 0.75 MPa at 24 h. The dynamic modulus of elasticity of concrete 1A, 1A SAP, 1A SRA and 1A BFS at 1 d was 40, 40, 36 and 16 GPa, respectively. The static modulus corresponded to 83 % of the dynamic modulus of elasticity. Furthermore, the coefficient of thermal expansion ranged from 11 to 15  $\mu$ m/m/K.

The analysis of the mechanical properties during the first 24 h revealed that the modulus of elasticity initially developed faster than the splitting tensile strength. Sigmoid functions may be more suitable than linear functions for describing the development over time in this period.

#### 4 Shrinkage cone method for measuring autogenous shrinkage

#### 4.1 Introduction

The efficient and precise measurement of the autogenous shrinkage is a precondition to the indepth analysis of restrained ring tests or other tests under restraint conditions. In the past, many test methods have been suggested and are still in use, many of them not validated though. Validation was hampered by the fact that it is practically impossible to determine a true value for the autogenous shrinkage. Recently, the corrugated tube method has received growing acceptance all over the world, however, it is suitable only for isothermal conditions. Another practical disadvantage is the small tube inlet which prevents sticky or stiff concretes from being poured into the tubes. Here, the development and validation of a new test method is described which is to represent an alternative to the corrugated tube method for measurements both under isothermal and non-isothermal conditions. The autogenous shrinkage of the main concretes used in this study is given as well.

# 4.2 Setup and measurement procedure

The shrinkage cone method for testing the autogenous shrinkage is based on the shrinkage cone apparatus [Schlei]. The method was introduced in 2008 [Epp 08b] and subsequently underwent some modifications which partly were described in [Epp 09]. The basic setup of the shrinkage cone apparatus consists of a laser unit which is mounted on a stand, and a cylindrical jar which forms a cone on the inside. Jars of two different sizes are available. In the own investigations the smaller jar was used that is recommended for paste or concrete up to a maximum particle size of 2 mm. The nominal height of the inner cone is 100 mm, the diameter of the opening is approx. 115 mm. The nominal volume is 350 ml. The temperature of the material can be controlled by a coolant that flows through the jar. In addition, the manufacturer provides cone-shaped protection sheets which minimize friction, and reflector pads made of plastic (Figure 4–1). The data is recorded in a data logger which can be connected to and controlled by a personal computer.



**Figure 4–1** Left: Shrinkage cone jar with protective sheet inside, placed under the laser unit; top right: laser unit with laser and optical sensor; bottom right: reflector pad as provided by the manufacturer

To test the *plastic* shrinkage of a cementitious material, it is first poured into the jar. Then the reflector is placed on top of the material and the jar is positioned under the laser unit so that the focused laser beam is reflected by the reflector pad. The diffusive part of the reflected light is received by an optical sensor. The distance to the reflector can be calculated by means of the triangulation principle. Lasers with different resolution are available. The lasers used in the own tests had a resolution of 0.6 and 0.1 µm. After the test the actual height of the specimen needs to be measured. It is the reference for the calculation of the linear strain.

Assuming deformations to be uniform, the cone geometry of the specimen ensures that the change of the recorded distance to the specimen surface corresponds to the linear length change. The ratio of radius r to height h is constant for any volume; therefore the ratio of two volumes  $V_1$  and  $V_2$  is equal to the third power of the ratio of the respective heights  $h_1$  and  $h_2$  (Figure 4-2). In a cylinder the relation of radius and height is variable prior to solidification. For this reason a cylinder is appropriate for the one-dimensional measurement of volume changes before setting only if it is much longer than wide.

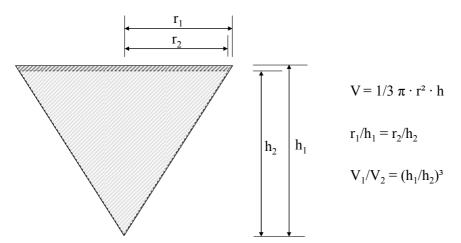


Figure 4-2 Cross section of two cones with different radius and height and identical angles. Equations describing the resultant volumes V<sub>1</sub> and V<sub>2</sub>

The shrinkage cone originally had been developed for measuring the plastic shrinkage [Kau 04]. To measure the autogenous shrinkage, it is required to prevent evaporation from the surface. A thin and transparent plastic film was used to cover the surface [Toppits]. Sealing the surface sometimes led to the formation of bubbles under the plastic film during the test, notably altering the position of the reflector pad when this was put on top of the film [Epp 08b]. To avoid this effect, the reflector pad was directly placed into the concrete. Prior to that, the film and the pad were tightly connected to each other by double-faced adhesive tape. A small part of the film was removed in the center of the pad to allow for the laser to directly hit the concrete surface. Pads made of concrete were used instead of the plastic pad to ensure a more neutral behavior in terms of thermal expansion. Diameter and height of the pads were approximately 3 mm and 18 mm, respectively. Before being used the pads were stored for at least one week to minimize the influence of shrinkage of the pad. Possible swelling when immersed into concrete could be neglected. In preliminary tests the concrete which was used for producing the pads (concrete 1A) was found to hardly take up any water.

The consecutive steps of test preparation and measurement are shown in Figure 4–3. Concrete was carefully compacted. The sealing film was additionally fixed to the jar's outside by double-faced adhesive tape (Figure 4–3, step 11). After the jar had been placed under the laser unit, the laser was focused on the reflector pad and the measurement was started. After the measurement, the film was removed and the specimen was taken out of the jar to measure its actual height. The height of the non-shrinking reflector pad was accounted for. The deformation measured by the laser was related to the effective height of the specimen to obtain the linear strain.

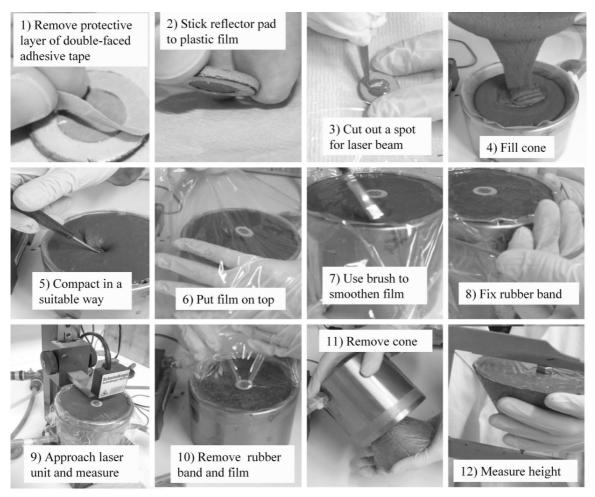


Figure 4–3 Measurement of autogenous shrinkage by the shrinkage cone method

The mean weight loss in the particularly relevant first 24 h was approx. 0.15 % with respect to the mass of water contained in the concrete sample. The influence of this mass loss on the measured strains was neglected. To further reduce it, an additional aluminum foil can be glued to the plastic film and fixed along the outer circumference of the jar by adhesive tape (Figure 4–4). This alternative is recommended in particular for tests at temperatures higher than 20 °C since higher temperatures may reduce the tightness of the plastic film.



Figure 4-4 Sealing of the jar with a combination of plastic and aluminum foil, fixed to the jar by adhesive tape

#### 4.3 Temperature control

The concrete temperature significantly influences the autogenous shrinkage (cf. 2.5.2). For that reason it is an essential requirement for any complete test method that the concrete temperature can be effectively controlled. The effectiveness of the temperature control of the shrinkage cone method, tested on concrete 1A, is illustrated in **Figure 4–5**. Without temperature control there was a significant increase in the centre of the cone of approximately 3.8 K due to the release of heat of hydration. The increase was approximately 0.7 K when a coolant (water, 20 °C) flowed through the jar. Furthermore, the cooling quickly reduced the relatively high temperature soon after mixing.

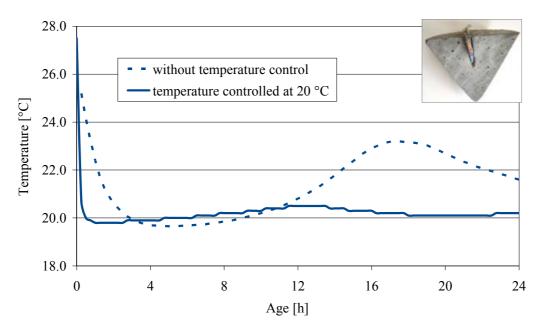


Figure 4–5 Temperature in the center of the cone with and without temperature control, concrete 1A

There are two alternatives to conduct tests under non-isothermal conditions. The first is to place the setup in a room with constant air temperature and to control the concrete temperature by means of the coolant that flows through the jar. Theoretically, in this case notable differences between the temperature in the centre of the specimen and at its surface may occur

since the temperature at the surface may be influenced by the air temperature which deviates from the temperature of the coolant. To investigate this, a non-isothermal test was carried out during which the temperatures in the centre and approx. 2 mm underneath the surface were measured. The temperature regime comprised some abrupt changes of temperature. A maximum difference between core and surface temperature of approximately 0.5 K was observed which can be considered negligible (Figure 4–6, left). Although the room temperature (20 °C) was lower than the concrete temperature, the temperature at the surface was slightly higher than in the centre of the specimen. However, the differences were hardly significant. Possibly the observed effect was due to the heat conduction of the aluminum foil placed on top of the surface (cf. Figure 4–4),

The second alternative is to place the complete shrinkage cone apparatus inside a temperature chamber. Then the temperature of the concrete can be controlled by the air that surrounds the apparatus. In this case, one concern may be the slow adaption of the air temperature by the concrete. The result of a respective test is shown in **Figure 4–6** on the right side. The test was carried out in an oven which normally is used for drying of materials. Despite the abrupt changes of air temperature the temperature in the centre of the specimen changed rather steadily. The maximum time shift between air and concrete temperature was approx. 1 h. If the air temperature could have been controlled more precisely, the control of the concrete temperature would have been better.

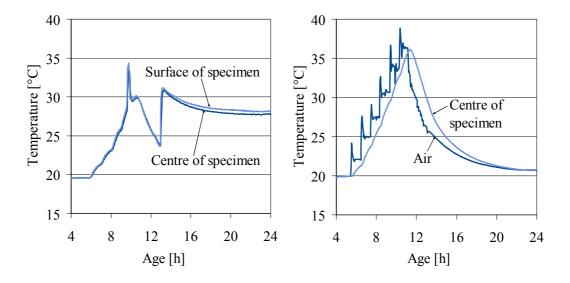


Figure 4–6 Concrete temperature in shrinkage cones: temperature control by coolant (left) and by air (right)

Hence, reasonable control of the concrete temperature can be achieved with both methods. However, if the complete apparatus is placed in a temperature chamber, the thermal deformations of each element (stand, base, laser, connecting parts) need to be accounted for in the calculation of the autogenous shrinkage. These elements are made of different materials, making a correction of thermal deformations very difficult.

Contrary to that, if concrete temperature is controlled by the coolant, only the thermal deformation of the bottom part of the jar needs to be considered. The geometry of the upper part, i.e. the area from the tip of the cone to the top, is not temperature dependent. The angle  $\gamma$  is constant for arbitrary temperatures.

A strongly exaggerated thermal expansion of the jar is shown in **Figure 4–7** to illustrate this effect. Under the assumption that the base on which the jar is placed does not deform, the position of the cone tip is altered only by the displacement t. The displacement can be calculated from the temperature change, the distance from the bottom to the tip of the jar (approx. 10 mm) and the coefficient of thermal expansion. The jar is made of an aluminum alloy with a coefficient of thermal expansion of approximately 23  $\mu$ m/m/K.

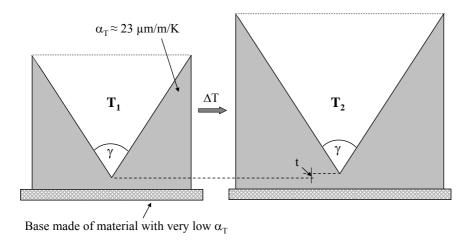


Figure 4–7 Shrinkage cone: thermal deformation of the jar at two different temperatures  $T_1$  and  $T_2$  (schematic)

In the following figure the jar is supposed to be filled with concrete. Jar and concrete are assumed to undergo a uniform increase of temperature. Shrinkage is neglected and thermal deformation of concrete is assumed to be uniform. Then the difference in the distance between laser and concrete surface  $(d_1-d_2)$  is the sum of the displacement t of the tip of the cone and the displacement a caused by the thermal deformation of the concrete specimen. If the coefficients of thermal expansion and the temperatures are known, a and t can be calculated and accounted for in the calculation of shrinkage.

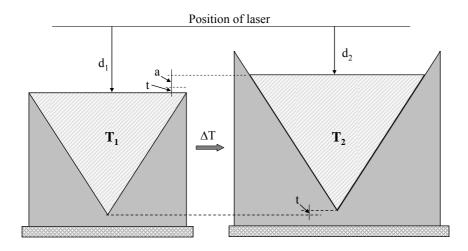


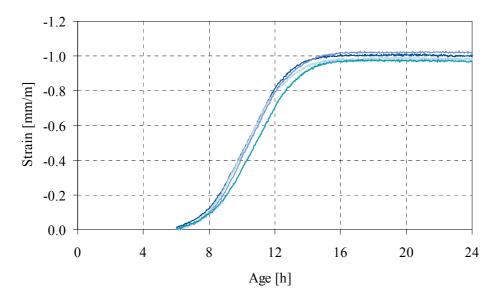
Figure 4–8 Shrinkage cone method: thermal deformation of jar and concrete (cf. Figure 4–7)

# 4.4 Precision under quasi-isothermal conditions

# 4.4.1 Repeatability

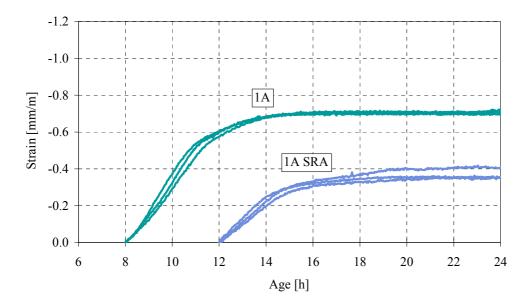
To investigate the repeatability of the shrinkage cone method, the autogenous shrinkage of concrete 1A was measured five times in a row under quasi-isothermal conditions (20 °C). Cement batch 2 was used for these tests (**Table 9–1**). The deformations in this case were evaluated from initial setting (approximately 6 h) in order to also cover the period which usually is more difficult to measure correctly (cf. 2.3).

The mean strain at the age of 24 h amounted to 0.99 mm/m (**Figure 4–9**). Generally, the scatter was low. The standard deviation at the age of 24 h was 0.02 mm/m and the coefficient of variation 2.0 %. The repeatability precision according to ISO 5725-2:2002 (assuming a 5 % error probability) was 0.056 mm/m, equivalent to 5.6 % with respect to the mean.



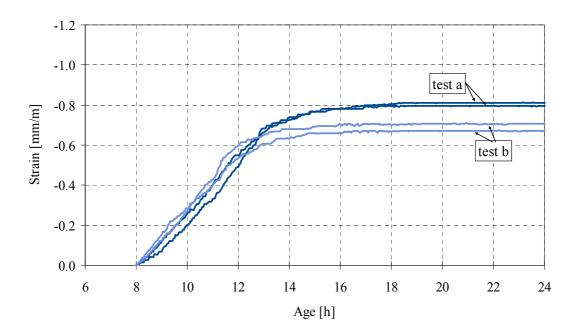
**Figure 4–9** Strain under sealed conditions between the age of 6 and 24 h measured five times in a row with the shrinkage cone method, concrete 1A

Further investigations were carried out with concrete compositions 1A and 1A SRA, partly involving a second laboratory (BAM, Federal Institute for Materials Research and Testing, Berlin, Germany). Both concretes were tested thrice with the shrinkage cone method under repeatability conditions. Cement batch 4 was used for this test series (cf. **Table 9–1**). The onset of stress development in restrained ring tests was used as time-zero (1A: 8 h, 1A SRA: 12 h). The results of the own investigations (Lab 1) are shown in **Figure 4–10**. Again the scatter was low.



**Figure 4–10** Strain under sealed conditions measured by the shrinkage cone method, onset of stress development in deformation-restrained tests as time-zero, three tests in a row with each concrete

At the time these investigations were carried out, only one shrinkage cone was available at German Cement Works Association. The second laboratory (Lab 2) was equipped with two shrinkage cone apparatus. The following figure shows the results of two out of three consecutive tests on concrete 1A, carried out with two shrinkage cones in parallel (Figure 4–11). A third test failed. The agreement of results obtained from the two mixes can be considered good, the agreement between the results of the single tests very good. Presumably the mixes had a slightly different shrinkage behavior.

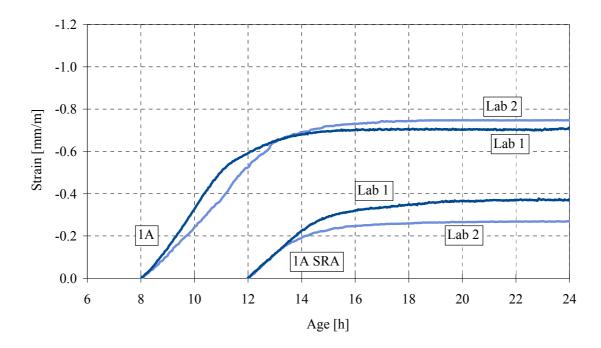


**Figure 4–11** Strain under sealed conditions measured by the shrinkage cone method (lab 2), onset of stress development as time-zero, two tests with two shrinkage cone apparatus, concrete 1A

It is noted that the method as well as the concrete compositions were formerly unknown to the staff of the second laboratory. Hence, the achieved results indicate that the measurement procedure is sufficiently simple to be adopted quickly and easily. The evaluation of the test data from the age of one hour is given in the Annex (**Figure 9–6**, **Figure 9–7**). Even for this longer period including almost all of the chemical shrinkage, curves were in good agreement.

# 4.4.2 Reproducibility

A comparison of the mean values obtained at the two laboratories for concretes 1A and 1A SRA is shown in **Figure 4–12**. The curves are in a reasonable agreement. The somewhat larger differences with concrete 1A SRA possibly were caused in part by an erroneous mixing procedure used in one case. To determine the reproducibility as per ISO 5725-2:2002, a round robin test with a larger number of participants would be required.



**Figure 4–12** Mean strain obtained with the shrinkage cone method under sealed conditions, measurements carried out two or three times in a row at two different laboratories, onset of stress development as time-zero

# 4.4.3 Shrinkage cone method vs. corrugated tube method

### 4.4.3.1 General remarks

To further validate the shrinkage cone method, parallel tests with the corrugated tube method were carried out. The corrugated tube method is a well established method for measuring the autogenous shrinkage of pastes and mortars under isothermal conditions (cf. 2.4). It has been standardized by ASTM [ASTM 2]. Preliminary tests were carried out to examine the precision of the corrugated tube method and the specific setup. The testing frames which were used in the own investigations accommodated three or four molds. They were placed horizontally on a table in a room with constant air temperature of 20 °C (Figure 4–13). Additional quasi-isothermal tests at other temperatures were carried out in a temperature chamber.

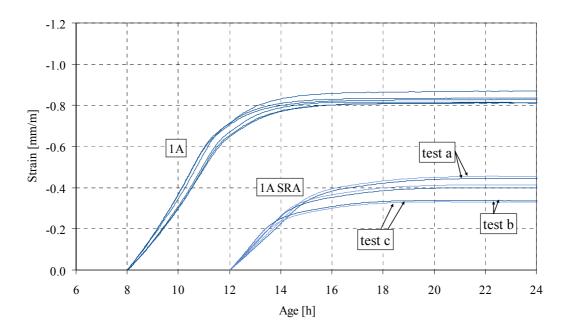
In the principal test series the tubes, having an inlet diameter of approximately 20 mm, were filled with the help of an applicator gun to facilitate the procedure. The concrete, in most cases being too sticky to be poured into the tube opening, was filled into the apposite cartridge and pressed into the tube through a long rubber hose (**Figure 4–13**). For this purpose the tubes were fixed vertically to a stand which was placed on a vibrating table during the process of filling. The plastic end plugs for closing the tubes were retrofitted with metallic caps providing a particularly smooth end surface and allowing for the fixation of one end of the specimen to the frame by means of a magnet (**Figure 4–13**). Digital feather touch spring push probes (Solartron DT/2/S) with extra-low tip force (0.18 N) were used to avoid compression of the tubes.



**Figure 4–13** Filling of the corrugated tube with applicator gun (left), dilatometer benches (top right), probe and metallic cap (mid right), opposite end fixed with a magnet (bottom right)

# 4.4.3.2 Repeatability precision of tests with corrugated tube method

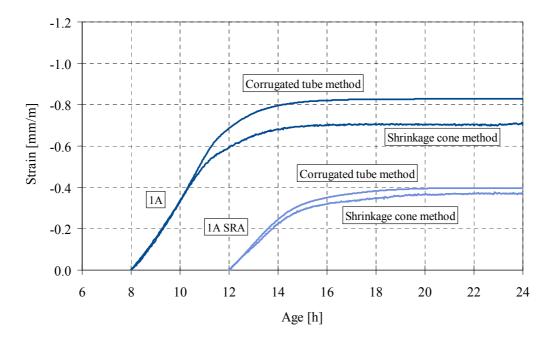
The autogenous shrinkage of concretes 1A and 1A SRA was tested with the corrugated tube method three times in a row with two specimens. With concrete 1A, the scatter among the six specimens was low. The standard deviation at the age of 24 h was 0.02 mm/m. In relation to the mean value of 0.83 mm/m this corresponded to a coefficient of variation of 2.7 %. The repeatability precision according to ISO 5725-2:2002 (assuming a 5 % error probability) was 0.056 mm/m, equivalent to 6.7 % with respect to the mean. The results for the three mixes of concrete 1A SRA showed more notable differences (**Figure 4–14**). Since the agreement was very good among the two specimens made from each mix, the overall scatter can be mainly attributed to different shrinkage behavior of the single mixes. Obviously the shrinkage behavior of different mixes of concrete 1A SRA generally varied more than that of concrete 1A (cf. **Figure 4–10** and **Figure 4–12**).



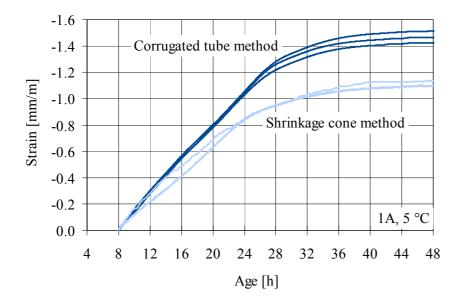
**Figure 4–14** Strain under sealed conditions measured by the corrugated tube method, onset of stress development as time-zero, three tests in a row for each concrete, two specimens per test

#### 4.4.3.3 Comparison of corrugated tube method and shrinkage cone method

The following two figures show comparisons of the results obtained with the shrinkage cone and the corrugated tube method for concretes 1A and 1A SRA. All results are based on own experiments which were carried out on concrete samples from identical mixes. The autogenous shrinkage strain at the age of 24 h was higher when measured with the corrugated tube method (Figure 4–15, 1A: +19 %, 1A SRA: +6 %). This tendency was even more pronounced at a constant temperature of 5 °C (Figure 4–20, +32 %).



**Figure 4–15** Shrinkage cone method vs. corrugated tube method: comparison of mean autogenous shrinkage strain for concretes 1A and 1A SRA, onset of stress development as time-zero



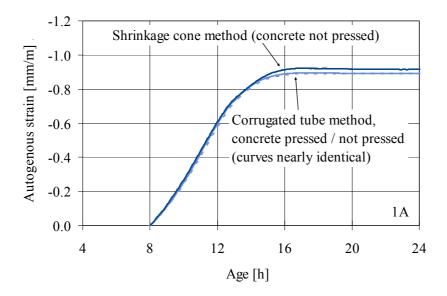
**Figure 4–16** Shrinkage strain (sealed conditions) from the age of 8 h, measured with 3 corrugated tubes (filled with applicator gun) and 2 shrinkage cones, concrete 1A, T = 5 °C

The low temperature of 5 °C reduced the efficiency of the superplasticizer and thereby led to much stiffer concrete. It was hypothesized that the use of the applicator gun may influence results. The pressure that is required to push the concrete into the tubes considerably depends on the workability of the mix and may alter the pore structure and the shrinkage. To further investigate this hypothesis, an additional test series was carried out. It comprised parallel tests with the shrinkage cone method, the corrugated tube method (with and without use of the applicator gun) and mercury intrusion porosimetry.

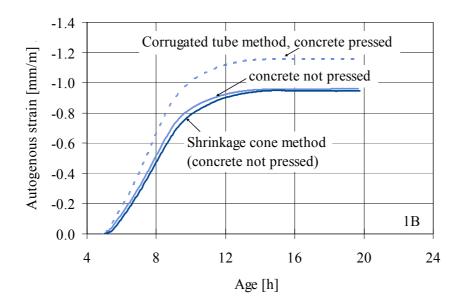
The concretes 1A and 1B were used for this test series. Due to the higher water demand of cement B, concrete 1B was considerably stiffer than concrete 1A. Therefore greater pressure needed to be exerted to fill the tubes by means of the applicator gun. Since it was not possible to pour the relatively sticky concretes into the corrugated tubes, a spoon was used with those tubes that needed to be filled without the help of the applicator gun. Two mercury intrusion tests were carried out at early age on samples obtained with and without using the applicator gun. All tests were carried out at 20 °C. The corrugated tubes were slightly vibrated throughout the filling process. The concrete in the shrinkage cone was compacted by carefully beating the filled jar on the table 10 times. The samples for the porosity tests were vibrated for approx. 20 s. Before starting mercury intrusion they were vacuum-dried for 24 h.

The resultant autogenous shrinkage for concrete 1A and 1B (mean of 3 specimens) is shown in **Figure 4–17** and **Figure 4–18**, respectively. The onset of stress in earlier restrained ring tests was taken as time-zero. With the sticky but fluid concrete 1A, the results did not depend on the test method. Furthermore, results obtained with the corrugated tube method did not depend on the use of the applicator gun. With the much stiffer concrete 1B the results obtained with the shrinkage cone method and the corrugated tube method without applicator gun again were almost identical. Contrary to that, strains measured for concrete 1B were significantly higher when the concrete was pressed into the tubes by means of the applicator gun.

Note that the deviation of absolute values for concrete 1A from results of earlier or later tests (cf. e.g. **Figure 4–20**) can have several reasons, for instance aging of materials, different temperatures and variation of mixes. The potential influence of a slightly deviating hydration was illustrated in **Figure 2–6**.



**Figure 4–17** Autogenous shrinkage measured with the shrinkage cone method and with the corrugated tube method with and without use of an applicator gun (pressed / not pressed), concrete 1A

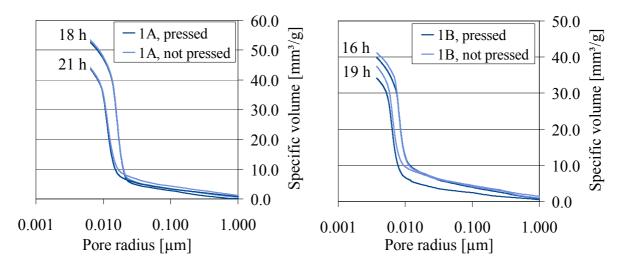


**Figure 4–18** Autogenous shrinkage measured with the shrinkage cone method and with the corrugated tube method with and without use of an applicator gun (pressed / not pressed), concrete 1B

The results of the porosity measurements are shown in Figure 4–19. Porosities at early age of pressed and not pressed concrete were almost identical with concrete 1A and notably differed with concrete 1B. Obviously concrete 1A was sufficiently fluid to be pressed into the tubes

without significant changes to the pore structure. Contrary to that, it appears that the pore structure of concrete 1B was significantly changed by the pressure. These observations, along with the different shrinkage behavior of pressed and not pressed concrete 1B, confirm the assumption that using an applicator gun can influence the results of shrinkage measurements.

In view of the excellent agreement that was achieved when there was no disturbing influence of the applicator gun, the discrepancies observed in the earlier test series (Figure 4–15, Figure 4–16) largely can be attributed to this influence. Moreover, the agreement indicates the validity and precision of both test methods.



**Figure 4–19** Cumulative pore volume tested at early age, concrete sampled with and without the use of an applicator gun (pressed / not pressed), concrete 1A (left) and 1B (right)

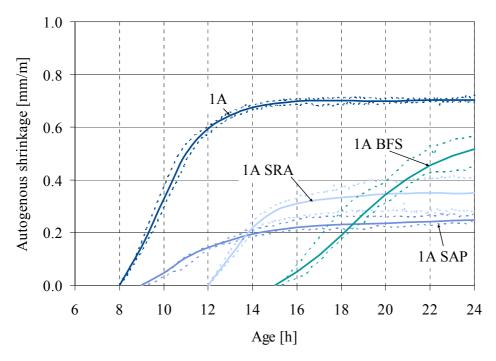
# 4.5 Autogenous shrinkage of the investigated concretes at 20 °C

The autogenous shrinkage of concretes 1A, 1A SRA, 1A SAP and 1A BFS, measured three to four times in a row with the shrinkage cone method under quasi-isothermal conditions (20 °C) is shown in Figure 4–20. Cement batch 4 (cf. Table 9–1) was used for these tests of the autogenous shrinkage as well as for the corresponding restrained ring tests and the tests of mechanical properties described in section 3.5. The onset of stress development in restrained ring tests was taken as time-zero (cf. Figure 6–8). The solid curves represent best-fit functions obtained from regression analysis of the mean values of the single tests. Dotted curves indicate the minimum and maximum values obtained in the single tests. The results of all tests are given in the Annex (Figure 9–8 to Figure 9–11). The best-fit functions were used for the further evaluation of the restrained ring tests (cf. 6.4.5.2 and 6.4.5.3).

Autogenous shrinkage developed most rapidly with concrete 1A which also showed the highest shrinkage throughout the entire test period of 24 h. Although no asymptotic end value was determined, it can be assumed that more than roughly 70 % of the autogenous shrinkage occurred between 8 h and 16 h. The shrinkage strain was 0.69 mm/m at 16 h and hardly increased anymore afterwards. The period from 8 h to 16 h can be further divided into two phases. Between 8 h and 11 h the rate of shrinkage was virtually constant at approximately 0.2 mm/m/h. In the second phase between 11 h and 16 h the rate decreased continuously and converged to nil.

The autogenous shrinkage of concrete 1A BFS started to develop from 15 h and proceeded thereafter at considerable rates. The strain at 24 h was 0.52 mm/m. Compared to the reference concrete 1A, the autogenous shrinkage of the concretes 1A SAP and 1A SRA was substantially reduced. The value at 24 h was approximately 0.25 mm/m and 0.35 mm/m, respectively, corresponding to a reduction of the autogenous shrinkage by 64 % and 50 % with respect to concrete 1A.

The scatter among the single results was very low with concrete 1A. The more notable scattering with the other concrete compositions may be attributable to larger variations among the single mixes or to a lower homogeneity of the mixes. A similar trend was observed with concrete 1A SRA in the investigation of the reproducibility of the test method, cf. 4.4.3.2.



**Figure 4–20** Autogenous shrinkage measured with the shrinkage cone method, concretes 1A, 1A SRA, 1A SAP, 1A BFS, dotted curves: minimum and maximum values of three to four consecutive tests

#### 4.6 Tests under non-isothermal conditions

To further examine the suitability of the shrinkage cone method for measuring the autogenous shrinkage also under non-isothermal conditions, three consecutive tests with concrete 1A were carried out during which the temperature of the coolant was increased between 18 h and 24 h. This period was chosen for this tentative test series since the autogenous shrinkage at that age proceeded at very low rates (at a concrete temperature of 20 °C). This facilitated the separation of thermal and shrinkage strains.

The concrete had a constant temperature of approximately 21.5 °C before the heating began. The jar was placed on two prisms of borosilicate glass having a very low coefficient of thermal expansion in order to minimize the influence of increasing temperature on the base of the stand (cf. **Figure 4–7**). The temperature of the concrete was measured in a second jar in order not to disturb the measurement of the autogenous shrinkage. The concrete temperature was increased constantly to approx. 33 °C at 24 h.

The measured strains were compensated for by the thermal strains according to the explanations given above (4.3). The coefficient of thermal expansion of the concrete was assumed to be constant at 11.3  $\mu$ m/m/K (cf. **Figure 3–12**). The relevant height of the bottom part of the jar and the jar's coefficient of thermal expansion were assumed to be 10 mm and 23  $\mu$ m/m/K, respectively.

Results are shown in **Figure 4–21**. The measured strain (mean values) decreased with increasing temperature since the thermal expansion of the jar and of the specimen reduced the distance to the laser unit. The shrinkage strain was calculated by adding the thermal strains of jar and specimen to the measured strains. Between 19 and 24 h the autogenous shrinkage increased by approx.  $80 \mu m/m$ , whereas the autogenous shrinkage at a constant temperature of  $20 \,^{\circ}\text{C}$  did not significantly increase after the age of 16 h anymore. Thus, the result would signify that the rate of autogenous shrinkage was considerably increased by higher temperatures. This can be considered plausible, however, further tests are required to validate the method.

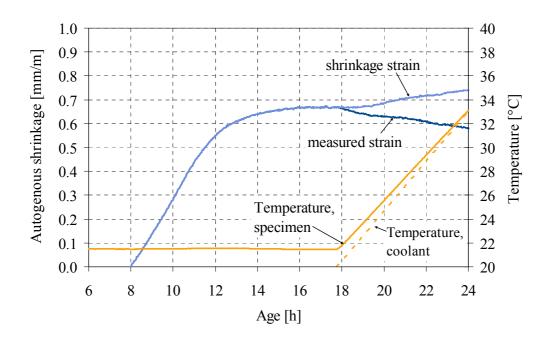


Figure 4–21 Autogenous shrinkage of concrete 1A under non-isothermal conditions

#### 4.7 Summary

The shrinkage cone method for measuring the autogenous shrinkage, developed on the basis of an existing testing apparatus, was examined as to its suitability, reliability and precision. It allows for a precise and efficient measurement of the autogenous shrinkage of cement pastes and mortars containing grains up to 2 mm. A number of tests on the method's reproducibility and repeatability were carried out sufficient to consider the method validated, validation being limited though to isothermal conditions. The test method allows tests under non-isothermal conditions, too, making it a potential reference method for such tests. In addition, it is comparably easy to use and suitable also in the case of sticky and stiff mortars.

For the measurement of the autogenous shrinkage of the investigated concretes, the onset of stress development in restrained ring tests was used as time-zero. The reference concrete 1A

yielded the highest autogenous shrinkage until the age of 24 h with approximately 0.69 mm/m. This value was already achieved at about 16 h. The autogenous shrinkage was considerably reduced by the shrinkage-reducing admixture and the superabsorbent polymers. The reduction amounted to 50 % (1A SRA) and 64 % (1A SAP) with respect to the 24 h-value of concrete 1A. In both cases the autogenous shrinkage like with concrete 1A started to converge to an end value. Contrary to that, with concrete 1A BFS the autogenous shrinkage continued to increase at considerable rates from the onset at 15 h to 24 h and reached approximately 0.52 m/m at 24 h.

#### 5 Stress and cracks due to restrained autogenous shrinkage

#### 5.1 Introduction

In this chapter the most relevant aspects of restraint stress and cracking due to restraint stresses are described, serving as a general introduction into the subject and the subsequent investigation of the autogenous shrinkage cracking propensity (cf. 6). The first parameter dealt with is the degree of restraint. Its calculation is explained in detail; the given formulas are used later for finding an analytical solution for ring tests under non-isothermal conditions (cf. 6.3.2.3). The next point that is addressed is the formation of cracks; some related aspects like durability, self-healing and the effect of fibers are outlined as well. Then the strong influence of viscoelastic deformations on restraint stress at early age and especially at very early age is discussed; a definition of the very early age is suggested. Also, the difficulties that result from the importance of the very early age in terms of creep investigations are analyzed; it is shown that there is hardly any relevant data for the influence of creep on early age cracking. The stress-strength criterion for quantifying the autogenous shrinkage cracking propensity is described, including its capacities and limitations. The role of temperature is highlighted, being one of the fundamental difficulties in the investigation of restraint stress and related cracking. Finally, relevant results on restraint stress and stress-strength ratios obtained by other researchers are depicted.

### 5.2 Degree of restraint

Autogenous shrinkage can lead to significant stresses in hardening concrete if deformations are restrained. Restraint in concrete structures is a common issue and often unavoidable. It can be internal or external, according to whether it originates from the concrete itself or from its surroundings [Ben 04]. In the case of autogenous shrinkage, internal restraint can mainly arise from enclosed aggregates [Iga 02] and reinforcement, external restraint from adjacent structures and from formwork. It was hypothesized that anhydrous cement grains significantly contribute to internal restraint [Lou 99], [Gua 06]. Volume changes due to autogenous shrinkage are largely uniform and therefore do not generate Eigenstresses. Eigenstresses are caused by local gradients of moisture (e.g. due to drying shrinkage) or temperature (e.g. in mass concrete).

Restraint is either partial or complete. The deformations of the lower concrete layers of a dam during cooling can be considered almost completely restrained by the bedrock [Spr 95b]. However, except for in such special applications restraint is virtually complete only around initial setting and notably decreases with increasing modulus of elasticity of the concrete. The variable degree to which deformations in practice are restrained can be accounted for by the so called degree of restraint. In a general form it can be expressed by the ratio of stiffness of two ideal elastic bodies restraining each other. The stiffness is the ratio of an external force, acting in arbitrary direction on an elastic body of arbitrary geometry, to the resulting non-rigid displacement. (Displacements are called rigid if a body is shifted as a whole, and non-rigid if they occur within the body itself.)

The most simple case is the mutual restraint of two ideal elastic springs 1 and 2 with respective spring constants  $k_1$  and  $k_2$  and lengths  $l_1$  and  $l_2$  (Figure 5–1). For any force F, the ratio of deflection of both springs (length change  $u_1$  and  $u_2$ ) is constant and equal to the inverse ratio of the spring constants. (In the following, absolute values are considered to avoid signs which

would be required with vectors, hence  $|u_1|$  and  $|u_2|$ .) The degree to which deformations of spring 1 are restrained by spring 2 goes to infinity for  $|u_2|$  converging to 0. The springs restrain each other to the same degree if their deflection is identical ( $|u_1| = |u_2|$ ,  $\psi = I$ ).

$$|u_{I}| = l_{I} - l_{I,F}$$

$$|u_{2}| = l_{2} - l_{2,F}$$

$$|u_{2}| = l_{2} - l_{2,F}$$

$$|u_{2}| = k_{2} / k_{I} = |u_{I}| / |u_{2}|$$

$$|u_{1}| = l_{I} - l_{I,F}$$

$$|u_{2}| = k_{2} / k_{I} = |u_{I}| / |u_{2}|$$

$$|u_{2}| = k_{2} / k_{I} = |u_{I}| / |u_{2}|$$

$$|u_{1}| = l_{I} - l_{I,F}$$

$$|u_{2}| = k_{I} / k_{I} = |u_{I}| / |u_{2}|$$

$$|u_{2}| = k_{I} / k_{I} = |u_{I}| / |u_{I}|$$

$$|u_{2}| = k_{I} / k_{I} = |u_{I}| / |u_{I}|$$

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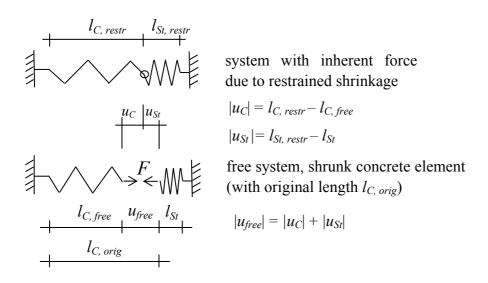
$$|u_{I}| = k_{I} / k_{I} = |u_{I}| / |u_{I}|$$

$$|u_{I}| = k_{I} / k_{I} = |u_{I}| / |u_{I}|$$

$$|u_{I}| = k_{I} / k_{I} = |u_{I}| / |u_{I}|$$

**Figure 5–1** The degree of restraint  $\psi$  as ratio of the stiffness k of two ideal springs

If restrained concrete structures are concerned, the degree of restraint is commonly expressed as ratio of prevented to free deformations. May spring 1 be a shrinking concrete element whose length  $l_{C,orig}$  would shrink by  $|u_{free}|$  if there was no restraint (**Figure 5–2**). Then the degree of restraint  $\psi$  exerted on the shrinking concrete element by an adjacent steel element (spring 2 with original length  $l_{Sl}$ ), given as value between 0 and 1 ( $\psi \le 1$ ), results from Equation 5–1.



**Figure 5–2** The degree of restraint  $\psi$  as ratio of prevented to free shrinkage  $(0 \le \psi \le 1)$ 

$$\psi_{St \to C} = \frac{|u_C|}{|u_C| + |u_{St}|} = \frac{|u_C|}{|u_{free}|} = \frac{|u_{free}| - |u_{St}|}{|u_{free}|} = 1 - \frac{|u_{St}|}{|u_{free}|} \text{ with } 0 \le \psi \le 1 \quad \text{Equation } 5-1$$

Depending on the direction of force and displacement, the stiffness is called either extensional, flexural or rotational. The extensional stiffness is given as the ratio of a force F and a body's resultant non-rigid displacement u in the direction of force. It is noted that the displacement u can be rewritten as the product of the unit elongation  $\varepsilon$  (linear strain) and the initial length l. Furthermore, the linear strain is the ratio of the stress  $\sigma$  to the modulus of elasticity E (Hooke's Law), and stress is the ratio of the force (load) to the loaded area A. Given this, the axial extensional stiffness  $k_{\rm ext}$  of an elastic and laterally unrestrained prism can be written as the product of the modulus of elasticity in the direction of force and the cross-sectional area perpendicular to it, divided by the length (Equation 5–2).

$$k_{ext} = \frac{F}{u} = \frac{E \cdot A_{\perp}}{l}$$
 Equation 5–2

Equation 5–2 illustrates that stiffness is not the property of a material but of a three-dimensional body. The extensional stiffness goes to infinity if the length in the direction of force goes to zero. At given length the stiffness increases proportionally with increasing modulus of elasticity. As the modulus of elasticity of concrete increases with time, the stiffness of a concrete body increases as well. The stiffness can be calculated for a given point in time if the modulus of elasticity is known. If the stiffness of the restraining counterpart is constant, the degree of restraint increases proportionally with the stiffness of the concrete element.

At given mechanical properties and shrinkage strains, the stress in a concrete element due to restrained shrinkage increases with increasing degree of restraint. If concrete was a purely elastic material, the stresses in a shrinking concrete element within a period of constant modulus of elasticity would be a linear function of the degree of restraint and the free shrinkage strain  $\varepsilon_{SH}$ , and in case of either no shrinkage or no restraint the stress would be nil. The stress after arbitrary time intervals i with variable mean modulus of elasticity E and degree of restraint  $\psi$  could be calculated by means of Equation 5–3.

$$\sigma(t) = \sum_{i} E(t_i) \cdot \varepsilon_{SH}(t_i) \cdot \psi(t_i)$$
with  $0 \le \psi \le 1$ 
Equation 5–3

However, concrete is a viscoelastic material. Under load it undergoes time-dependent deformations which are partly elastic and reversible and partly plastic and irreversible. These time-dependent deformations may considerably influence the stress, especially at earlier ages when the viscoelastic behavior is more pronounced (5.4).

#### **5.3** Formation of cracks

A crack is a local failure due to stresses. Depending on whether the non-elastic deformations prior to failure are large or small one speaks of ductile or brittle fracture. The failure of plain, hardened cementitious materials is brittle. Concrete is an aging material whose brittleness increases with time. The pronounced brittleness of concrete, in conjunction with its relatively high modulus of elasticity, is disadvantageous because the strain capacity in many cases is too small to prevent cracking. Reinforcement with ductile materials, for instance steel, reduces the brittleness.

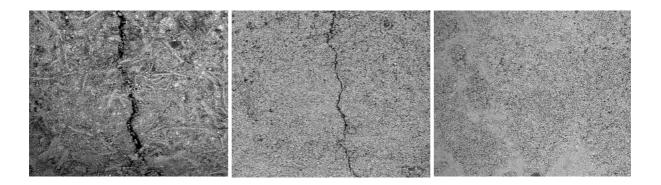
Stresses due to restrained shrinkage or thermal deformations can lead to cracks in concrete. Cracking in many cases occurs at early age [Ben 08c]. Early age cracks due to restrained autogenous shrinkage were observed as well [Taz 95], [Iga 02], [Gra 04]. Cracks can be the consequence of restraint stress caused by superimposing thermal and shrinkage strains. In practice, multiaxial stress states prevail, whereas experimental investigations of the cracking risk typically regard only one dimension. The rate of stress increase and the duration of stresses can influence the formation of cracks in concrete.

Numerous comprehensive investigations of early-age cracking were carried out because cracks may have unfavorable effects on the durability of concrete and considerably reduce the service life time of structures [Spr 95], [Mih 00], [Kov 01], [Ben 03a]. Reinforcement and concrete may gradually corrode due to a number of deleterious reactions caused by various agents possibly penetrating a structure through cracks, for instance chlorides [Sta 01]. Furthermore, cracks can significantly impair the serviceability of certain structures like tanks and containments. Cracks can also be a serious aesthetic issue, particularly with architectural concrete.

Cracks in concrete can be micro- or macro-cracks, however, there is no univocal definition of these terms. Micro-cracks commonly are smaller than approximately 20 to 50 µm and invisible to the eye, either because of their size or their being located inside the concrete [Len 98]. Depending on the circumstances, micro-cracks may prevent or facilitate later macro-cracks. Macro-cracks may become through-cracks. Depth and width of cracks are important parameters for the occurrence and progress of potential corrosion processes. However, the surface width of cracks may not be an adequate indicator for the potential of corrosion of reinforcing bars located some tens of millimeters inside the concrete [Tam 09].

Under specific conditions, basically in the presence of moisture, the size of cracks in concrete may decrease with time due to the formation of stable reaction products on the crack surface, mainly calcium carbonate crystals [Edv 99], [Lie 08]. This so called self-healing may also be used for active crack control [Ahn 10]. In own investigations with ultra-high strength concrete a crack with a width of 15 to 20 µm was completely closed after a period of 28 d at 95 % relative humidity (Figure 5–3). Permeability in the area of the closed crack and in the surrounding area of uncracked concrete can be assumed to be similar. It is unclear to which extent the mechanical properties were restored as well. A larger crack with a width of approximately 300 µm became significantly smaller but was not closed after 35 d in 95 % relative humidity (cf. [Mue 10a]).

The widening of micro- to macro-cracks that seriously affect the durability may be prevented by appropriately spaced reinforcement bars or by fibers [Pas 09]. Especially the size of cracks due to foreseeable loads can be minimized efficiently by reinforcement [Seo 08], [Lie 08]. In comparison, magnitude and direction of crack-inducing stresses due to restrained deformations may be more difficult to predict. Two or more stress-inducing mechanisms may be active and the sources of restraint may be difficult to anticipate. External restraint can lead to considerable stress concentrations. Randomly distributed fibers, either alone or in combination with steel bars, may be more efficient in preventing cracks than steel bars alone.



**Figure 5–3** Electron-microscopical image of a crack in ultra-high strength concrete with an original width of approx. 15 to 20 μm (left), after 7 days in 95 % RH (centre) and after 28 days in 95 % RH (right); concrete 1A (cf. Table 3–1), cf. [Mue 10a]

Although ultra-high strength concrete in most applications is reinforced with fibers, the assessment of the cracking risk for a number of reasons needs to be based on investigations with unreinforced concrete. Crack-inducing stresses due to restrained autogenous shrinkage can occur shortly after the solidification of ultra-high strength concrete. The influence of stress relaxation during that period is particularly strong and not fully understood (5.4, 5.5). Fiber reinforcement leads to a complex composite system that is even more difficult to investigate than plain concrete.

The behavior of the composite system is influenced by the bond between concrete and reinforcement which in turn depends on the mechanical properties of the concrete [Mue 10c]. These properties change rapidly at very early age. Furthermore, typical amounts of fibers (approx. 1 to 6 Vol.-%) do not completely prevent the formation of micro-cracks. Whether fibers are able to transfer tensile stresses across micro-cracks in a way that prevents the appearance of macro-cracks depends on numerous parameters, for example the type, the amount, the distribution and the orientation of fibers [Mec 05]. Finally, stress concentrations may occur in areas with less or unfavorably oriented fibers.

These were the main reasons not to consider fibers in this study. Nevertheless the reinforcement of ultra-high strength concretes with fibers in practice will remain the principal solution for optimizing its performance, and substantial progress was made in that field of research in the past decade [For 05], [Uch 05], [Hab 06].

The primary way to avoid cracks due to restraint stress in unreinforced concrete would be to avoid deformations. Although autogenous shrinkage is caused by a lack of water, external curing with water in most cases can hardly mitigate the internal desiccation since relevant types of concrete have few capillary pores which would take up the water in time [Ben 04] (cf. 2.1). This gives particular importance to modifications of concrete compositions that are suited to reduce the autogenous shrinkage without compromising desired properties. Three principal measures have been suggested: the use of supplementary cementitious materials, the addition of chemical admixtures and internal water-curing (cf. 2.5.3). All three measures were adopted in this study (cf. 3.2).

# 5.4 Very early age and importance of stress relaxation

Concrete is generally considered a viscoelastic material. Accordingly, the load-dependent deformations are the sum of immediate elastic and time-dependent viscous (or 'viscoelastic') deformations [VDZ 08]. Deformations due to viscous flow are partly elastic (reversible) and partly plastic (irreversible). The hardening or hardened cement paste, in particular the water contained in it, is constitutive for the viscous deformation component. Due to the increasing formation of micro-cracks, the fraction of plastic deformations in the viscous deformations increases at stress levels beyond approximately 40 % [Mue 09], [Atr 03].

Phenomenologically, the time-dependent deformations strongly depend on the age at loading and the stress-strength ratio. Viscous deformations are larger at earlier loading ages and at higher stress levels. The viscoelastic behavior of concrete commonly is investigated with creep and relaxation tests, carried out at different stress levels and for different loading ages of at least 1 d, furthermore at constant room temperature. In these tests, creep is time-dependent deformation due to constant stress, and relaxation is time-dependent decrease of stress due to constant deformation. Load-independent deformations within the test period need to be measured and accounted for as well.

Results are used for setting up so called compliance functions which describe the long-term deformations of concrete under load. Apart from the measured deformations, empirical models for predicting creep typically consider the concrete's modulus of elasticity and compressive strength at 28 d, the geometry of the concrete member, the relative humidity of the ambient environment as well as factors that may strongly influence the hydration progress, in first place the type of cement and the temperature [Baz 01], [Mue 09], [Atr 03], [VDZ 08], [fib 10].

A large part of the autogenous shrinkage of ultra-high strength concrete takes place within the first 24 h (cf. 2.5 and 4.5). High stress levels can occur few hours after solidification if deformations are restrained. In that case the viscous deformations are very large and strongly reduce the stress that would arise at purely elastic behavior, the so called 'theoretical-elastic' stress [Epp 08c].

In tests with three different ultra-high strength concretes made from ordinary Portland cement the actual restraint stress at the age of one day was approximately 10 % of the theoretical-elastic stress [Sch 07]. Hence, 90 % of the stress that would have occurred under restraint conditions if the concrete behaved completely elastically, were eliminated by relaxation. In addition, the modulus of elasticity between 12 h and 3 d was tested; a compressive stress of one third of the compressive strength was applied for 10 s. The time-dependent (viscoelastic) deformations after that short period were divided by the instantaneous elastic deformations. The ratio exponentially increased with decreasing age (Figure 5-4).

Hence, the influence of creep and relaxation on restraint stresses shortly after solidification can be very strong. In comparison, restraint stresses due to negative thermal strains develop not until the concrete temperature starts to decrease, hence, several or many hours after solidification (cf. e.g. [Bre 89]). To adequately specify this difference, it is helpful to differentiate between 'early age', a well-established term for the first days of hardening, and 'very early

age'. Actually, the latter term is already in use [Bou 08b], [Ham 01], [Ong 06], [Rad 98], [Ste 09]. However, a threshold that separates early and very early age was not agreed upon, yet.

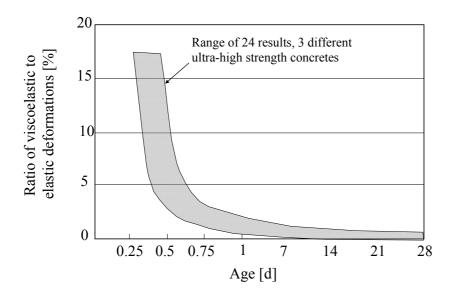
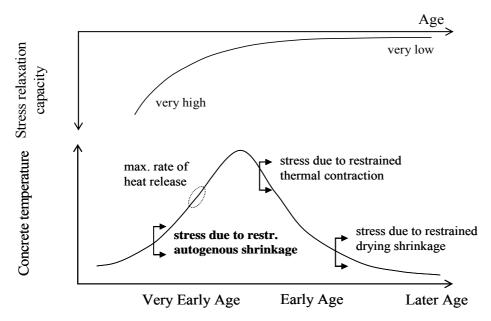


Figure 5–4 Ratio of time-dependent viscoelastic deformations after 10 s of stress of 1/3 of the compressive strength to the immediate elastic deformations (from [Sch 07])

The definition that is suggested for the term 'very early age' is illustrated in Figure 5–5. The figure comprises the principal types of tensile restraint stress. Their typical onset is related to the qualitative course of temperature in a small or medium-sized concrete element, stored at constant temperature. The temperature peak is due to heat of hydration. Stresses due to restrained autogenous shrinkage can occur from solidification and prior to the maximum rate of heat release, stresses due to thermal contraction only after the concrete has started to cool down. Stresses due to drying shrinkage typically begin even later.

The stress relaxation capacity strongly decreases with time [Atr 03]. It is very high at very early age, as was shown above. Stress relaxation still can have a very notable influence on stresses due to negative thermal strains, yet, reasonable results in some cases were achieved without its consideration [Spr 95b], [Thi 95]. Elastic relations may be sufficient to predict stresses due to restrained drying shrinkage, however, creep and relaxation are still noticeable [Bis 07].

It is suggested to use the term 'very early age' for the period from solidification to the moment the concrete temperature starts to decrease. A superposition of tensile stresses due to autogenous shrinkage and thermal contraction according to this definition can occur only after the end of the very early age, i.e. at early age. Increasing temperatures at very early age would have the usually beneficial effect to counteract the negative strains due to shrinkage. Notably, this does not predetermine the worst case in terms of cracking. Temperature changes do not only cause thermal strains, they also influence the autogenous shrinkage as well as the mechanical properties. Cracking at very early age is a consequence of the interaction of these parameters.



**Figure 5–5** Very early age: period from solidification (first appearance of stress due to restrained autogenous shrinkage) to maximum temperature, characterized by very high stress relaxation capacity

The fact that stresses and cracking due to restrained autogenous shrinkage at very early age are strongly influenced by the high stress relaxation capacity make the experimental investigation and the mathematical description of these phenomena more difficult than they are at early age. However, no complete assessment of the cracking risk of ultra-high strength concrete is feasible without a quantification of this influence.

### 5.5 Creep and cracking - further methodological aspects

The essential task of most investigations of creep of concrete is to predict the long-term deformations that occur in structures due to dead loads or post-tensioning forces. The thorough understanding of time-dependent deformations is vital. An unfortunate example is the Koror-Babeldaob Bridge in Palau. At the time of its erection in 1977 it was the world's largest prestressed concrete box girder bridge. It collapsed in 1996, basically due to wrong estimations of tensile creep; two people were killed [Baz 08], [Kri 08].

The experimental investigation and mathematical description of creep and relaxation of concrete is largely based on the assumption of linear viscoelasticity. According to this concept the time-dependent deformations are proportional to the imposed stresses. This assumption is appropriate for stress levels below 40 to 50 %, i.e. in the range of service loads. For this reason most experiments are carried out at stress-strength ratios below 50 to 60 % [Mue 09].

Another assumption that is usually made in empirical creep models is the superposition principle. Accordingly, creep due to variable stresses is the sum of the creep deformations at each stress increment, i.e. a potential influence of the stress history on creep is neglected. This allows for a relatively straightforward consideration of the aging of concrete, i.e. the development of mechanical properties that changes the stress level after loading. Compared to the rapid development of mechanical properties of ultra-high strength concrete at very early age,

the influence of aging on creep in most cases is very moderate. The superposition principle is considered valid for stress levels below approximately 40 to 50 % [Mue 09].

Both assumptions, the superposition principle and the linearity of viscoelastic deformations, are no longer valid for stress levels higher than 40 to 50 %. The time-dependent deformations increase disproportionally with increasing stress level (**Figure 5–6**). The so called proportionality limit at early and very early age possibly is even lower than 40 % [Cho 10], [Bur 08b], [Kam 09].

Notably, investigations of the risk of cracking have to cover the whole range of stress-strength ratios up to failure. As a consequence, the consideration of the influence of creep and relaxation on cracking cannot be based on the aforementioned principles. The non-linearity of creep has to be accounted for. One solution can be to increase the linear creep coefficient by a stress level-dependent factor [Baz 89a], [Baz 89b], [Baz 91], [Baz 79], [Maz 03]. However, such solutions are relatively difficult to find and were never applied to the very early age.

In the case of ultra-high strength concrete, the analysis is further complicated by the rapid development of mechanical properties and the associated change of the stress-strength ratio. It is unclear whether some modified superposition principle is applicable. Moreover, creep occurs due to tensile stresses, whereas much more experiences exist with compressive creep. Although mechanisms possibly are the same, compressive and tensile creep may significantly differ [Alt 02], [Atr 03], [Gar 09].

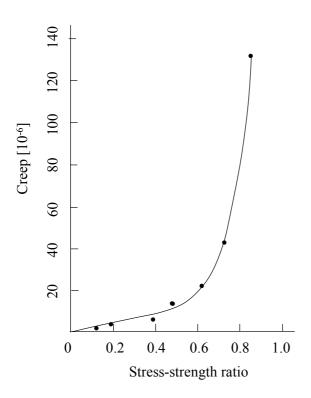


Figure 5–6 Relation between compressive creep after one minute load and the stress-strength ratio, concrete age: 5 d [Jon 36], from [Atr 03]

Very few investigations of tensile creep were carried out that considered stress levels beyond the proportionality limit. No such investigation is known that would have included the very early age. Yet, there are some investigations that addressed in some way one of the elements of the problem, either tensile creep, ultra-high performance concrete, the very early age or very high stress levels.

Creep under compression at very early age was analyzed in [Cho 10]. The stress level was 20 %. The focus was on separating quasi-instantaneous and time-dependent deformations. It was found that the solidification theory ([Baz 89a], [Baz 89b]) underestimated creep and that the rate of stress increase had to be accounted for to satisfactorily model creep at very early age.

Basic creep of C-S-H phases from the point of solidification was treated in [Ste 09]. It was tried to model the creep compliance with the 'effective Young's modulus' concept. The effects of aging could not be satisfactorily embodied.

A comprehensive study of tensile and compressive creep at early age was presented by Atrushi [Atr 03]. The experiments with high performance concrete considered stress levels up to 80 % and ages at loading of 1 to 8 d. The term 'sealed creep' was introduced to account for the unavoidable drying creep component due to autogenous shrinkage. Creep was higher under tension than under compression. The proportionality limit for tensile creep was found at a stress level of approximately 60 % (age at loading: 1 d). Model parameters obtained from compressive creep tests led to a better agreement between calculated and measured tensile creep than the parameters obtained from tensile creep tests. As a possible explanation the author mentioned the uncertainties in separating tensile creep and autogenous shrinkage. Additional tests of stress relaxation at different temperature levels led to inconsistent results.

Very high stress levels were considered in a study on tensile creep of high strength concrete [Rei 06]. Specimens were loaded at the age of 28 d. Failure occurred after seconds or few minutes at stress-strength ratios of 90 and 95 %. Specimens did not fail at a stress level of 80 % maintained for two years. It was found that loaded concrete shrank and hydrated more than non-loaded; similar results previously had been found by Kovler [Kov 99].

Recently Garas et al. presented a study on the tensile creep behavior of ultra-high strength concrete incorporating steel fibers [Gar 09]. Specimens were cured at different temperatures and loaded at 7 d to stress-strength ratios of 40 or 60 %. Thermal treatment at 90 °C for 48 h and the incorporation of steel fibers were found to reduce the tensile creep after 7 d of loading.

A model for the early-age stress development that considered tensile creep was proposed [Pan 08b]. The model was based on a comprehensive investigation of splitting tensile strength, modulus of elasticity, autogenous shrinkage, coefficient of thermal expansion and heat of hydration [Pan 08a]. Good agreement between model and measurements were found, however, experiments were limited to a stress level of 30 %.

Kamen et al. investigated the early age viscoelastic behavior of fibre reinforced ultra-high strength concrete under tension [Kam 09]. Specimens were loaded at the age of 35, 46 and 72 h, the stress level ranged from 30 to 63 %. The proportionality limit was found at a stress

level of 35 %. The undertaken numerical simulation based on Maxwell chains led to a reasonable agreement with test results, however, only within the proportionality limit.

A comprehensive study of creep of ultra-high strength concrete was carried out by Burkhart and Müller [Bur 08a], [Bur 08b]. Specimens were loaded at 1, 3 and 28 d and stress levels used were 30 and 60 %. Creep was much higher at earlier age at loading. The final scope of the study is a thermodynamically sound constitutive nonlinear model.

More investigations and models could be added, for instance [Pig 00], [Mar 02], [Tao 06], [Koe 08], [Sch 09], [Swi 09], [Swi 10] as well as the valuable summary of early age creep compiled by Lange and Altoubat [Lan 03]. However, the essence would not change: there is no investigation that focused on tensile creep at early or very early age with stress levels close to cracking, and none of the existing models can provide a precise solution of this problem.

Experimentally, the combination of very high stress levels, tensile forces and a start of tests from shortly after the solidification represents an enormous challenge. Such investigations in principle can be carried out with temperature-stress testing machines or similar uniaxial stress rigs (cf. 6). However, few institutions dispose of such a machine and very few of more than one. At high stress levels, specimens may easily fail prematurely due to imperfections, stress concentrations and overestimation of the tensile strength. As a consequence, a large number of tests is required. Insofar the considerable costs for tests with temperature-stress testing machines represent a significant disadvantage. Technical shortcomings will be addressed in chapter 6.

Conventional creep tests require a constant stress. Thus, such tests are designed in accordance with the requirements of structural engineers to predict the deformations for a period of 50 to 100 years. The loads in most cases are constant and well known in advance.

In the case of restrained autogenous shrinkage of ultra-high strength concrete the situation is completely different. The load is a complex function of autogenous shrinkage, modulus of elasticity and relaxation. The autogenous shrinkage depends on the concrete composition and in a hardly understood way on temperature, and the modulus of elasticity changes with time and temperature. Furthermore, cracking is not a matter of years but of hours.

The considerable experimental and mathematical difficulties described above as well as the fundamental differences between long-term creep due to service loads and relaxation-dependent cracking give good reason to question whether the conventional approach is appropriate for the investigation and description of the influence of creep and relaxation on stresses and cracking due to restrained autogenous shrinkage. Possibly a different approach would be more efficient.

In the passive restrained ring test, strains and stresses cannot be controlled and therefore not kept constant. However, there are also remarkable advantages. The test is relatively simple and economic, hence a large number of tests is feasible and premature failure is no issue. Due to the geometry no coupling elements are required and stress concentrations are negligible, so the test can capture the entire stress development due to restrained shrinkage without any problems, including the very early age.

The influence of time-dependent deformations on stresses in restrained ring tests can be quantified by relating the actual stress to the theoretical-elastic stress (cf. 6.3.2.2). The suitability of this approach for the very early age was not analyzed, yet. This is done below (6.4.5.3 and 6.4.11).

# 5.6 Autogenous shrinkage cracking propensity

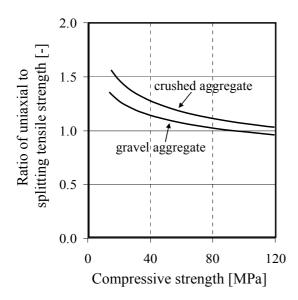
The principal goal of investigations of restraint stresses in concrete is to quantify the risk of cracking. This risk commonly is investigated one-dimensionally and specified as stress level, i.e. as ratio of uniaxial tensile stress to uniaxial tensile strength [Bre 00], [Miz 00], [Ros 00], [Sch 07], [Hay 08b]. Cracking is expected to occur at latest at a ratio of 1. The stress-strength criterion is to indicate the risk of crack initiation [Kan 08]; a precise description of the crack growth requires a fracture mechanical approach [Mue 02], [Mec 00], [Bra 88]. Alternatively, a strain criterion can be applied to assess the risk of crack initiation. Accordingly, cracking would be supposed to occur if the ultimate tensile strain is reached. The stress-strength criterion was considered more reliable than the strain criterion [Ham 07].

A general problem is the precise quantification of the resistance to failure. Uniaxial tensile strength tests are difficult to carry out, in particular at very early age. Some kind of mechanical coupling between specimen and testing machine is required. Usually steel parts are used which either clamp the specimen, are embedded in the concrete or glued to the specimen's surface [Kas 75], [Ham 07]. The mechanical coupling of specimens during or shortly after solidification is particularly demanding. Also, results of uniaxial tensile strength tests are influenced by test geometry and setup.

To determine the resistance to failure, splitting tension tests often are preferred to uniaxial tests since they are easier to perform. Results can be converted to the uniaxial tensile strength by empirically determined coefficients. However, results of splitting tension tests depend on a number of parameters, for instance the type, size and age of the specimen; widely differing coefficients have been suggested [Roc 99], [Mal 07], [fib 08]. Olesen et al. pointed out the non-elastic failure at early age [Ole 06]. Malarics et al. showed that no single formula is suitable for the conversion of splitting into uniaxial tensile strength values [Mal 10]; for cylindrical specimens with diameter and width of 150 mm and 300 mm, respectively, it was found that the ratio of uniaxial to splitting tensile strength considerably increased with decreasing concrete strength (Figure 5–7). A similar tendency was observed in [Sch 07].

The stress-strength criterion is based on several simplifying assumptions. First, strength is assumed to develop equally under restrained and under free conditions, i.e. independent of the stress history. Yet, it is known that compressive forces can increase the strength when exerted on hardening concrete [Ric 95]. Tensile forces can have an opposite effect [Tao 06], [Hay 08b]. They may promote micro-cracking [Alt 02] and, at very high stress levels, alter hydration through changes in the state of water [Rei 06].

Second, it is assumed that the influence of different loading rates on failure is negligible. The loading rate in static short-time tensile strength tests exceeds that in tests under restraint conditions many times over. Plastic deformations, more pronounced at earlier age, possibly alter the failure mechanism under restraint [IPACS].



**Figure 5–7** Ratio of uniaxial to splitting tensile strength as function of concrete compressive strength for two concretes with crushed and gravel aggregate, respectively (from [Mal 10])

Third, the stress-strength ratio usually is determined in a limited range. Two concretes with an identical maximum stress-strength ratio of e.g. 0.5 may have a different stress relaxation capacity beyond that stress level. Hence, in practical applications with higher stress levels they may turn out to have a different propensity to form cracks.

In conclusion, the assessment of the risk of cracking due to autogenous shrinkage by means of the stress-strength failure criterion presently is an approximate solution. In particular, the relation may not be linear. In other words, cracking is surely more probable at a ratio of 0.8 than at 0.4 but not necessarily twice as probable.

In view of these uncertainties and the scarce knowledge about the influence of creep and temperature, the term 'cracking risk', often associated with a probability of cracking, in the case of restrained autogenous shrinkage appears less suitable. To avoid a statistical connotation and to indicate the remaining uncertainties, here the term 'autogenous shrinkage cracking propensity' is used instead.

### 5.7 Role of temperature history

The autogenous shrinkage as well as the mechanical properties of concrete at early and very early age are considerably influenced by temperature (2.1, 2.5.2). This also appears to hold true for the autogenous shrinkage cracking propensity [Mar 08]. Few investigations of the autogenous shrinkage cracking propensity of ultra-high strength concrete were carried out; insofar it is not surprising that even less investigations considered the influence of temperature and that there is hardly any relevant data. The methodological challenges are considerable. Nevertheless, tests at a constant temperature of for instance 20 °C do not represent realistic on-site conditions and are, at least for the time being, only a provisional basis for the assessment of the cracking propensity of different concretes.

Experimentally, the major problem is that temperature changes influence strains and stresses in two ways: they alter the autogenous shrinkage and they lead to thermal strains. Since no general relation between temperature regime and autogenous shrinkage was found yet, the coefficient of thermal expansion of the concrete is required to individually account for the effects of thermal and shrinkage deformations. Notably, the coefficient of thermal expansion varies with the concrete, the concrete age and the temperature regime. It is very difficult and laborious to precisely measure its development (2.3.3).

From an engineering point of view, two scenarios need to be distinguished: temperature effects due to heat of hydration and those due to changes of the ambient temperature. The latter involves arbitrary curves of concrete temperature. Stresses due to restrained autogenous shrinkage can be superimposed by stresses due to restrained thermal expansion or contraction at any point of time. This scenario is common in practice. Generally, the worst case is a strong temperature decrease in the period of highest autogenous shrinkage cracking propensity. Then the stresses due to thermal contraction add up to those due to autogenous shrinkage. Such complex situations will not be further considered herein, however it should be kept in mind that cracks are possible even if the autogenous shrinkage cracking propensity is not very high.

The other scenario of virtually constant ambient temperatures is less complex and therefore more suitable for gradually improving the understanding of the influence of temperature on autogenous shrinkage, restraint stress and cracking propensity. Also, in a number of applications, for instance in precast works, this scenario comes relatively close to the real conditions. Accordingly, temperature changes are caused only by heat of hydration. Their magnitude mainly depends on the size of the concrete element and on the concrete composition. To further reduce the complexity, thermal strains can be completely eliminated (quasi-isothermal conditions).

In detail, four cases can be specified (Figure 5–8). The first and easiest case is a constant concrete temperature equivalent to the common laboratory temperature, in Germany e.g. 20 °C. No additional temperature control is required to avoid a significant temperature increase due to heat of hydration if specimens are sufficiently small. The second case comprises investigations at other practically relevant temperature levels, approximately in the range of 0 to 75 °C (concrete temperature). The higher the temperature level, the more difficult it becomes to provide for quasi-isothermal conditions because the temperature increase becomes larger with increasing hydration rate. The third case are combinations of different temperature levels, for instance a temperature of 40 °C until shortly before the onset of autogenous shrinkage and 20 °C afterwards. Thus the influence of temperature history can be investigated without the need of separating thermal and shrinkage strains. The forth case are variable concrete temperatures as they occur in medium-sized or large concrete elements if the concrete temperature is not actively controlled. In this case it is indispensable to know the thermal strains in order to analyze the influence of temperature on stresses and cracking due to restrained autogenous shrinkage.

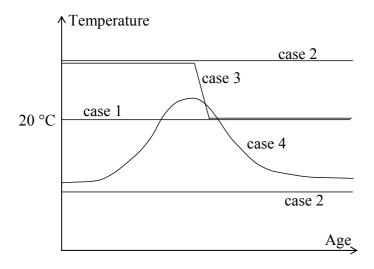


Figure 5–8 Four cases relevant for the experimental clarification of the influence of temperature on the autogenous shrinkage cracking propensity

Presently, the autogenous shrinkage cannot be modeled as function of temperature and time (2.5.2). Likewise, no general relation exists between temperature regime and autogenous shrinkage cracking propensity. Not even a tendency can be given. Hence, there is also no way to establish which temperature regime is worst in terms of cracking. A concrete could be less prone to cracking than another at a given temperature regime, but more if the temperature regime is different, irrespective of whether the temperature is constant or variable.

In conclusion, a stepwise, systematical investigation is essential to better understand the influence of temperature and to find out whether there are large differences in the autogenous shrinkage cracking propensity at different temperature regimes. Tests under quasi-isothermal conditions may or may not be a suitable way for comparing different concretes in that regard. Possibly the improved understanding will allow for omitting the most difficult and laborious case of variable temperatures in the future.

# 5.8 Further state of knowledge

### 5.8.1 Preliminary remarks on test methods

The following section is to summarize the most relevant results of previous quantitative investigations concerned with the cracking risk or stresses due to restrained autogenous shrinkage. The existing test methods can be subdivided into three groups: (1) tests with temperature-stress testing machines and other uniaxial stress rigs, (2) restrained ring tests and (3) other test methods. The test methods used most frequently, the temperature-stress testing machine and the restrained ring test, will be discussed in detail in chapters 6.2 and 6.3, respectively.

The most common test method of the third group is based on restraint due to reinforcement. Restraint stress in a prismatic concrete member can be calculated from the deformations of reinforcement bars [Sat 99], [Mat 99], [Ohn 99], [Sul 01], [Ima 04], [Cus 05]. Sato et al., for instance, found a reasonable agreement of predicted and measured stresses due to restrained thermal strains and autogenous shrinkage [Sat 99]. As was pointed out above, tests with reinforced concrete are more difficult to evaluate (5.3, cf. also [Sul 04]). Here, only investigations of unreinforced concrete will be considered.

Bentz et al. embedded a spherical sensor in sealed cement paste. The sensor consisted of two connected glass marble halves and contained a special wire whose conductivity changed with the surrounding pressure [Ben 01b]. This method may be a valuable alternative for the investigation of stresses due to autogenous shrinkage restrained by large aggregates. Bentur and Kovler, who classified early age cracking test methods, in addition mentioned panel tests which commonly are used to investigate plastic shrinkage cracking, and tests in which cracks result from non-uniform stresses in overlays or repair coatings which are restrained by the substrate [Ben 03b]. Here, these test methods are disregarded as they are not or less suitable for a comprehensive assessment of the autogenous shrinkage cracking propensity.

# **5.8.2** Quantitative investigations under restraint conditions

Schachinger et al. conducted tests with ultra-high strength concrete in a cracking frame, the precursor of the temperature-stress testing machine (cf. 6.2). They found that the maximum cracking risk, expressed as ratio of restraint stress to splitting tensile strength, was approximately 0.5 for ultra-high strength concrete made with a slag cement [Sch 02]. For ultra-high strength concrete with ordinary Portland cement, the ratio of stress to splitting tensile strength partly exceeded 1 between the age of 9 and 13 h, however, no cracks were observed. Schachinger attributed this to the fact that the uniaxial tensile strength at early age was higher than the splitting tensile strength [Sch 07] (cf. Figure 5–7).

Several investigations were carried out by means of temperature-stress testing machines. Hayano et al. found that the stress-strength ratio of high-strength concrete increased with decreasing water-cement ratio [Hay 08b] (Figure 5–9). In their study the uniaxial tensile strength at 7 d was measured directly in the temperature-stress testing machine on two specimens per concrete. For the age of 1, 3 and 7 d splitting tension tests on cylinders were carried out as well. The splitting tensile strength at the age of 7 d was similar to the uniaxial tensile strength. The ratio of stress to splitting tensile strength of the concrete with the lowest water-cement ratio (0.20) was between 0.8 and 0.9 at the age of 1, 3 and 7 d. Remarkably, the stress-strength ratio did not change very much with time, although the maximum restraint stress was reached after approximately 2 d (cf. [Hay 08b]). No cracks were observed.

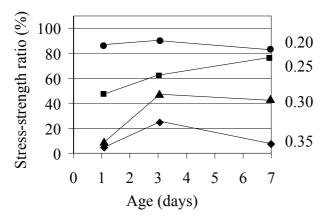


Figure 5–9 Influence of water-cement ratio (0.20, 0.25, 0.30, 0.35) on stress-strength ratio, from [Hay 08b]

Several other investigations with temperature-stress testing machines showed that presaturated light-weight aggregates and porous ceramic aggregates as well as expansive agents and shrinkage reducing agents can significantly reduce the stresses due to restrained autogenous shrinkage [Cus 05], [Suz 08], [Tan 08]. The maximum stress-strength ratio of high strength concrete that contained an expansive agent was less than 0.15 at 1, 3 and 7 d, whereas similar concretes without the expansive agent led to much higher stress-strength ratios [Hay 08b], cf. also **Figure 5–9**.

Bouasker et al. investigated the influence of temperature on stresses and cracking. Restrained ring tests with sealed cement pastes at constant temperatures of 20 °C and 25 °C were carried out [Bou 08a]. (The measurement setup employed is shown in **Figure 6–7**). The age at cracking was lower at higher temperature (**Figure 5–10**); the degree of hydration at cracking, simulated with the model CemHyd3D, was similar for both temperature histories (**Figure 5–11**).

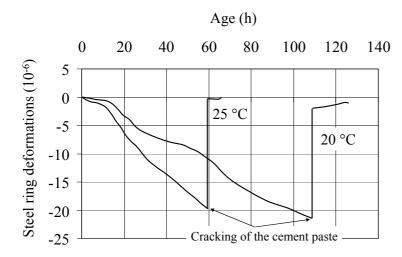


Figure 5–10 Restrained ring tests on cement paste, w/c = 0.30, constant temperatures of 20 °C and 25 °C: steel ring deformation vs. age [Bou 08a]

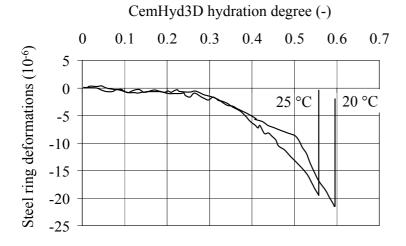


Figure 5–11 Restrained ring tests on cement paste, w/c = 0.30, constant temperatures of 20 °C and 25 °C: steel ring deformation vs. simulated degree of hydration [Bou 08a]

Weiss et al. conducted restrained ring tests on cement paste to investigate the influence of a shrinkage-reducing admixture [Wei 08]. A dual ring was employed to also monitor compressive stresses (cf. Figure 6-4). Specimens were sealed until the age of 7 d. The tensile stress at

that age was approx. 3.5 MPa and 2.5 MPa in pure cement paste and in cement paste with shrinkage reducing admixture, respectively (**Figure 5–12**). Visible cracks occurred only after drying from the top was allowed for.

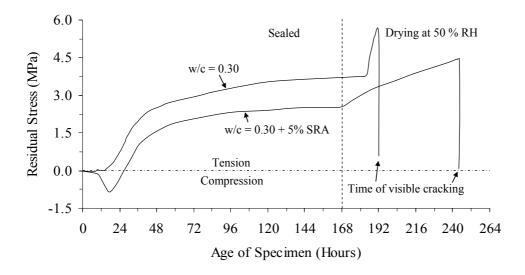


Figure 5–12 Restrained ring tests (dual ring) on cement paste with and without shrinkage-reducing admixture (SRA) (from [Wei 08])

Bentz et al. investigated the influence of cement fineness on stresses [Ben 08b]. Again, a dual ring was employed. Tests were carried out at constant temperature of 23 °C. Two Portland cements (ASTM type I/II) were used with a fineness of 380 cm²/g and 311 cm²/g, respectively. The water-cement ratio was 0.35 and the cement pastes contained 0.5 % of a water-reducing agent. The finer cement caused early age cracking while the coarser cement did not (Figure 5–13). The coarser cement between approximately 10 and 60 h led to compressive stresses, whereas the finer cement caused tensile stresses during the whole test period.

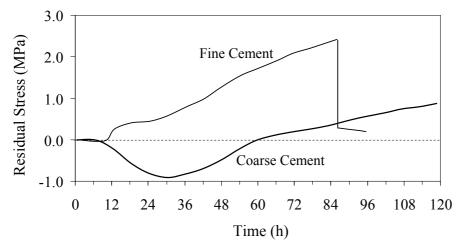
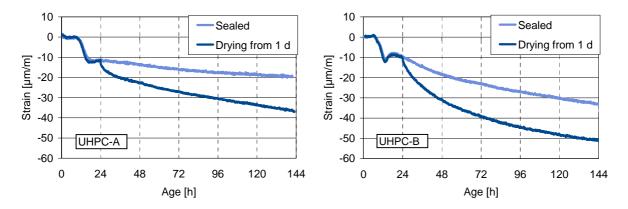


Figure 5–13 Restrained ring tests (dual ring) on sealed cement paste: influence of fine (380 cm<sup>2</sup>/g) and coarse (311 cm<sup>2</sup>/g) cement (w/c = 0.35, T = 23 °C, 0.5 % water-reducing agent) [Ben 08b]

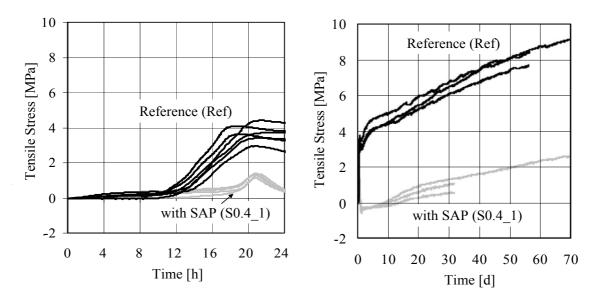
Hossain et al. employed the restrained ring test to investigate restraint stresses due to autogenous and drying shrinkage [Hos 08a], [Hos 08b]. The focus of the investigations was on the effect of different fly ashes. The authors used a typical ASTM class F fly ash and an ultra-fine fly ash with a mean particle size of 25 µm and 3 µm, respectively. The mortars that were produced contained 50 Vol.-% of sand. The water-cement ratio was 0.35, cement ASTM type I was used. Since the concrete ring and the specimens for free shrinkage had relatively large cross sections (approx. 65 x 75 mm² and 75 x 75 mm², resp.) a temperature increase of more than 5 K was observed due to heat of hydration within the first 24 h. No attempt was made to separate thermal and shrinkage strains and their effect in restrained ring tests during this period. The cracking propensity after that age was quantified as ratio of restraint stress to splitting tensile strength, however, rings were allowed to dry from top and bottom. Hence, no conclusions as to the autogenous shrinkage cracking propensity were possible.

Own results of restrained ring tests on ultra-high strength concrete with an equivalent water-cement ratio of 0.20 and a cement content of 800 kg/m³ were presented in [Epp 08c]. Two concretes named UHPC-A and UHPC-B were tested, differing only in the used cement. The cements A and B significantly differed in the content of C<sub>3</sub>A and were similar in most other respects (cf. **Table 9–1**). The content of C<sub>3</sub>A was approximately 6 to 7 % higher with cement B (cf. 2.5.1). Six rings per concrete were produced and stored at 20 °C. Two rings remained sealed over the whole test period. Four rings were unsealed at the age of 24 h. These rings dried out from top and bottom at 65 % relative humidity. The concrete temperature during the entire test period was between 21 °C and 22.7 °C. The deformations imposed on the steel ring, being proportional to the hoop stresses, were higher with the cement with higher C<sub>3</sub>A content (**Figure 5–14**). Drying significantly increased the strains, and strains continued to increase during the whole test period of seven days. The degree of restraint during these tests constantly exceeded 80 %. Neither in the sealed nor in the unsealed rings any visible cracks occurred. From the absence of cracks in the unsealed rings it was concluded qualitatively that the cracking propensity in the sealed specimens between the age of 24 h and 144 h was low.



**Figure 5–14** Restrained ring tests under quasi-isothermal conditions with ultra-high strength concrete: influence of different cements [Epp 08c]

Mechtcherine and Co-Workers comprehensively investigated the effect of internal curing by means of superabsorbent polymers on restraint stress in ultra-high strength concrete [Mec 06], [Mec 08], [Mec 09], [Dud 10]. Stress was much lower when superabsorbent polymers (0.4 mass-% of cement) were added to the reference concrete (Figure 5–15).



**Figure 5–15** Tensile stress in restrained ring tests with sealed ultra-high strength concrete with and without superabsorbent polymers, 0-24 h (left) and 0-70 days (right) [Dud 10]

### 5.8.3 A full-scale model for assessing the cracking risk at very early age

Sorelli et al. recently presented a full-scale thermo-chemical-mechanical model for the assessment of the early-age cracking risk in structures made of ultra-high strength concrete [Sor 08]. The model is to combine autogenous shrinkage and thermal deformations due to heat of hydration. As was shown above, the general relation between temperature and autogenous shrinkage is unclear and maturity approaches may not be applicable (2.5.2). Remarkably, the development of shrinkage in the model is based on a single value, namely the final drying shrinkage strain.

In the reported application a specific fine-grained ultra-high strength concrete was used whose properties were described in a report issued by the Federal Highway Administration of the US Department of Transportation [Fed 06]. The long-term shrinkage measurements mentioned in that report were carried out according to ASTM C 157 on unsealed specimens and started at the age of approx. 23 h. The final drying shrinkage strain reported for concrete without heat-curing was  $555 \,\mu\text{m/m}$ . The final set apparently occurred at approx. 16 h.

The model was applied to a bridge girder made of the aforementioned ultra-high strength concrete reinforced with fibers. The value for final drying shrinkage was taken from [Fed 06]. The bridge girder's deformations were restrained by formwork. The measured concrete temperature varied between (approx.) 10 °C until 50 h and a maximum of 70 °C at 80 h. From the late increase of temperature at the age of 50 h it can be concluded that hydration was considerably retarded, either due to the low temperature alone or, more probably, also due to high amounts of chemical admixtures. Hence, possibly the concrete composition used for the girder was different from the one the shrinkage was measured for.

The monitored mid-span section of the bridge girder did not begin to deform until (approx.) 55 h to 60 h after casting. The calculated maximum stress-strength ratio due to restrained deformation was 0.39 and occurred 63 h after casting, i.e. few hours after the development of the first strains and roughly 17 h hours before the concrete temperature started to decrease.

According to the definition suggested above, the age of 63 h in this case would belong to the very early age (cf. 5.4).

Good agreement was observed between calculated and measured strains in this special case. However, the assumption that the risk of cracking due to restrained autogenous shrinkage at very early age and under non-isothermal conditions can be predicted with the final drying shrinkage strain as the only shrinkage input value, is in entire disagreement with fundamentals. Moreover, the development of the tensile strength was modeled according to the linear model of Laube. This simple model was shown not to be applicable at very early age (3.5.4).

Certainly simplifications are necessary for assessing the cracking risk of structures made of ultra-high strength concrete. However, more sophisticated input functions for the autogenous shrinkage and the development of the tensile strength appear to be required. The approach of Gawin et al. may be appropriate, however, it is very complex [Gaw 06a], [Gaw 06b].

# 5.9 Summary

Autogenous shrinkage is caused by self-desiccation that occurs predominantly in concretes with very low water-cement ratio (2.1). It can lead to cracks if deformations are restrained (5.3). For obvious reasons it is not sufficient to measure the free autogenous shrinkage in order to investigate the cracking propensity. Cracking at early and very early age is a complex process that considerably depends on the stress relaxation capacity and the development of mechanical properties in general (5.4).

The risk of cracking commonly is expressed as stress-strength ratio at a specific age or as the maximum stress-strength ratio occurring within a given period. The stress-strength ratio failure criterion is based on assumptions partly not validated. Results therefore are not necessarily precise and generally valid (5.6). The term 'cracking propensity', in contrast to the term 'cracking risk', is to indicate that results presently do not allow for a reliable calculation of the probability of cracking.

As far as the autogenous shrinkage cracking propensity is concerned, the present knowledge is scarce. There is very few data for the cracking propensity at very early age. Most investigations covered only a few discrete ages and not necessarily those of highest cracking propensity. The few results available indicate a high cracking propensity of concretes with very low water-cement ratio (5.8.2). However, most of these results are based on the splitting tensile strength. It is unclear if and how splitting tension tests can be used to determine the resistance to uniaxial tensile failure at very early age (5.6).

Some additional results of the restraint stress development are available (5.8.2). The investigations carried out so far mainly confirm in a rather qualitative manner the strong influence of the autogenous shrinkage on stresses. Several measures that reduce the autogenous shrinkage were shown to reduce the restraint stress as well, for instance shrinkage-reducing admixtures, super-absorbent polymers, less fine cements and cements with less C<sub>3</sub>A.

The autogenous shrinkage cracking propensity presumably is significantly influenced by temperature history (5.7). This needs to be confirmed. It is completely unclear which temperature history yields the maximum cracking propensity. Furthermore, stresses due to thermal con-

traction may superimpose shrinkage stresses. Hence, for comprehensively assessing the autogenous shrinkage cracking propensity it is not sufficient to carry out stress tests under quasi-isothermal conditions.

The influence of the relaxation capacity on the autogenous shrinkage cracking propensity has not been sufficiently investigated, yet. This can be attributed mainly to the substantial experimental difficulties, the enormous effort required and the lack of suitable and efficient test methods. Furthermore, the influence of creep and relaxation on cracking at very early age is difficult to model because basic assumptions made for mature concrete under service loads do not apply (5.4, 5.5).

Models will not allow for an efficient and comprehensive prediction of the autogenous shrinkage cracking propensity in the near future (cf. 2.2, 5.5, 5.8.3). To understand the behavior at early and very early age, it is important to learn more about involved micromechanisms, however, the complexity of the task impedes a short-term application to practical problems. Hence, the potential of the phenomenological approach needs to be evaluated and enhanced in order to facilitate the construction with ultra-high strength concrete.

If test methods were available that allowed for multiple testing, at best without any restrictions as to the temperature regime, useful empirical equations may be extractable from experimental data. Possibly, results of stress tests at variable temperature can be related to results of quasi-isothermal tests in a way that reduces future test efforts. Similarly, results of stress tests maybe can be related to the free autogenous shrinkage. Up to now, there is simply too few data to attempt this.

Before the major questions regarding the autogenous shrinkage cracking propensity and the influence of stress relaxation and temperature can be answered, the various test methods need to be analyzed and, if possible, improved, or new test methods need to be developed. Presently, the main methods are tests with uniaxial stress rigs and restrained ring tests. In the next chapter the suitability of temperature-stress testing machines being the most widespread type of uniaxial stress rig is analyzed. As a consequence of this analysis, the present study was not carried out with the available temperature-stress testing machine but focused on the potential of the restrained ring test for answering the open questions (cf. 6.3 and 6.4).

### 6 Investigation of the autogenous shrinkage cracking propensity

#### 6.1 Introduction

This chapter comprehends the own experimental investigations on restraint stress and cracking propensity. The presentation of the results is preceded by a discussion of two alternative test methods. First, the suitability of temperature-stress testing machines for investigating the autogenous shrinkage cracking propensity is examined. From the characteristics of such machines and the results of two round-robin tests the further investigation turns to the restrained ring test. First, its principal setup, possible variations and the basic evaluation of results are described. The influence of temperature changes in restrained ring tests is examined and an analytical solution for calculating the stress due to temperature changes is derived. In addition, it is shown how temperature changes can be used to investigate the effects of creep and relaxation. Then the own investigations with restrained ring tests are introduced, starting with a description of the setup and the measuring technique. The measured steel ring strains are converted to stresses; degree of restraint, theoretical-elastic stress and a creep factor are calculated. The further analysis comprises the suitability of the splitting tensile strength as resistance measure, the influence of radial stress redistribution and the use of ring tests at very low humidity for quantifying the resistance to cracking. The autogenous shrinkage cracking propensity is shown to be assessable as the ratio of restraint stress and corrected splitting tensile strength. Moreover, the thermal stress component is examined further and the period of maximum cracking propensity is analyzed. Finally, restraint stress and cracking propensity are looked at as functions of the autogenous shrinkage and the feasibility of a more detailed evaluation of the influence of creep and relaxation is studied.

# 6.2 Suitability of temperature-stress testing machines

### **6.2.1** Development, setup and use

The temperature-stress testing machine was developed by Springenschmid et al. in the 1980's. It is a uniaxial tensile and compressive stress testing machine that allows monitoring and controlling strains and stresses in a centrically restrained concrete specimen with a large cross section of typically 150 x 150 mm² and a length of approximately 1500 mm (Figure 6–1, Figure 6–2). A temperature-controlled mold is used to regulate the temperature of the specimen [Spr 85], [Sch 90]. The development was based on the cracking frame, consisting of a specimen restrained by a rigid steel frame [Bre 89], [Spr 95b]. The temperature-stress testing machine, contrary to the more simple cracking frame, can provide for full restraint during the whole test period. To this end the length of the specimen is kept constant by means of a movable crosshead, a principle which had already been applied by other researchers before [Pai 76], [Kas 75].

If the free deformations are measured in parallel on a second specimen, this data can be used to control the degree of restraint in the restrained specimen [Kov 94]. Any partial degree of restraint can be achieved. A constant degree of restraint of 60 %, for instance, requires to pull back the cross-head 3/5 of the free deformations. This is constantly repeated each time a defined deformation threshold is reached in the unrestrained specimen. Alternatively, partial degrees of restraint can be achieved by short-term measurements of the modulus of elasticity of the concrete in the temperature-stress testing machine and the appropriate consideration of the stiffness of the testing machine [Hay 08a].

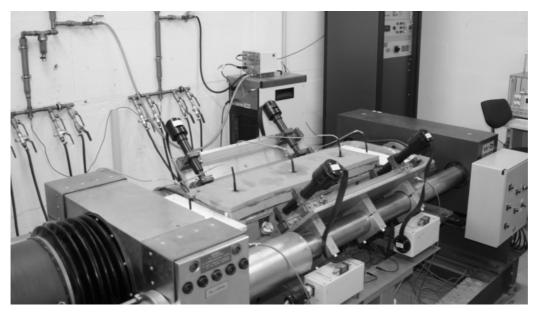
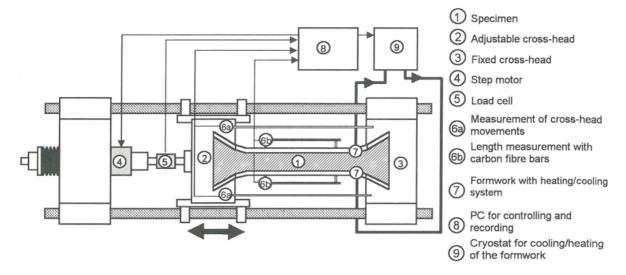


Figure 6–1 Temperature-stress-testing machine at VDZ (German Cement Works Association)



**Figure 6–2** Scheme of the temperature-stress testing machine [Spr 95]

At the time the temperature-stress testing machine was developed, the focus was on mass concrete. The influence of the heat of hydration of cements on cracking in restrained concrete was discussed controversially [Pai 76], [Spr 95]. The fundamental problem was well known. Decreasing temperatures can cause high Eigen- and restraint stresses. The resultant risk of cracking traditionally was addressed by keeping the concrete temperature as low as possible. In fact, tests with temperature-stress testing machines confirmed that the maximum temperature rise at given storage conditions is the most important property of a concrete composition in terms of cracking risk [Thi 95], [Hin 98].

Today, temperature-stress testing machines, due to their versatility and despite the considerable costs for acquisition and deployment, can be found in a number of institutions. Many questions regarding the use of mass concrete in dams, power plants or tunnel shells can be considered resolved. Tests mostly were conducted at full restraint. Results initially were trans-

formed to arbitrary degrees of restraint by elastic calculations [Spr 95b]. Models were developed to also account for creep [Ham 08]. Strains and mechanical parameters were modeled as functions of age and temperature [Lau 90]. Ample experimental data gave rise to models apt to predict stresses and cracking risk due to heat of hydration [Cze 05].

Contrary to mass concrete, high strength concrete, whose specific cracking propensity was fully realized in the 1980's and 1990's [Bui 80], [Zie 82], [Taz 92], [Sel 95] exhibits significant autogenous shrinkage. The autogenous shrinkage is notably influenced by temperature, however, this influence presently cannot be described by maturity functions (2.5.2). Changing temperatures have a twofold effect: they cause thermal strains and they alter the shrinkage strains. The stresses measured in tests under restraint conditions are the result of the sum of both kinds of strains. Two requirements arise from that. First, it is important to be able to precisely control the concrete temperature. This includes the reproducibility of temperature over time as well as the uniform distribution of temperature over the specimen. Second, knowledge of the coefficient of thermal expansion is required to separate the effects of thermal and shrinkage strains (5.7).

These requirements are difficult to meet with large specimens. To provide for quasi-isothermal conditions, RILEM recommended a maximum temperature increase due to heat of hydration of 2 K [Ham 06]. Beyond that value, thermal strains as well as the alteration of autogenous shrinkage and other properties may not be negligible anymore. Although some tests under quasi-isothermal conditions were carried out with temperature-stress testing machines [Sch 95], it appears questionable whether the mentioned threshold can be met under all circumstances with cross sections exceeding 100 mm x 100 mm. Especially with ultra-high strength concrete and at high temperatures the heat of hydration is released rapidly, so the external cooling may not be sufficient to avoid differences of more than 2 K.

Similar considerations apply to the determination of the coefficient of thermal expansion by means of temperature-stress testing machines. Hammer et al. suggested increasing and decreasing the concrete temperature stepwise and evaluating the respective length changes to determine the coefficient of thermal expansion [Ham 06]. Obviously, the steps of temperature increase must not be too large in order not to significantly alter hydration and not too small in order to achieve a sufficient precision. The main problem, however, is the thermal inertia of large specimens. Even with sophisticated cooling and heating systems it may not be possible in all cases to achieve an evenly distributed concrete temperature within a sufficiently short period of time.

Loser et al. recently suggested a procedure for measuring the coefficient of thermal expansion [Los 10]. They filled a condom with 10 ml of cement paste and immersed it into a temperature-controlled bath. Then the bath temperature was increased by 6 K. It took 28 min for the temperature to be equilibrated inside the small sample. In comparison, the volume that needs to be heated up in conventional temperature-stress testing machines is by far larger, so it takes more time for temperature to be equilibrated. The autogenous shrinkage during that time span superimposes the thermal strains. Hence, to calculate the coefficient of thermal expansion, the autogenous shrinkage and the thermal strains would have to be separated. This, however, would already require the coefficient of thermal expansion, i.e. the property that still needs to be determined. Alternatively, some mean temperature obtained after a few minutes may be used, or the autogenous shrinkage may be measured in parallel under quasi-isothermal condi-

tions. In any case, it appears questionable whether the coefficient of thermal expansion can be determined accurately and efficiently by means of conventional temperature-stress testing machines. Notably, comparable uniaxial stress rigs with much smaller specimens have been developed recently [Koe 06], [Hab 06], [Ma 05], [Tom 00].

Another important aspect of the use of temperature-stress testing machines is the large effort required for a single test. In view of the unclear influence of temperature and stress relaxation on the autogenous shrinkage cracking propensity, a number of tests as large as possible would be desirable (5.4, 5.7). However, in many institutions only one machine is available and therefore only one specimen at a time can be tested. Actually, most investigations with temperature-stress testing machines comprised only one specimen per investigated variable (or no information on the number of specimens was given), e.g. [Hay 08a], [Cus 07], [Lin 06], [Iga 06], [Bra 06], [Tao 06], [Ohy 06], [Tak 05], [Ma 05], [Lur 03c], [Atr 01], [Tom 00], [Iga 00]. From a statistical point of view such investigations have a very limited significance, especially if the precision of the method is unclear (cf. 6.2.2).

Finally, tests of the autogenous shrinkage cracking propensity with temperature-stress testing machines mostly were carried out at full restraint [Lin 06], [Iga 06], [Bra 06], [Ohy 06], [Koe 06], [Iga 02]. However, few applications of high- and ultra-high strength concrete with full restraint are conceivable. It is uncertain whether investigations at full restraint alone allow for a reliable comparison of the autogenous shrinkage cracking propensity of different concretes. Higher degrees of restraint are associated with higher stress levels. The stress relaxation behavior may be different at lower degrees of restraint.

# 6.2.2 Results of round robin tests

It was hypothesized that thermal and autogenous shrinkage deformations superimpose each other in a way that substantially increases the risk of cracking of high-strength concrete [Bjø 04]. Therefore, several European research groups in the 1990's commonly attempted to investigate the joint effect of thermal and autogenous shrinkage strains with temperature-stress testing machines. The comprehensive round robin test IPACS involved five laboratories. Free autogenous shrinkage was measured at a constant set temperature of 20 °C and at a realistic temperature history with a maximum temperature of 40 °C [IPACS]. In three laboratories, measurement was carried out on large prismatic specimens. In two laboratories, cylindrical specimens were used, allowing for a start of the measurement at a concrete age of 24 h. Results are given in Figure 6–3.

Results obtained under non-isothermal conditions were called discouraging [Ham NC]. The scatter at quasi-isothermal conditions can be considered unsatisfactory, too. Bleeding, the temperature control system and temperature effects on the equipment were mentioned as potential sources of error [IPACS]. A part of the scatter was attributable to the non-conformity of the concrete mixtures in the participating laboratories, noticeable from supportive tests. It may be added that the overall autogenous shrinkage was low compared to several hundreds of microstrain typically generated in ultra-high strength concrete (cf. 4.5). Thus, disturbing influences of absolute size can lead to a large relative error.

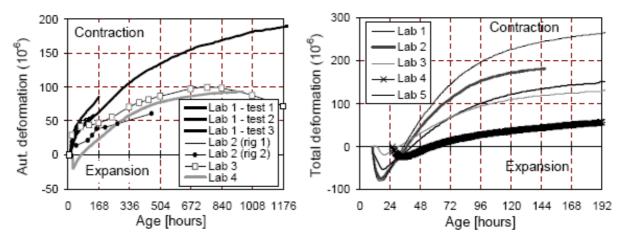


Figure 6–3 Results of the IPACS round robin test. Left: free autogenous deformation at a constant set temperature of 20 °C; right: overall deformations at a typical non-isothermal temperature regime with a maximum target temperature of 40 °C; measured on prismatic specimens (cross sections  $\geq$  100 x 100 mm², results zeroed at 12 h) or on cylinders (measurement in that case started at 24 h). [Ham NC]

Furthermore, tests under restraint conditions with temperature-stress testing machines were carried out. Full restraint was applied. Results for quasi-isothermal and non-isothermal conditions are given in **Figure 6–4** and in **Figure 6–5**, respectively. In view of the somewhat smaller scatter it was concluded that the "*intrinsic reproducibility*" of the test method was "*very high*" [IPACS]. Yet, this in part may also be explained by the different nature of the tests. Free deformations at very early age are more difficult to measure; when restrained, they contribute less to the build-up of restraint stress than later deformations since the modulus of elasticity is still low and relaxation capacity still high. Thus, the scatter possibly was diminished by the restraint.

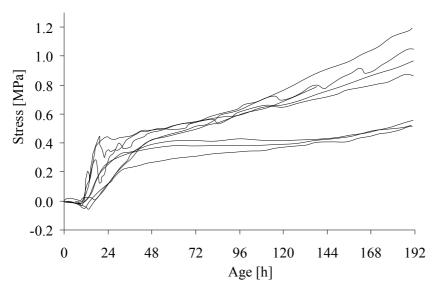
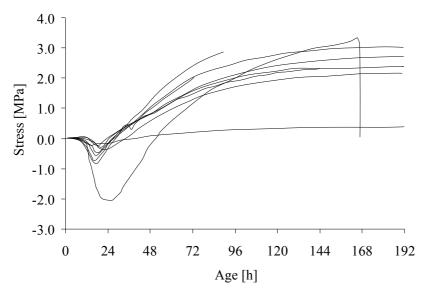
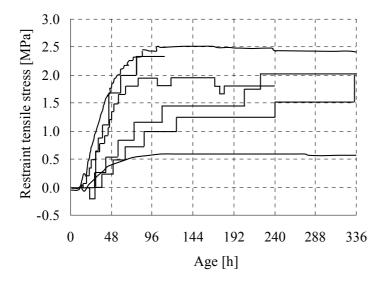


Figure 6-4 Results of the IPACS round robin test: restraint stress at a constant target temperature of 20 °C; measured in temperature-stress testing machines on prismatic specimens with cross sections ≥ 100 x 100 mm² (from [IPACS])



**Figure 6–5** Results of the IPACS round robin test: restraint stress at a typical non-isothermal temperature regime with a maximum target temperature of 40 °C; measured in temperature-stress testing machines on prismatic specimens with cross sections ≥ 100 x 100 mm² (from [IPACS])

Another round robin test was carried out more recently, including some participants of the former one [Bjø 06]. This time a more simple concrete composition was chosen in order to reduce the disturbing influence of differing concrete properties in the participating laboratories. The concrete contained 522 kg/m³ of a Portland cement CEM I 42.5 R, dry aggregates with a maximum particle size of 8 mm and superplasticizer as the only admixture. Presumably participants were aware of all potential sources of errors. However, results were hardly better than in the first round robin test (**Figure 6–6**). At the age of 48 h, for instance, the six tests with five temperature stress testing machines yielded a restraint stress of approx. 0.4, 0.4, 0.5, 1.4, 1.5 and 1.8 MPa, resulting in a mean value of 1.0, a standard deviation of approx. 0.64 MPa and a coefficient of variation of 64 %.



**Figure 6–6** Result of a round robin test: restraint stress due to autogenous shrinkage, tested with temperature-stress-testing machines (100 % restraint, constant set temperature of 20 °C) (from [Bjø 06])

Rather than proving a high reproducibility, the overall results of the two round robin tests justify the assumption that the test method when applied to autogenous shrinkage does not feature a sufficient reproducibility.

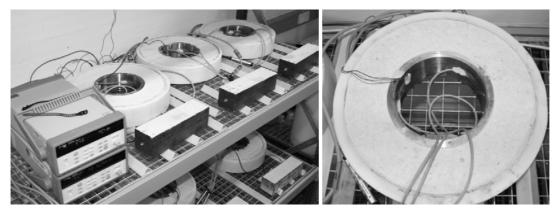
# **Restrained ring test - methodological foundations**

# 6.3.1 Setup and use

The ring test basically consists of a concrete ring whose contraction is restrained by a steel ring. Shrinkage of the concrete compresses the steel ring and induces stresses in both rings. The resultant deformation imposed on the steel ring can be measured; usually three to four strain gauges are glued to the inside of the steel ring for that purpose. Due to the largely concentric radial stresses the single strain gauges should give similar values as long as no cracking has occurred. A strong and sudden decrease of deformations indicates the formation of a major crack in the concrete ring.

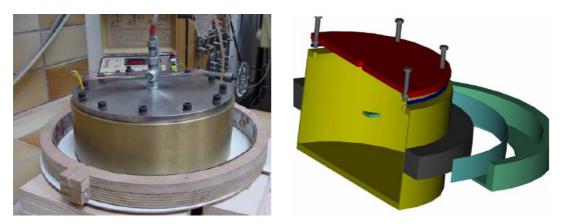
The cross section of the concrete ring may be chosen according to the maximum grain size. The degree of restraint is a function of the width of the two rings and of the modulus of elasticity (cf. 5.2). Different degrees of restraint may be achieved by different widths of rings [Wei 00]. The heights of steel and concrete ring commonly are equal.

The concrete ring can be considered an infinitely long prismatic specimen [Lam 04]. Restraint is created without any coupling parts or embedded elements. Thus, local stress concentrations are negligible. Contrary to uniaxial stress rigs, with the common ring test setup the stresses cannot be actively controlled during the test. Hence, the restrained ring test belongs to the passive test methods. A typical restrained ring test setup is shown in **Figure 6–1**.

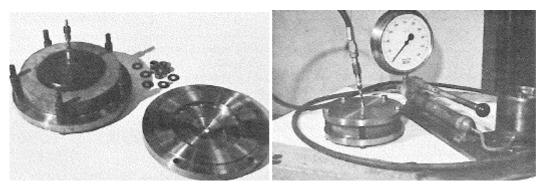


**Figure 6–1** Typical restrained ring test setup with rings drying from top and bottom (also shown: specimens for measuring free shrinkage), left: rings with data acquisition unit, right: top view on ring

Several modifications and alternative setups have been developed, for instance a pressurized ring for applying full restraint [Lam 04] (**Figure 6–2**). A similar setup was developed for testing the tensile strength on ring specimens [May 70] (**Figure 6–3**). A dual ring with inner and outer ring was used to additionally account for compressive stress [Wei 03], [Wei 08], and an eccentric ring was employed to investigate the effect of dimensional variations (**Figure 6–4**). An elliptic ring was built to achieve earlier cracking (**Figure 6–5**). Furthermore, there are two standards for restrained ring tests frequently used especially in the USA [AASHTO 1], [ASTM 1].



**Figure 6–2** Photo (left) and scheme (right) of a ring setup with pressurized cylinder [Lam 04]



**Figure 6–3** Test setup for testing the tensile strength on a concrete ring [May 70]

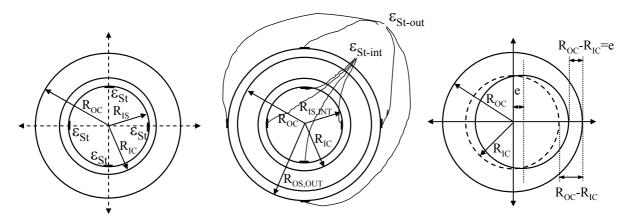
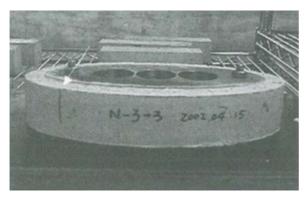


Figure 6-4 Ring test variations: standard ring (left), dual ring (centre), eccentric ring (right) (from [Rad 06])

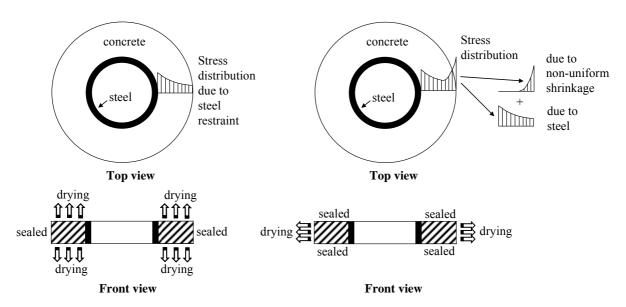


**Figure 6–5** Restrained ring test with elliptic ring [Zhe 04]

In the past the restrained ring test for the very most part was used for investigating cracking due to drying shrinkage [Swa 79], [Kov 93], [Wie 96], [Mar 00], [See 03], [Hos 04], [Cza 05], [Rad 06], [Hos 06], [Roz 07], [Pie 08], [Pas 09], more often with plain than with fiber-reinforced concrete. Presumably the first such tests were carried out by Carlson and Reading in the 1940's [Car 88]. Especially in earlier studies the evaluation was based on the age at cracking. The age at which the concrete ring cracks depends on the ring geometry and on the distribution of stresses over the cross section [Wei 00], [Att 04]. Therefore results of the age at cracking obtained with geometrically different test setups may not be comparable.

For a more quantitative analysis the tensile stresses can be computed from the measured steel ring deformations. Notably, the circumferential tensile stress, also called hoop stress, is not constant along the radius but changes nonlinearly. Given that the circumferential shrinkage is uniform in radial direction, the maximum hoop stress occurs at the inside of the concrete ring and decreases along the radius. Shrinkage can be considered uniform if the ring is completely sealed (autogenous shrinkage) or dries out from top and bottom (**Figure 6–6**, left). In this case the maximum hoop stress at the inner radius of the concrete ring is proportional to the steel ring deformation [Hos 04], [Hos 06]. The proportionality factor depends on the dimensions of the rings and the modulus of elasticity of the steel (cf. 6.3.2.2). Moisture and hoop stress gradients in the direction of the rotation axis of the concrete ring, occurring with unsealed top and bottom surfaces, usually can be disregarded.

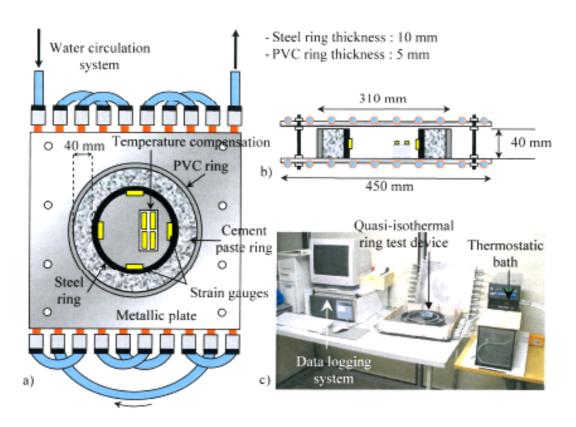
If the concrete ring dries out laterally, shrinkage is not uniform in the radial direction, making stress analysis more difficult [Moo 06]. The maximum stress may occur either at the inside or the outside of the concrete ring (**Figure 6–6**, right). Shah and Co-Workers recently provided stress solutions for this drying condition which also include the effect of fibers [Kwo 08a], [Kwo 08b], [Pas 09]. In a study with rings that dried from top and bottom, fibers were found to influence the cracking behavior only after the opening of cracks [Sha 06].



**Figure 6–6** Restrained ring test: direction of drying and stress distribution (from [Kim 03])

If the tensile strength is known, the cracking propensity may be calculated as ratio of hoop stress and tensile strength (cf. 5.6). Furthermore, if the free shrinkage and the modulus of elasticity of concrete are known, the degree of restraint and the influence of creep may be quantified as well. The evaluation of restrained ring tests is comprehensively dealt with in the next section (6.3.2).

Very few investigations of the autogenous shrinkage cracking propensity were carried out with restrained ring tests to date. The suitability of the restrained ring test for measuring stresses due to restrained autogenous shrinkage was hardly discussed, yet. Especially the effect of changing temperatures on stresses was largely disregarded. As concrete and steel ring have different coefficients of thermal expansion, thermal stresses may superimpose stresses due to autogenous shrinkage. Hence, temperature control is essential. Bouasker et al. developed a setup with temperature-controlled plates on top and bottom of the ring to provide for quasi-isothermal conditions (Figure 6–7) [Bou 06].



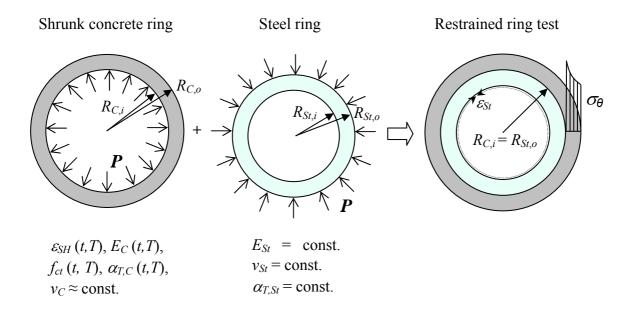
**Figure 6–7** Restrained ring test setup with temperature control system [Bou 06]

Since it is necessary to investigate the autogenous shrinkage cracking propensity also under non-isothermal conditions, the role of temperature deserves special attention (cf. 5.7). In section 6.3.2.3 a method for the separation of thermal and shrinkage stress component is presented. The use of stresses created by specific temperature changes is treated in section 6.3.3.

### **6.3.2** Evaluation of restrained ring tests

#### 6.3.2.1 Introduction

Consider two rings, one of concrete (subscript C) and one of steel (subscript St) with respective inner (i) and outer (o) radius  $R_i$  and  $R_o$ , furthermore  $R_{C,i} = R_{St,o}$  (Figure 6–8). While material parameters of concrete depend on time and temperature, relevant parameters of steel can be considered constant within the temperature range of common concrete applications. Due to restraint, shrinkage of the concrete ring yields an interface pressure P between the rings which is constant along the common surface (cf. Figure 5–2). This pressure leads to radial and circumferential stresses in both rings. The circumferential (hoop) stress  $\sigma_{\theta}$  in the concrete ring is positive, i.e. tensile stress. The circumferential deformations  $\varepsilon_{St}$  which are imposed on the steel ring can be measured by strain gauges mounted on the inside of the steel ring.



 $R_{C,i}$ : inner radius of concrete ring

 $R_{C,o}$ : outer radius of concrete ring

 $R_{St,i}$ : inner radius of steel ring  $R_{St,o}$ : outer radius of steel ring

 $\varepsilon_{SH}$ : free shrinkage

t: time

*T*: temperature

 $E_C$ : modulus of elasticity of concrete

 $f_{ct}$ : tensile strength of concrete

 $\alpha_{T,C}$ : coefficient of thermal expansion of concrete

 $v_C$ : Poisson's ratio of concrete  $E_{St}$ : modulus of elasticity of steel

 $v_{St}$ : Poisson's ratio of steel

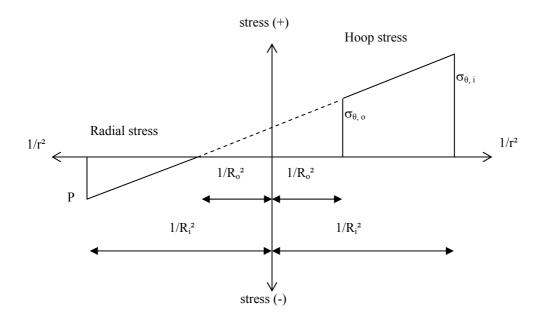
 $\alpha_{T,St}$ : coefficient of thermal expansion of steel  $\varepsilon_{St}$ : circumferential strain of the steel ring

 $\sigma_{\theta}$ : hoop stress

Figure 6–8 Scheme of restrained ring test and relevant parameters

Since pressurized tube sections are a common issue in engineering, the basic equations have been known for a long time. The general solution for stresses in an elastic pressurized ring is also called 'thick-wall' solution, in contrast to the 'thin-wall' solution which neglects the fact that the hoop stress is not constant along the radius. The solution for thin walls is applicable if the ratio of ring thickness and radius is smaller than approximately 1/30. With most setups of restrained ring tests this value is exceeded, so the thick wall solution needs to be applied, accounting for the non-linear distribution of stress along the ring's radius. In the following, the thick wall solution is described.

Consider a ring under internal pressure P with inner and outer radius  $R_i$  and  $R_o$ , respectively. This ring corresponds to the shrunk concrete ring in the restrained ring test. The function for radial and hoop stress at any radius r of the ring ( $R_i \le r \le R_o$ ) can be illustrated by a straight line plotted over  $1/r^2$ , the so called Lamé line (**Figure 6-9**) [Ros 87]. Equating similar triangles yields Equation 6–1. The radial stress is negative (compression) and becomes zero for  $r = R_o$ . The hoop stress is positive along the complete radius (tension) and reaches its maximum at the inner radius  $r = R_i$ . The difference in hoop stress at inner and outer radius expressed as percentage decreases with increasing radius and decreasing ring thickness.



**Figure 6–9** Hoop and radial stress in a ring with inner and outer radius  $R_i$  and  $R_o$  due to internal pressure P [Ros 87]

$$\sigma_{\theta}(r) = P \cdot \frac{\frac{1}{R_o^2} + \frac{1}{r^2}}{\frac{1}{R_i^2} - \frac{1}{R_o^2}}$$
 Equation 6–1

The maximum hoop stress at the inner radius can be calculated as per Equation 6-2. The stress is proportional to the pressure P. The proportionality factor results from the geometry

of the rings. The only unknown quantity is P. The next section describes how P can be calculated to obtain the maximum hoop stress in the concrete ring.

$$\sigma_{\theta,\text{max}} = P \cdot \frac{R_{C,i}^2 + R_{C,o}^2}{R_{C,o}^2 - R_{C,i}^2}$$
 Equation 6–2

# 6.3.2.2 Theoretical-elastic and actual stress, degree of restraint, creep factor

There are two different stress quantities that matter, the theoretical-elastic stress  $\sigma_{elastic}$  and the actual stress  $\sigma_{actual}$ . The respective interface pressures are  $P_{elastic}$  and  $P_{actual}$ . The interface pressure  $P_{elastic}$  would occur if concrete deformed purely elastically. It can be computed by means of elastic relations. The actual interface pressure  $P_{actual}$  can be calculated from the strains measured at the inner radius of the steel ring [Hos 04].

To obtain the theoretical-elastic stress, first consider a radial displacement  $|u_{free}|$  of the unrestrained concrete ring due to shrinkage. It can be calculated as per Equation 6–3 in which  $\varepsilon_{SH}$  is the free shrinkage strain, measured by an appropriate method.

$$|u_{free}| = R_{C,i} \cdot \varepsilon_{SH}$$
 Equation 6–3

 $\varepsilon_{SH}$ : free shrinkage strain

In the restrained ring system, the resultant interface pressure  $P_{elastic}$  due to shrinkage concentrically acts on the steel ring from the outside and on the concrete ring from the inside. Provided that the boundary condition  $R_{C,i} = R_{St,o}$  (6.3.2.1) remains valid under pressure, i.e. no gap occurs between the rings, the same relations as in 5.2 (**Figure 5–2**) may be used for further calculations. The pressure  $P_{elastic}$  leads to elastic displacements  $|u_C|$  and  $|u_{St}|$ , relative to the position of the shrunk concrete ring. The sum of these displacements is equal to the free displacement  $|u_{free}|$  (Equation 6–4).

$$|u_{free}| = |u_C| + |u_{St}|$$
 Equation 6–4

The displacements  $|u_{St}|$  and  $|u_C|$  due to  $P_{elastic}$  can be calculated using Equations 6–5 and 6–6, respectively [Hos 04]. The subscript SH specifies shrinkage as source of the interface pressure. Furthermore, the Poisson's ratio  $v_C$  is assumed to be constant.

$$|u_{St}| = P_{elastic,SH} \cdot \frac{R_{St,o}^2 \cdot \left[ (1 + v_{St}) \cdot R_{St,i}^2 + (1 - v_{St}) \cdot R_{St,o}^2 \right]}{E_{St} \cdot R_{St,o} \cdot (R_{St,o}^2 - R_{St,i}^2)}$$
Equation 6–5

$$|u_{C}| = P_{elastic,SH} \cdot \frac{R_{C,i}^{2} \cdot \left[ (1 + v_{C}) \cdot R_{C,o}^{2} + (1 - v_{C}) \cdot R_{C,i}^{2} \right]}{E_{C} \cdot R_{C,i} \cdot (R_{C,o}^{2} - R_{C,i}^{2})}$$
Equation 6–6

Equations 6–3 to 6–6 lead to Equation 6–7 for computing the increase of elastic interface pressure due to an increase of free shrinkage [Hos 04].

$$\Delta P_{elastic,SH} = -\frac{\Delta \varepsilon_{SH} \cdot E_{C}}{\frac{E_{C} \cdot ((1 + v_{St}) \cdot R_{St,i}^{2} + (1 - v_{St}) \cdot R_{St,o}^{2})}{E_{S} \cdot (R_{St,o}^{2} - R_{St,i}^{2})} + \frac{((1 - v_{C}) \cdot R_{St,o}^{2} + (1 + v_{C}) \cdot R_{C,o}^{2})}{(R_{C,o}^{2} - R_{St,o}^{2})}$$

Equation 6–7

Note that the contraction of the concrete ring and the resultant compression of the steel ring (negative signs) lead to an increase of the pressure and induce tensile hoop stresses (positive sign).

Then, the degree of restraint  $\psi$  (with  $\psi \le 1$ ) for time intervals i is given by Equation 6–8 (cf. 5.2). The time intervals are due to the variable modulus of elasticity [Hos 04].

$$\psi(t_i) = 1 - \frac{|u_{St}(t_i)|}{|u_{free}(t_i)|}$$
 Equation 6–8

The maximum theoretical-elastic stress due to shrinkage results from Equation 6–9 (cf. Equation 6–2).

$$\sigma_{\theta,\text{max},\textit{elastic}} = P_{\textit{elastic},\textit{SH}} \cdot \frac{R_{C,i}^2 + R_{C,o}^2}{R_{C,o}^2 - R_{C,i}^2}$$
 Equation 6–9

The actual interface pressure due to shrinkage at any point of time can be computed from the measured steel ring strains, the modulus of elasticity of the steel and the geometry of the rings (Equation 6–10) [Hos 04].

$$P_{actual,SH}(t) = -\varepsilon_{St,SH}(t) \cdot E_{St} \cdot \frac{(R_{St,o}^2 - R_{St,i}^2)}{2 \cdot R_{St,o}^2}$$
Equation 6–10

Then the maximum actual hoop stress can be calculated with Equation 6–11.

$$\sigma_{\theta, \max, actual} = P_{actual, SH} \cdot \frac{R_{C,i}^2 + R_{C,o}^2}{R_{C,o}^2 - R_{C,i}^2}$$
 Equation 6–11

Furthermore, a creep factor can be determined. Usually creep is investigated by means of uniaxial tests at constant stress (cf. 5.5, 6). This is not feasible with conventional restrained ring tests. However, the influence of time-dependent deformations within a time interval i can be quantified by comparing the change of actual and theoretical-elastic hoop stresses [Hos 04]. Equation 6–12 may be used for that purpose (cf. 6.4.5.3).

$$\varphi(t_i) = 1 - \frac{\Delta \sigma_{\theta, \text{max}, actual}(t_i)}{\Delta \sigma_{\theta, \text{max}, elastic}(t_i)}$$
Equation 6–12

### 6.3.2.3 Influence of temperature changes in restrained ring tests

Due to different coefficients of thermal expansion (CTE) of steel and concrete, a temperature change is associated with a change of stress in the restrained ring test. If, for instance, the CTE of the steel is higher, an increase of temperature leads to an increase of tensile hoop stress in the concrete ring since the steel ring expands more than the concrete ring does; the hoop stress would decrease if the temperature decreased. If the CTE of the concrete is higher than that of the steel, the effects of temperature changes are vice versa. Thus, changes of temperature irreversibly influence stress history. This effect in principle can be reduced by choosing a material for the inner ring with a CTE similar to that of concrete. However, the CTE of concrete undergoes notable changes at early age. Therefore, the thermal stress component cannot be completely avoided unless the temperature is constant. Commonly, the inner ring is made of stainless steel with a CTE of approximately  $16~\mu m/m/K$ . The CTE of hardened concrete typically is lower.

The influence of changing temperatures was mostly neglected or disregarded in the past. Restrained ring tests on drying shrinkage usually begin at the age of 24 h or later. At that age, temperature in specimens usually has decreased to room temperature. The peak of heat release due to hydration typically occurs in the first 24 h. In contrast to drying shrinkage, this is the most relevant period for the autogenous shrinkage. Kovler et al. investigated the effect of thermal expansion in a ring test with solid core. However, the given stress solution neglected creep [Kov 93]. This would not be appropriate for the very early age.

For a solution, consider that only the difference between the thermal strains of both rings is relevant. If the coefficients of thermal expansion of the rings were identical, uniform temperature changes would not alter the stress. Therefore the subscript  $\Delta\alpha$  is used in this context. The stress-effective free radial displacement at the common surface due to changing temperatures within a time-interval  $t_i$  is the product of the outer radius of the steel ring, the temperature change and the difference of the coefficients of thermal expansion (Equations 6–13 and 6–14, cf. Equation 6–4). The temperature is supposed to be identical allover the rings at any time.

$$\Delta u_{\Delta \alpha_i free}(t_i) = R_{St,\alpha} \cdot \Delta T(t_i) \cdot \Delta \alpha_T(t_i)$$
 Equation 6–13

$$\Delta \alpha_T(t_i) = \alpha_{TC}(t_i) - \alpha_{TSt}$$
 Equation 6–14

 $\alpha_{T,C}$ : coefficient of thermal expansion of concrete  $\alpha_{T,St}$ : coefficient of thermal expansion of steel

 $\Delta T$ : temperature change

 $t_i$ : time-interval

The deformations due to changes of temperature and due to shrinkage can be assumed to be restrained to the same degree. Hence, the stress-effective free radial displacement due to temperature change as per Equation 6–13, under the temporary assumption of elastic behavior, in the restrained ring test would split into radial displacements of steel ring and concrete ring according to the inverse ratio of their stiffness (cf. **Figure 5–2**). Applying Equation 5–1 to Equation 6–13 yields Equation 6–15 for the elastic radial displacement of the steel ring due to tem-

perature changes. Division by the outer steel ring radius  $R_{St,o}$  yields the steel ring strain due to temperature, still assuming elastic behavior, i.e. neglecting creep (Equation 6–16).

$$\Delta u_{St.elastic,\Delta\alpha}(t_i) = R_{St.o} \cdot \Delta T(t_i) \cdot \Delta \alpha_T(t_i) \cdot (1 - \psi(t_i))$$
 Equation 6–15

$$\Delta \varepsilon_{St \, elastic \, \Delta \alpha}(t_i) = \Delta T(t_i) \cdot \Delta \alpha_T(t_i) \cdot (1 - \psi(t_i))$$
 Equation 6–16

Then, in order to calculate the creep factor under non-isothermal conditions, Equation 6–7 first is modified to Equation 6–17, considering the thermal instead of the shrinkage strain.

$$\Delta P_{elastic,\Delta\alpha} = -\frac{\Delta T(t_i) \cdot \Delta \alpha_T(t_i) \cdot E_C}{\frac{E_C \cdot ((1 + \nu_{St}) \cdot R_{St,i}^2 + (1 - \nu_{St}) \cdot R_{St,o}^2)}{E_S \cdot (R_{St,o}^2 - R_{St,i}^2)} + \frac{((1 - \nu_C) \cdot R_{St,o}^2 + (1 + \nu_C) \cdot R_{C,o}^2)}{(R_{C,o}^2 - R_{St,o}^2)}$$

Equation 6–17

The change of theoretical-elastic stress due to temperature results from Equation 6–18. This stress increment is needed for the recalculation of the creep factor (Equation 6–19).

$$\Delta\sigma_{\theta,\max,elastic,\Delta\alpha} = \Delta P_{elastic,\Delta\alpha} \cdot \frac{R_{C,i}^2 + R_{C,o}^2}{R_{C,o}^2 - R_{C,i}^2}$$
 Equation 6–18

$$\varphi(t_{i}) = 1 - \frac{\Delta \sigma_{\theta, \max, actual, \Delta \alpha}(t_{i})}{\Delta \sigma_{\theta, \max, elastic, SH}(t_{i}) + \Delta \sigma_{\theta, \max, elastic, \Delta \alpha}(t_{i})}$$
Equation 6–19

Now the elastic steel ring strains as per Equation 6–16 can be turned into actual strains by considering creep as per Equation 6–20. The larger the creep factor is, the lower is the actual strain of the steel ring. The age-dependency of all parameters needs to be considered by evaluating sufficiently small time intervals (cf. 6.4.11).

$$\varepsilon_{St,actual,\Delta\alpha}(t) = \sum_{i} \Delta T(t_i) \cdot \Delta \alpha_T(t_i) \cdot (1 - \psi(t_i)) \cdot (1 - \varphi(t_i))$$
 Equation 6–20

Finally, to compensate for the thermal stress component in restrained ring tests under non-isothermal conditions, the steel ring strain as per Equation 6–20 can be subtracted from the measured overall strain to calculate the hoop stress due to shrinkage alone (Equation 6–21).

$$\sigma_{actual,\Delta\alpha=0}(t) = -\left[\varepsilon_{St,measured}(t) - \varepsilon_{St,actual,\Delta T}(t)\right] \cdot E_{St} \cdot \frac{R_{St,o}^2 - R_{St,i}^2}{2 \cdot R_{St,o}^2} \cdot \frac{R_{St,o}^2 + R_{C,o}^2}{R_{C,o}^2 - R_{St,o}^2}$$

Equation 6–21

Notably, Equations 6–20 and 6–21 yield approximate results. With very large thermal stress components they may not be applicable. The prevailing stress level with and without thermal stress component is different. The creep behavior depends on the stress level and presumably on the stress history as well (cf. 5.5). This presently cannot be accounted for. Furthermore, the creep factor may be difficult to determine, especially if the elastic stress in the time-interval observed does hardly change (6.4.5.3, 6.4.11).

# 6.3.3 Use of temperature changes for the investigation of creep and relaxation

As was outlined above, the creep and stress relaxation behavior of concrete under quasiisothermal conditions can be quantified by comparing the actual to the theoretical-elastic stress (Equation 6–12). The latter can be calculated from the modulus of elasticity and the free shrinkage (Equation 6–9). Furthermore, in the previous section the influence of changing temperatures on the actual and the theoretical-elastic stress was described. If the coefficient of thermal expansion of the steel is higher than that of the concrete - as it is usually the case with stainless steel and hardening concrete -, a temperature increase leads to a stress increase. This effect can be used to systematically investigate the influence of time-dependent deformations.

To this end the temperature is increased at a given point of time  $t_i$  (cf. **Figure 6–10**). The resultant increase of theoretical-elastic stress until the point of time  $t_{i+1}$  can be calculated by Equation 6–18, whereas the actual stress results from the steel ring strains. The creep factor for the time period from  $t_i$  to  $t_{i+1}$  can be calculated by Equation 6–19. Different steps of temperature increase  $\Delta T$  lead to different stress levels (**Figure 6–10**). The evaluation of a test series yields a matrix of creep factors (Equation 6–22) which then may be used to describe the effect of creep and relaxation on stresses.

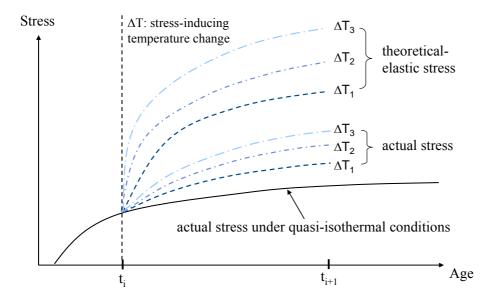


Figure 6–10 Scheme for investigating stress relaxation in restrained ring tests through temperature changes

$$\varphi(t_{i}, \sigma_{\text{max}} / f_{ct}) = \begin{bmatrix} \dots & \dots & \dots \\ \dots & 1 - \frac{\sigma_{actual}(t_{i}, \Delta T_{i}) - \sigma_{actual}(t_{i+1}, \Delta T_{i})}{\sigma_{elast}(t_{i}, \Delta T_{i}) - \sigma_{elast}(t_{i+1}, \Delta T_{i})} & \dots \\ \dots & \dots & \dots \end{bmatrix}$$

It is noted that the stress level will not be constant. Also, with longer test intervals the influence of temperature on hydration has to be accounted for. As an alternative to temperature changes, the stress can be increased by drying shrinkage. The ring needs to be unsealed and exposed to various relative humidities (cf. 6.4.6.4). The increase of theoretical-elastic stress in this case is calculated with Equation 6–9, based on the drying shrinkage measured separately.

Since this method has not been systematically applied yet, its potential and limitations are difficult to specify. It will hardly allow setting up compliance functions (5.5). However, it may be particularly suitable for determining the residual stress capacity and for analyzing the behavior at stress levels close to cracking. Conventional tensile creep tests are extremely difficult at very early age and very high stress levels. The advantages that the restrained ring test has in that regard were described above (5.5). Furthermore, clear increases of theoretical elastic stress can be provided for rather easily. This is essential to achieve reliable results (cf. 6.4.11).

### 6.4 Own investigations with the restrained ring test

#### **6.4.1** Setup

The setup used for the conducted restrained ring tests is described. Since the maximum particle size of the investigated concretes was 0.5 mm, the concrete ring could be particularly small. It had a nominal width and height of 24 mm and 25 mm, respectively. Height and width of the steel ring was 25 mm and 15.7 mm, respectively. It was made of austenitic stainless steel (Type 1.4301) and had an outer diameter of 190 mm. The steel was found to have a coefficient of thermal expansion of  $\alpha_T = 16.5 \cdot 10^{-6}$  /K. The modulus of elasticity was assumed to be 200 GPa. The mold consisted of a base plate and a pipe section, both made of plastic. Friction between steel and mold as well as between steel and concrete was minimized by grease. The steel ring was centered on the base plate and fixed by screws. The film which was used to seal the concrete after the casting, was fixed by double-faced adhesive tape and additional tape to the circumference of the mold (**Figure 6–1**, **Figure 6–2**). After sealing, the screws were unscrewed.

To obtain the mean steel ring deformation, three strain gauges with a nominal resistance of 120 Ohm and a length of 6 mm were glued to the inner surface of each steel ring, evenly distributed over the ring's inner circumference and at mid height (**Figure 6–3**). Change of resistance was directly measured by a digital multimeter (Agilent 34970A). The proportional strain could be measured with a resolution better than  $0.01 \,\mu\text{m/m}$ . A four wire configuration was used to cancel out the lead resistance. Due to the high sensitivity and resolution of the digital multimeter, precise data were obtained without the need of a Wheatstone bridge.

Temperature was measured by so called resistance temperature detectors one of which was glued to the inner surface of each steel ring (Figure 6–3, right). The detectors were made of platinum and had a nominal resistance of 100 Ohm. The first meter of the lead was in two

wire configuration and the rest in four wire configuration. The aforementioned data acquisition unit was used for the resistance temperature detectors as well. In preliminary tests under quasi-isothermal conditions, temperature of concrete and steel ring were found to be in very good agreement during the whole test period. Differences were below 0.5 K. Thus, temperatures of steel and concrete ring were assumed to be identical throughout the tests. Furthermore, a uniform distribution of temperature was assumed.



Figure 6–1 Setup of restrained ring test, before (left) and after casting (right)

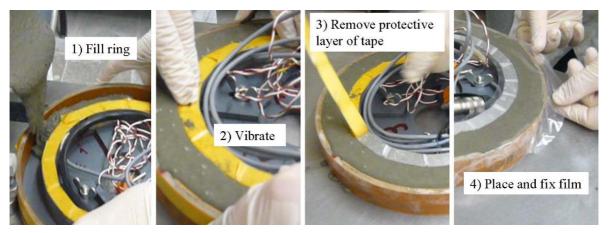


Figure 6–2 Casting, vibrating and sealing of the ring



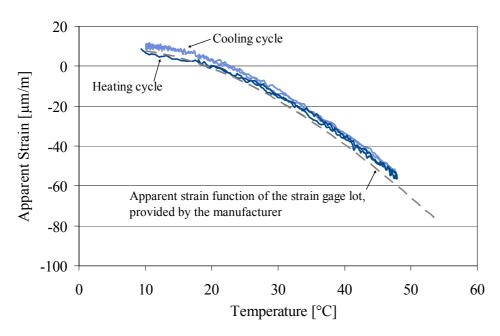
**Figure 6–3** Left: glued strain gauge, strain relief. Right: strain gauge and resistance temperature detectors, wired and protected

Eight rings were available. The measuring cabinet is shown in the Annex (Figure 9–12). Quasi-isothermal tests with sealed rings were conducted in a climate room at 20 °C and 65 % relative humidity. In tests with sealed rings the mean weight loss in the most relevant first 24 h amounted to approximately 0.15 % with respect to the mass of water contained in the specimen. The loss of moisture was sufficiently low to consider the specimens entirely sealed.

A small airproof container was used for specific tests with rings drying from top and bottom A photo of the container can be found in the Annex (Figure 9–12, right). During these tests the humidity in the container was rapidly decreased by means of dry lithium chloride. Additional ring tests with sealed rings under non-isothermal conditions were carried out in an oven. In this case, the relative humidity was kept at approximately 80 % to avoid drying shrinkage.

# **6.4.2** Compensation of disturbing temperature effects

Strain gauges were to measure only those deformations which were imposed on the steel ring by the concrete ring. A potential direct influence of temperature changes on the measured resistance needed to be suppressed. Two measures were taken. Firstly, strain gauges were adapted to the coefficient of thermal expansion of the steel ( $\alpha_T = 16.5 \cdot 10^{-6} / \text{K}$ ). Thus, linear deformations of the steel due to variable temperature were largely excluded from the measurement. Secondly, residual non-linear changes of resistance were accounted for. So called 'apparent strains' occur at changing temperatures. They are due to the specific behavior of the different materials strain gauges are made of. The manufacturer of the strain gauges provided a polynomial function for the temperature-dependent apparent strain of each strain gauge lot. Since the temperature increase due to heat of hydration under quasi-isothermal conditions was in the range of only 1 to 2 K, this standard function was sufficient to compensate for apparent strains due to temperature changes.



**Figure 6–4** Precision of strain gauges glued to steel rings at variable temperature: apparent strain vs. temperature of strain gauge

Mainly to further qualify the test setup for non-isothermal tests, the apparent strain was individually determined for each glued strain gauge by simply measuring the resistance in a temperature range of 10 °C to 50 °C. To avoid a non-uniform temperature distribution within the steel ring, temperature was increased slowly (max. approx. 20 K/h). The resistance temperature detectors were additionally protected from the influence of the air temperature. An exemplary result is shown in **Figure 6–4**. The individual heating and cooling cycles yielded a good agreement of the apparent strains. A small hysteresis was noticeable in the range of approximately 10 to 25 °C. The measured strain curve was very similar to the compensation function.

# 6.4.3 Repeatability

The precision of the restrained ring test under repeatability conditions was determined. A high-strength concrete composition with comparably low cracking susceptibility (c =  $600 \text{ kg/m}^3$ , w/c = 0.3, max. grain size: 0.5 mm) was used for this investigation in order to minimize the potentially irregular influence of micro-cracks. Concrete was mixed four times in a row. Two rings were produced from each mix. All eight rings remained sealed during the first 24 h and then were unsealed on top and bottom and stored at 20 °C and 65 % RH. The resultant steel ring strains are shown in Figure 6-5, the standard deviation and the coefficient of variation in Figure 6-6. Several points can be noticed. First, strains were close to zero before stresses started to develop at the age of approx. 11 h, indicating the reliability of the measuring system. Second, due to the desired high degree of restraint the maximum strain within the first 24 h was approx. 15 μm/m, compared to 50 μm/m recommended for good precision in [Moo 06]. Third, due to the very low strains the coefficient of variation around the onset of stress development was very high and went down to more acceptable values only some hours later. Notably, the standard deviation between 11 and 24 h was as low as 0.8 to 1.5 µm/m; significantly lower values could be achieved only with a much higher effort. In any case, a reasonably high number of ring tests is required for investigations at very early age. The observed variation of strains mainly can be attributed to the variation of batches, of temperature, of concrete ring dimensions as well as of the strain gauges and their assembly within the setup.

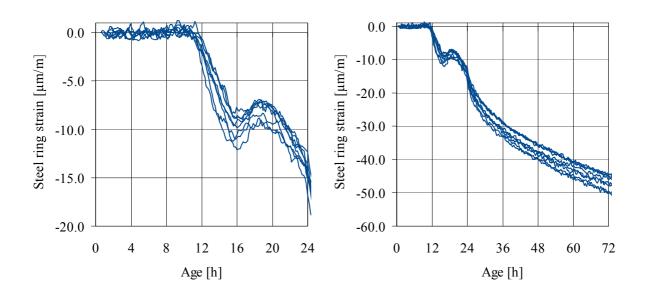


Figure 6–5 Eight restrained ring tests under repeatability conditions with a high-strength concrete ( $c = 600 \text{ kg/m}^3$ , w/c = 0.3, max. grain size of 0.5 mm): steel ring strain vs. age, left: 0-24 h, right: 0-72 h

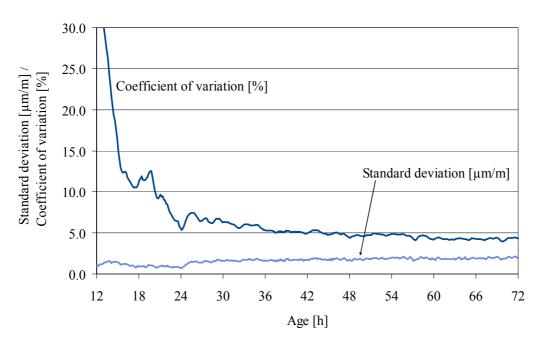


Figure 6–6 Standard deviation and coefficient of variation for eight restrained ring tests under repeatability conditions with a high-strength concrete ( $c = 600 \text{ kg/m}^3$ , w/c = 0.3, max. grain size of 0.5 mm)

### **6.4.4** Measured steel ring strains

The main investigations with restrained ring tests involved the reference concrete composition (1A), the concrete composition with a large amount of ground-granulated blast furnace slag (1A BFS) and the two concrete compositions with superabsorbent polymers and shrinkage-reducing admixture (1A SAP, 1A SRA). Six rings were produced from three mixes of each concrete composition, except for concrete 1A BFS which was mixed only twice, hence four rings were produced in this case. Measurements were continued until the age of 72 h. The single results of all tests can be found in the Annex (Figure 9–13 to Figure 9–16). Here, the evaluation is limited to the first 24 h which turned out to be most relevant. The mean steel ring strains until the age of 24 h are shown in Figure 6–7.

Steel ring strains were close to zero until the age of 8 h. This indicates the reliability of the data (cf. 6.4.2). The steel rings started to deform at approximately 8, 9, 12 and 15 h for concrete 1A, 1A SAP, 1A SRA and 1A BFS, respectively. Strains then constantly increased with concrete 1A BFS. With concrete 1A SAP this was the case until 16 h, thereafter the strain was almost constant. By contrast, with concretes 1A and 1A SRA the strains reached a maximum at approx. 14 h (14  $\mu$ m) and 19 h (6  $\mu$ m/m), respectively, and significantly decreased afterwards, obviously due to time-dependent deformations. At 24 h the strains were in the range of 4 to 10  $\mu$ m/m for all concretes. The reference concrete 1A yielded the highest strains. No cracks were observed on any of the rings.

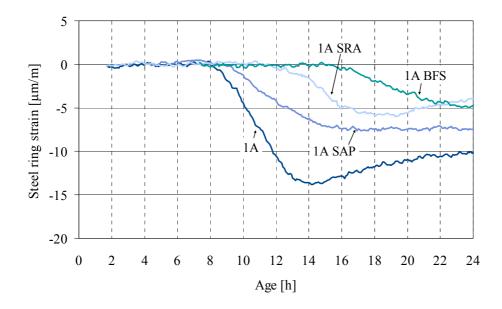


Figure 6–7 Steel ring strain up to the age of 24 h

# 6.4.5 Simple stress analysis

### 6.4.5.1 Maximum hoop stress

The maximum restraint stress, i.e. the hoop stress at  $r = R_{C,i}$  as per Equation 6–11, is shown in **Figure 6–8**. It is reminded that this stress is proportional to the steel ring strain (**Figure 6–7**). For the conversion of strains to stresses, the Poisson's ratio of concrete was assumed to be 0.18. Concrete 1A yielded the highest stress at 14 h (1.87 MPa); at 24 h the stress amounted to 1.39 MPa, corresponding to a decrease of approx. 25 %. Concrete 1A SAP reached a stress of 1.0 MPa at 16 h which did not significantly change anymore afterwards. The lowest stress at the age of 24 h was found for concrete 1A SRA (0.54 MPa); the maximum in this case occurred at approx. 19 h (0.8 MPa). Concrete 1A BFS yielded a tensile stress of 0.67 MPa at 24 h and was the only concrete with a positive stress rate at 24 h.

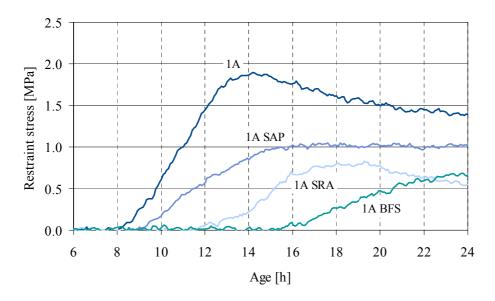


Figure 6–8 Restrained ring tests: stress due to restrained autogenous shrinkage in ultra-high strength concrete

Additional tests with an alternative superplasticizer were carried out (SP2, cf. Table 3–1). As is shown in Figure 6–9, the superplasticizer significantly influenced the development of restraint stresses. Although concrete 1A SP2 contained more superplasticizer than the reference concrete 1A, hydration was much less retarded. Stresses were higher and increased more rapidly. Again, the maximum stress was followed by a characteristic decrease.

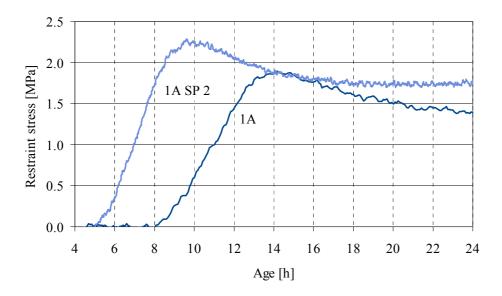


Figure 6–9 Influence of different superplasticizers on restraint stress in ultra-high strength concrete

### 6.4.5.2 Degree of restraint

As was illustrated in section 5.2, the degree of restraint describes how two elastic bodies are restrained by each other when a force acts in between them. The resultant deformations are proportional to the inverse ratio of the stiffness of the two bodies (**Figure 5–1**). For the restrained ring tests the degree of restraint was calculated as ratio of prevented to free radial displacement of the concrete ring in the restrained ring test (cf. Equation 5–1).

The static modulus of elasticity which is required to compute the prevented deformations (cf. Equations 6–4 to 6–8), evolves with time (cf. 3.5.3). The fitting procedure that was required to obtain the development of the modulus of elasticity was described in section 3.5.4. To compute the degree of restraint until the age of 24 h by means of Equation 6–8, intervals of 1 h were considered.

The development of the degree of restraint is shown in Figure 6–10. Note that the dashed lines represent the range of extrapolation of the modulus of elasticity. Restraint was complete until the onset of steel ring deformations, i.e. the degree of restraint was 1. With concretes 1A and 1A SAP the degree of restraint decreased to approximately 0.84 at the age of 24 h. Reflecting the slower development of the modulus of elasticity, the degree of restraint with concretes 1A SRA and 1A BFS decreased later and amounted to approximately 0.86 and 0.93 at 24 h, respectively.

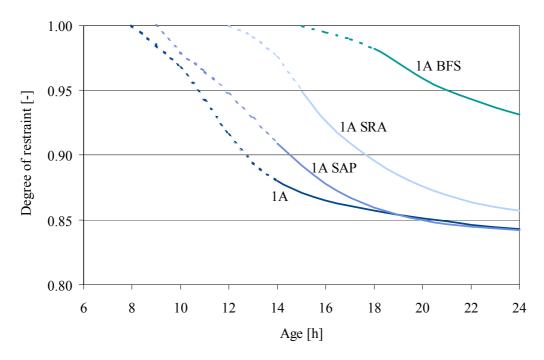


Figure 6–10 Degree of restraint in restrained ring tests (dashed: extrapolation of modulus of elasticity)

Notably, the fact that the degree of restraint differed, does not mean that the further results obtained from the four concretes were not comparable. The same stress levels that resulted from the respective degree of restraint in the restrained ring tests would have occurred in praxis if shrinkage was restrained not by the steel ring but by an object with the steel ring's stiffness. On the other hand, the degree of restraint influences the stress history, i.e. the stress level over time. The maximum autogenous shrinkage cracking propensity of the four concretes possibly would have been different and would have occurred at another point of time if the stress history had been different. To investigate this, different ring geometries with different degrees of restraint would be required.

### 6.4.5.3 Theoretical-elastic stress and creep factor

The theoretical-elastic stress is important for the evaluation of restrained ring tests because it allows quantifying the influence of stress relaxation (cf. 5.4 and 6.3.2.2). It essentially depends on the modulus of elasticity and the autogenous shrinkage. The following figure shows the development of the theoretical-elastic stress as per Equation 6–9 (Figure 6–11). If concrete behaved entirely elastically, autogenous shrinkage of concrete 1A would have yielded a stress of approximately 5.8 MPa at the age of 24 h. As a consequence of the lower modulus of elasticity and/or lower autogenous shrinkage, the other concretes led to much lower theoretical-elastic stresses.

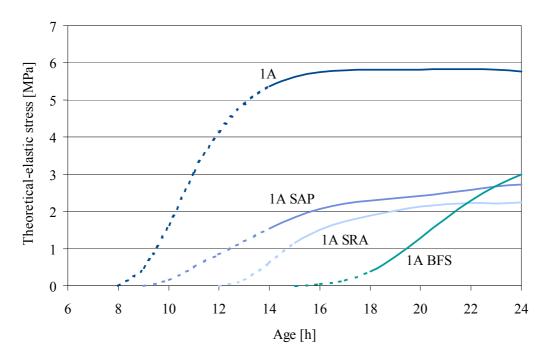
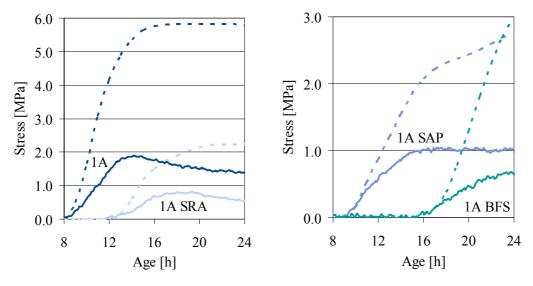


Figure 6–11 Theoretical-elastic stress up to an age of 24 h (dashed: extrapolated)

Theoretical-elastic and actual stress are compared in **Figure 6–12**; for the sake of clarity the illustration was split into two graphs. Here the dashed lines represent the theoretical-elastic stress. Note the different scaling of the stress axis. The data shown in **Figure 6–12** can be used as basis for the quantification of creep and relaxation. A creep factor can be calculated by means of the aforementioned equation (6–12):

$$\varphi(t_i) = 1 - \frac{\Delta \sigma_{\theta, \max, actual}(t_i)}{\Delta \sigma_{\theta, \max, elastic}(t_i)}$$



**Figure 6–12** Theoretical-elastic stress (dashed) and actual stress (solid): 1A and 1A SRA (left), 1A SAP and 1A BFS (right)

The result depends on the time interval that is considered. **Figure 6–13** shows the creep factor for the entire period observed (0 to 24 h), calculated from theoretical-elastic and actual stress at the age of 24 h. Between 64 and 78 % of the stresses which would have been expected with purely elastic behavior were eliminated by creep and relaxation. The creep factor of concretes 1A, 1A SRA and 1A BFS was very similar. The slightly different behavior of concrete 1A SAP is addressed in chapter 6.4.9.

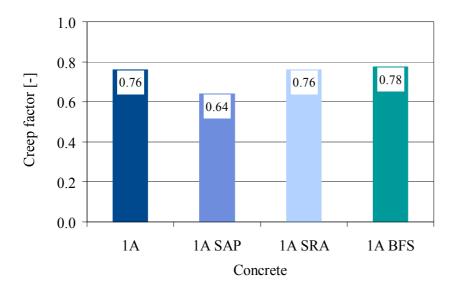


Figure 6–13 Creep factor up to the age of 24 h

The creep factors were generally lower than the 0.9 that were observed in [Sch 07] and [Sch 02] for ultra-high strength concrete. However, in that investigation the autogenous shrinkage was measured with a test method with unknown precision and the autogenous shrinkage was higher than in any other previous investigation of ultra-high strength concrete, cf. Figure 2–5. An overestimation of the autogenous shrinkage would lead to an overestimation of the influence of relaxation and creep. Of course there are other potential sources of error as well, for instance the extrapolation of the very early age values of the modulus of elasticity. The concrete composition was different as well. Insofar one can speak of a reasonable agreement of the creep factors observed for the first 24 h. A creep factor for long periods with rapidly changing concrete properties is a very rough description of the viscoelastic behavior, anyway. To possibly obtain a more detailed picture of the creep behavior, shorter time-intervals need to be considered (cf. 6.4.11).

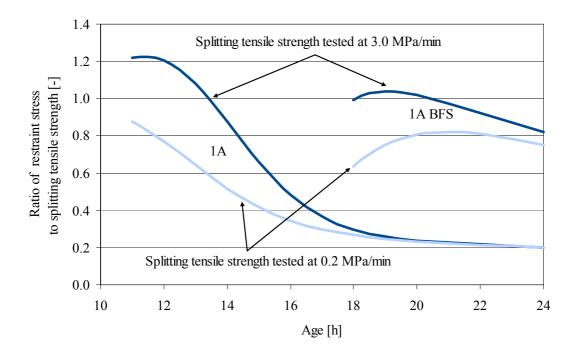
# 6.4.6 Autogenous shrinkage cracking propensity - further analysis

#### 6.4.6.1 Introduction

To compute the autogenous shrinkage cracking propensity as function of time, the stresses due to restrained autogenous shrinkage can be related to the tensile strength development (cf. 5.6). This approach was employed here, too. To achieve reliable results, the two input parameters of the stress-strength ratio were comprehensively analyzed. It is noted that for the following computations the stress values shown in **Figure 6–8** were fitted to best-fit functions. The data used for the stress analysis can be found in the Annex (**Table 9–5** to **Table 9–8**).

### 6.4.6.2 Ratio of restraint stress to splitting tensile strength

If the cracking propensity is quantified as ratio of stress to strength, cracking is to be expected at ratios close to 1, and ratios higher than 1 should not occur. However, the ratio of observed restraint stress to the splitting tensile strength measured at a rate of stress increase as per EN 12390-6 (3.0 MPa/min) exceeded 1, both with concrete 1A and concrete 1A BFS (Figure 6–14). Cracks would have occurred if this was to correspond to the cracking propensity. With the splitting tensile strength obtained at a much lower rate of stress increase (0.2 MPa/min), the ratios were significantly lower because the splitting tensile strength resulted higher (cf. 3.5.2).



**Figure 6–14** Ratio of stress due to restrained autogenous shrinkage to the splitting tensile strength measured at two different rates of stress increase, concretes 1A and 1A BFS

From experience with tests at later age it would have been expected that the splitting tensile strength increased with increasing rate of stress increase (3.5.2). No clear explanation can be given for the opposite tendency. Remarkably, the calculation of the stress-strength ratio yielded more reasonable values with the strength values obtained at the lower rate of stress increase. Possibly there are unknown effects at early and very early age that influence the results of splitting tension tests (cf. 5.6).

The ratio of restraint stress to splitting tensile strength of all concretes investigated is shown in **Figure 6–21**. In comparison to the reference concrete, the ratio was generally much lower with concrete 1A SAP as well as with concrete 1A SRA. In both cases the maximum ratio was in the range of 0.4, and the ratio at 24 h was roughly 0.1.

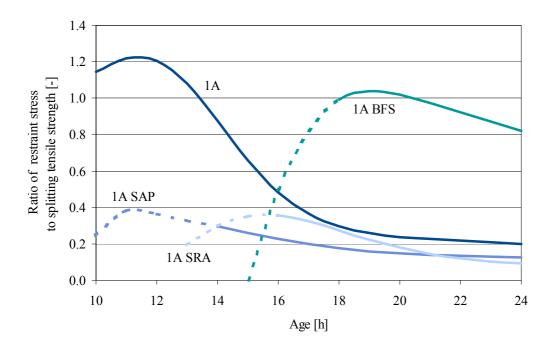
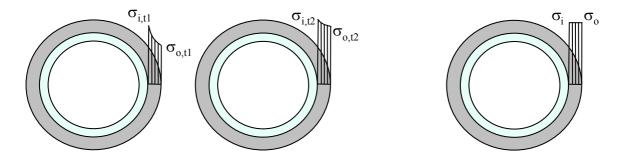


Figure 6–15 Ratio of restraint stress to splitting tensile strength, concretes 1A, 1A SAP, 1A SRA and 1A BFS

#### **6.4.6.3** Stress redistribution

As described above, the steel ring strains measured in restrained ring tests usually are converted to hoop stresses under the assumption of an elastic non-linear stress distribution along the radial direction of the concrete cross section (cf. 6.3.2). According to the elastic solution the hoop stresses are highest at the inner and lowest at the outer radius of the concrete ring. The hoop stress at an arbitrary radius r results from the Lamé line (cf. **Figure 6–9**).

However, in this and in other studies a strong influence of stress relaxation on stresses due to restrained autogenous shrinkage was observed (cf. Figure 6–13). In such cases it is not reasonable to assume that the hoop stresses are distributed along the concrete ring radius like in a purely elastic material. Instead, it must be assumed that a hoop stress redistribution occurs during the period of high stress relaxation capacity. Larger deformations caused by higher stress-strength ratios at the inner concrete radius lead to an outward stress redistribution (Figure 6–16, left).

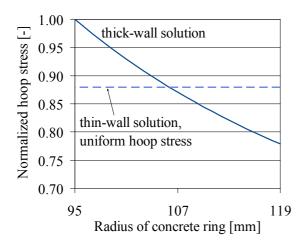


**Figure 6–16** Stress redistribution in restrained ring tests. Left: hoop stresses between two points in time increase more at the outer than at the inner radius  $(\Delta \sigma_{o,tl \to t2} > \Delta \sigma_{i,tl \to t2})$ ; right: stress equilibrium as the theoretical state of maximum stress redistribution  $(\sigma_i = \sigma_o)$ .

Without additional measures it is not possible to prove or to quantify the redistribution of hoop stress. To estimate the effect, its theoretical maximum can be calculated by assuming that stress redistribution has resulted in a completely uniform hoop stress (**Figure 6–16**, right). The resultant hoop stress can be calculated by the solution for thin walls that neglects hoop stress differences in tube sections (Equation 6–23):

$$\sigma_{\theta} = P \cdot \frac{R_i}{R_o - R_i}$$
 Equation 6–23

The following figure shows the normalized hoop stress distribution for the utilized ring geometry, i.e. the hoop stress as a function of the radius and related to the maximum hoop stress (Figure 6–17). Under the conventional assumption that no radial stress redistribution takes place, the application of the elastic 'thick-wall' solution yields a hoop stress at the outer concrete radius of 78 % of the maximum hoop stress at the inner radius. In comparison, the constant hoop stress according to the 'thin-wall' solution would be 88 % of the maximum hoop stress as per the 'thick-wall' solution.



**Figure 6–17** Normalized hoop stress vs. radius of concrete ring, solid curve: elastic 'thick-wall 'solution, dashed curve: elastic 'thin-wall' solution as approximation of the potential of stress redistribution

Hence, the actual hoop stress at the inner radius of the concrete ring may be overestimated by up to 12 % if stress redistribution is disregarded. The actual redistribution in the own tests most probably remained lower than this theoretical value. However, stress redistribution in general can be assumed to be particularly pronounced at higher stress-strength ratios. Hence, the maximum autogenous shrinkage cracking propensity actually may be considerably overestimated.

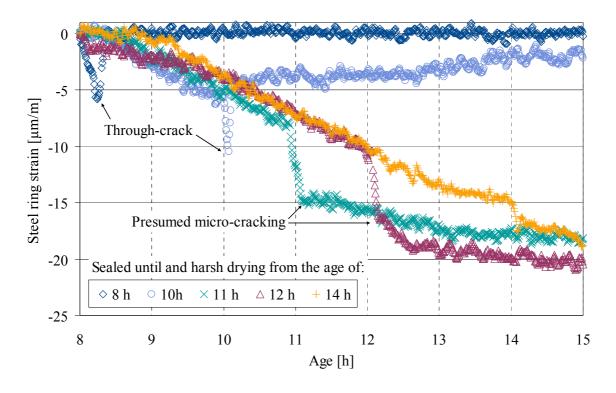
### 6.4.6.4 Modified ring tests at very low humidity

The cross section of the concrete ring can be assumed to be subjected to a roughly uniform uniaxial tensile stress. Hence, the uniaxial tensile strength would be the appropriate resistance quantity for the calculation of the autogenous shrinkage cracking propensity. However, uniax-

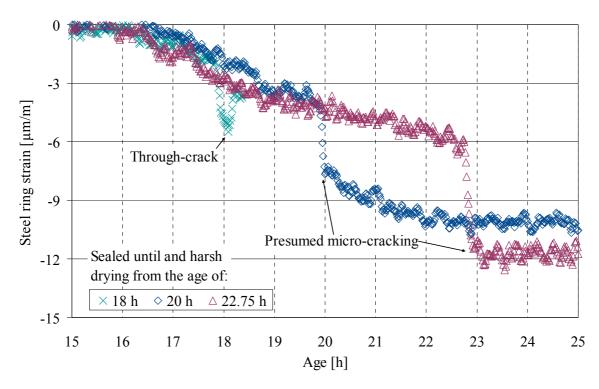
ial tensile strength tests at very early age would be very difficult and results possibly not reliable. Here, splitting tension tests were carried out instead (cf. 3.5.2). As the uniaxial tensile strength was not tested, the ratio of splitting to uniaxial tensile strength could not be determined. Other studies did not provide appropriate coefficients, either.

To examine the relation of the tested splitting tensile strength to the tensile strength in the restrained concrete rings, modified ring tests were carried out with concretes 1A and 1A BFS. First, rings remained sealed like in the main test series. Then, starting from different ages, ring by ring was allowed drying from top and bottom at very low relative humidity. To this purpose, the rings were placed in an airtight container, together with approximately 200 g of dry lithium chloride (cf. **Figure 9–12**). The lithium chloride decreased the relative humidity inside the ventilated container within 15 min to approximately 30 %. The temperature increased by 2 to 3 K due to the exothermic reaction of the lithium chloride, however this temperature increase can be disregarded in this context.

The drying led to a rapid increase of strains. With concrete 1A, macro-cracks occurred after few minutes when drying began at 8 h and 10 h (Figure 6–18). Later onsets of drying led to a rapid increase of strains, followed by a more (11 h) or less (12 h) abrupt decrease of the strain rate, probably indicating substantial micro-cracking. Drying from the age of 14 h led to a smaller peak of strains; possibly micro-cracks occurred, however, the strain rate afterwards was almost as high as before the peak. Similarly, with concrete 1A BFS, harsh drying from 18 h led to a through-crack, whereas micro-cracking is supposed to have occurred in the rings that were unsealed at 18 h and 22.75 h (Figure 6–19).



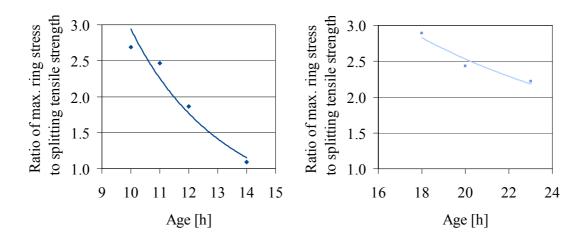
**Figure 6–18** Results of modified restrained ring tests with rings rapidly drying out from top and bottom from different ages, steel ring strain vs. age of concrete, concrete 1A



**Figure 6–19** Results of modified restrained ring tests with rings rapidly drying out from top and bottom from different ages, steel ring strain vs. age of concrete, concrete 1A BFS

Since the rings dried out from top and bottom but not from the circumference, the moisture profile in radial direction can be assumed to have remained uniform. Hence, the same equations apply that were used to calculate the maximum hoop stress due to restrained autogenous shrinkage. With concrete 1A, the stress at failure at the age of 8 h and 10 h was calculated from the respective maximum steel ring strains. In the case of harsh drying from 11, 12 and 14 h, not leading to complete failure, the stress which corresponded to the local strain peak was calculated. The same was done for the through crack that was observed at 18 h with concrete 1A BFS and the presumed micro-cracking at 20 and 23 h.

The obtained peak and failure stresses then were divided by the respective splitting tensile strength values (obtained at a rate of stress increase of 3.0 MPa/min, cf. 3.5.2). The resultant ratios of maximum hoop stress to splitting tensile strength as function of concrete age are shown in **Figure 6–20** for concrete 1A (left) and 1A BFS (right). With concrete 1A, the ratio strongly decreased from approximately 2.7 at 10 h to 1.1 at 14 h, with concrete 1A BFS from 2.9 at 18 h to 2.2 at 23 h. Non-linear trends were tentatively added to the single values. Obviously the splitting tensile strength needs to be corrected by a non-linear function to be comparable to the tensile strength in the rings at very early age.



**Figure 6–20** Ratio of failure/peak stress measured in modified restrained ring tests to splitting tensile strength, as a function of concrete age; single values and trend. Left: concrete 1A, right: concrete 1A BFS

### 6.4.6.5 Very early age effects and autogenous shrinkage cracking propensity

In the previous sections three potential 'very early age effects' were described: the influence of the rate of stress increase on the results of splitting tension tests (6.4.6.2), stress redistribution in restrained ring tests (6.4.6.3), and the fact that restraint stresses at failure were much higher than the splitting tensile strength (6.4.6.4). Hence, the computation of the autogenous shrinkage cracking propensity at very early age required a correction of the input parameters hoop stress and splitting tensile strength.

The splitting tensile strength was corrected on the basis of recently published findings of Malarics and Müller [Mal 10], (cf. also 5.6). In their comprehensive study the ratio of uniaxial to splitting tensile strength overproportionally increased with decreasing compressive strength (**Figure 5–7**). Based on experimental and numerical investigations, the authors suggested various power functions for the conversion of the two types of tensile strength. For cylindrical specimens (150 mm/300 mm) and concrete with crushed aggregate the following formula was proposed (Equation 6–24, cf. also **Figure 5–7**).

$$A = 2.64 \cdot f_{cm}^{-0.20}$$
 Equation 6–24

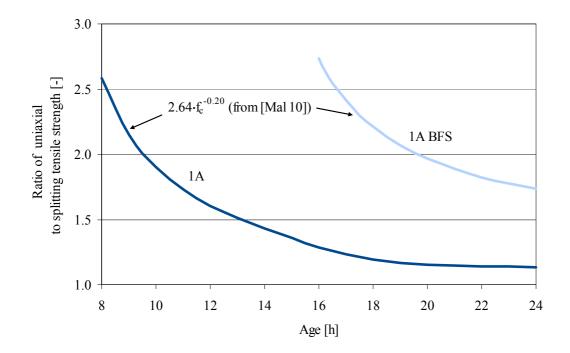
A: ratio of uniaxial to splitting tensile strength;  $f_{cm}$ : mean compressive strength

Three other formulas concerned drilled cores or concrete with gravel aggregate and were considered less suitable to represent cylinders made from fine-grained ultra-high strength concrete. Notably, specimens for splitting tension tests in the own investigation were considerably smaller and the test setup was different. However, there was no other conversion formula in literature that would have been as appropriate.

The input parameter in Equation 6–24 is the compressive strength which in the own study was not tested at very early age. Therefore a constant ratio of compressive strength to splitting tensile strength of 10 was assumed (cf. [fib 10]). This factor was applied to the function that was fitted to the results of splitting tensile strength obtained at a rate of stress increase of 3.0 MPa/min.

The lowest compressive strength of the concretes tested by Malarics and Müller was 20 MPa. Here, the calculated compressive strength partly was much lower. For instance, the compressive strength of concrete 1A increased from approximately 5 to 12 MPa between 10 and 12 h.

The development of the ratio A as per Equation 6–24 is exemplified with concrete 1A and 1A BFS (**Figure 6–21**). Both concretes yielded ratios A above 2.5 for the period of first restraint stress (1A: 8h, 1A BFS: 15 h, cf. **Figure 6–8**); the ratio decreased disproportionally with increasing age. For the calculation of the autogenous shrinkage cracking propensity the splitting tensile strength of all concretes was multiplied by the respective ratios A.



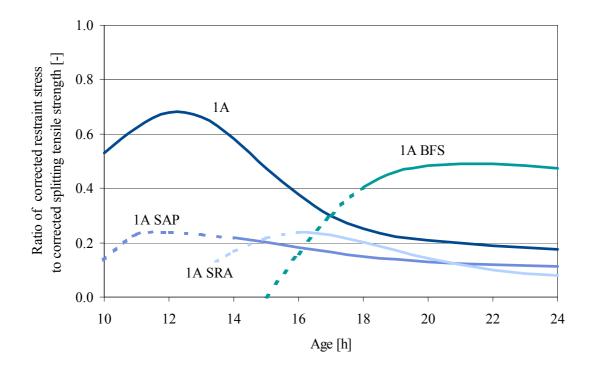
**Figure 6–21** Expected ratio of uniaxial to splitting tensile strength as a function of age, calculated from the 10fold splitting tensile strength by means of a conversion formula suggested in [Mal 10]

Furthermore, stress redistribution in restrained ring tests due to plastic deformations needed to be accounted for. For concrete 1A stress redistribution was assumed to be at its theoretical maximum until 11 h and nil from the age of 16 h (cf. 6.4.6.3). The assumed uniform stress distribution until 11 h was calculated by Equation 6–23 ('thin-wall' solution). Values between 11 and 16 h were interpolated; the maximum hoop stress after 16 h was not corrected. A more precise consideration was neither possible nor necessary. In comparison to the strong correction of the splitting tensile strength with factors exceeding 2, the influence of stress redistribution was low.

For concrete 1A BFS stress redistribution was assumed to be maximal at 17 h and nil at 24 h. Again, other values were interpolated. Stress redistribution was neglected with concretes 1A SAP and 1A SRA which led to much lower stress-strength ratios; here only the splitting tensile strength was corrected.

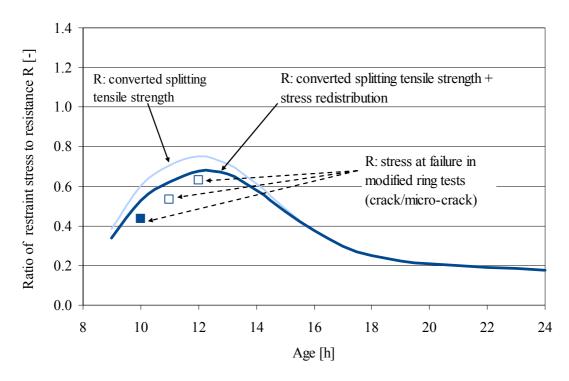
The resultant ratio of (corrected) restraint stress to (corrected) splitting tensile strength is shown in Figure 6–22. The maximum stress-strength ratio was approximately 0.68 and 0.49 for

concrete 1A and 1A BFS, respectively; this is coherent with the fact that no cracks were observed. In comparison to concrete 1A BFS, the stress-strength ratio of the reference concrete decreased much more rapidly after the maximum. The superabsorbent polymers and the shrinkage reducing admixture reduced the maximum ratio to approximately 0.24. At 24 h, the ratio was roughly 0.1. Notably, with all concretes the maximum ratio occurred few hours after solidification.

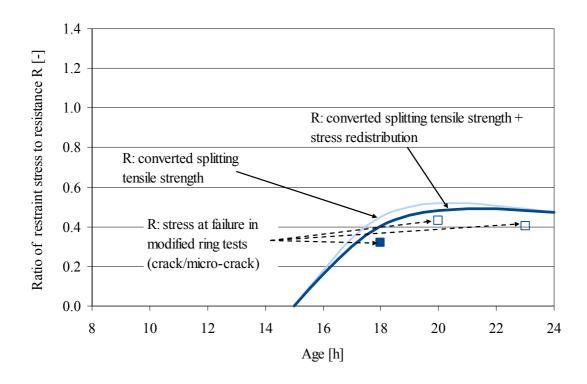


**Figure 6–22** Ratio of corrected restraint stress to corrected splitting tensile strength, concretes 1A, 1A SAP, 1A SRA and 1A BFS

To further examine the applicability of the stress-strength failure criterion and the validity of the assumptions that were required for the previous calculations, the curves as per Figure 6–22 were compared to the ratio of restraint stress to stress at failure in the modified restrained ring tests, i.e., the corrected splitting tensile strength as denominator in the stress-strength ratio was substituted by the failure stress (cf. Figure 6–18 and Figure 6–19). Since the stress at failure was tested only for selected ages, single points resulted from this. Filled points indicate through-cracks, others micro-cracking. To illustrate the rather little influence of stress redistribution, the curves were recalculated without its consideration (light curves).



**Figure 6–23** Ratio of stress due to restrained autogenous shrinkage to various measures of resistance R as function of age, concrete 1A



**Figure 6–24** Ratio of stress due to restrained autogenous shrinkage to various measures of resistance R as function of age, concrete 1A BFS

As is shown in Figure 6–23 and Figure 6–24 for concrete 1A and concrete 1A BFS, respectively, the curves are in good and partly very good agreement with the single points. This suggests that the modified ring tests were a suitable method for testing the ultimate strength of the con-

crete rings. The good agreement furthermore indicates that the required assumptions were appropriate and that the stress-strength failure criterion can be applied also at very early age. It can be assumed that the curves that were obtained by correcting hoop stress and splitting tensile strength give a realistic picture of the development of the autogenous shrinkage cracking propensity under quasi-isothermal conditions (Figure 6–22).

### **6.4.7** Thermal stress component

## 6.4.7.1 Quasi-isothermal tests

At changing temperatures the hoop stress in restrained ring tests is influenced by a thermal stress component (6.3.2.3). Due to the small cross section of the rings used in this investigation, temperatures of steel and concrete changed uniformly and temperature did not increase by more than 2 K in the quasi-isothermal tests. In order to examine to which extent this small temperature increase influenced results, the theoretical-elastic hoop stress due to temperature changes was calculated for concrete 1A (cf. 6.3.2.3). The maximum of approximately 0.06 MPa occurred at 14 h (Figure 6–25). The overall theoretical-elastic stress at that age was approximately 4.7 MPa (Figure 6–12). Thus, the thermal stress component corresponded to 1.3 % of the overall elastic stress and was therefore negligible in its effect on the actual stress.

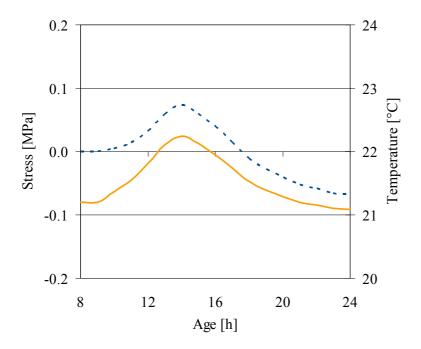


Figure 6–25 Theoretical-elastic stress (dashed) due to temperature changes (solid), concrete 1A

# 6.4.7.2 Tests under non-isothermal conditions

In section 6.3.2.3 a way to compensate for the thermal stress component in non-isothermal ring tests was suggested. The application of this method to very early age results is very difficult, partly because the concrete's coefficient of thermal expansion varies during that period. Its precise determination is very demanding (cf. 2.5.2 and 5.7). The required creep factor may be difficult to determine, too (cf. 6.3.2.2 and 6.4.11). The decreased precision of strain gauge measurements at changing temperatures and a non-uniform temperature increase throughout the ring system are other potential sources of error. Furthermore, the quality of the stress

analysis depends on the accurate measurement of the autogenous shrinkage and the elastic modulus under the same, non-isothermal conditions. As was mentioned above, there presently is no widely accepted method for measuring the autogenous shrinkage under non-isothermal conditions (cf. 2.3.3).

To examine the applicability of the suggested stress solution, the effect of temperature changes was investigated at a later age. By doing so, it was possible to largely overcome the aforementioned difficulties. Specific ring tests were carried out with reference concrete 1A between the age of 26 and 33 h. Hydration at that age had slowed down considerably. The autogenous shrinkage proceeded at high rates only until approx. 16 h and thereafter at very low rates close to zero (cf. 4.5). Here it was assumed that the autogenous shrinkage had come to a complete stop at 26 h. Likewise, the stress component due to autogenous shrinkage was taken as nil between 26 and 33 h. The CTE was assumed to be 11.2 μm/m/K (cf. 3.5.5).

Three rings were produced from a single mix of concrete 1A. The rings were sealed and placed in an oven. The relative humidity was maintained at approximately 80 % during the tests to avoid evaporation. At the age of (approx.) 26.5 h the air temperature inside the oven was rapidly increased from 24 °C to 35 °C, leading to an increase of ring temperatures and negative strains of 10 K and 9.5  $\mu$ m/m, respectively, until 29 h (**Figure 6–26**). Note that the initial peak in the strain curve is an artefact owing to the abrupt increase of air temperature and the associated non-uniform heating of the rings and the strain gauges. Then, at 29 h the air temperature was reduced rapidly from 35 °C to 22 °C. As a consequence, the ring temperature decreased by 12 K and the strain by 13.5  $\mu$ m/m until 32 h. Again, the abrupt change of temperature led to a strain peak not corresponding to the change of stresses, this time smaller and in opposite direction.

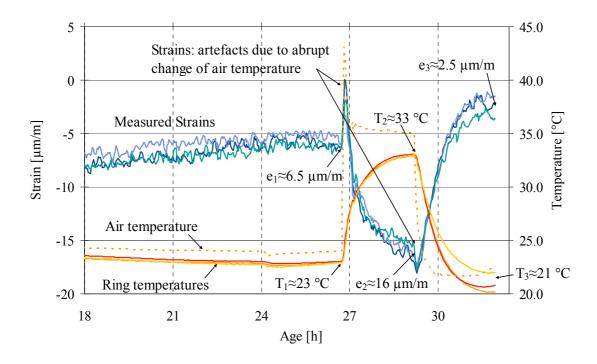


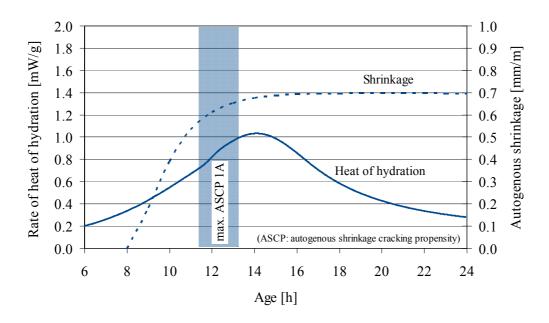
Figure 6-26 Non-isothermal restrained ring tests: mean steel ring strain and temperature vs. age

The change of stress was calculated from the changes of strain and temperature. The increase of temperature between 26.5 h and 29 h led to an increase of the actual stress of 1.29 MPa. The increase of the theoretical-elastic stress was 1.48 MPa. The creep factor amounted to 0.128. Hence, 12.8 % of the theoretical-elastic stress were eliminated by creep or relaxation. In view of the much advanced hydration this appears a realistic value. At very early age the creep factors were much higher (cf. 6.3.2.2 and 6.4.11).

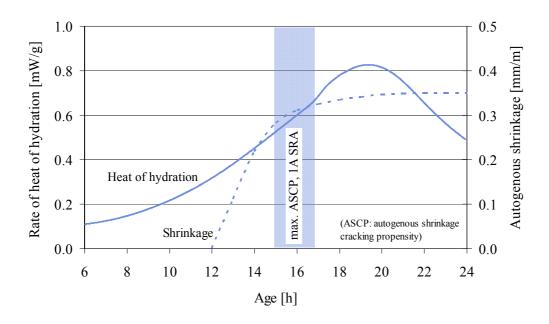
The calculation of the theoretical-elastic stress was carried out with an elastic modulus of 35.0 GPa for the entire period of temperature increase, whereas for the period of cooling a value of 36.5 GPa was assumed as a consequence of advanced hydration at increased temperatures. The decrease of the theoretical-elastic stress due to cooling amounted to 1.84 MPa. The actual stress within that period decreased by 1.84 MPa, too. Hence, the creep factor was zero. This can be seen as an indication of the reliability of the measurements and calculations because creep can be expected to play a negligible role whenever the stress decreases constantly and rapidly. However, slightly different input parameters would have significantly changed the value of the elastic stress. For instance, a coefficient of thermal expansion of  $11.5 \,\mu\text{m/m/K}$ , i.e. only  $0.3 \,\mu\text{m/m/K}$  higher than the applied one, would have yielded an elastic stress of  $1.74 \, \text{MPa}$ , i.e. a notable difference. Also, several simplifying assumptions were made. More work is required to eventually establish the suitability of the outlined method.

## 6.4.8 Period of maximum cracking propensity

For future investigations of the autogenous shrinkage cracking propensity it can be helpful to localize in advance the period of maximum cracking propensity. Then especially the required tensile strength tests can be intensified during that period to get more reliable results. The following figures show the autogenous shrinkage and the rate of heat of hydration over time for concrete 1A and 1A SRA (Figure 6–27 and Figure 6–28). In addition, the period of highest autogenous shrinkage cracking propensity is indicated by the shaded bar.



**Figure 6–27** Period of maximum autogenous shrinkage cracking propensity (ASCP, shaded area) in comparison to free autogenous shrinkage and rate of heat of hydration, concrete 1A

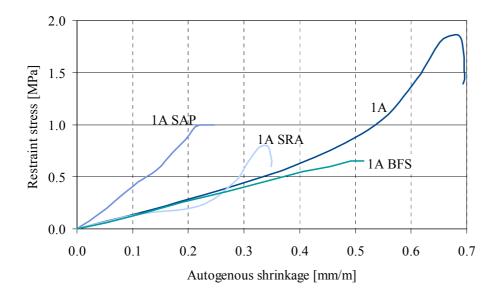


**Figure 6–28** Period of maximum autogenous shrinkage cracking propensity (ASCP, shaded area) in comparison to free autogenous shrinkage and rate of heat of hydration, concrete 1A SRA

In both cases the maximum cracking propensity occurred before the maximum of heat release: 2 h earlier with concrete 1A and 4 h with concrete 1A SRA, approximately. Furthermore, in both cases the period of maximum cracking propensity coincided with the transition period from very rapid to very slow shrinkage. Results are based on investigations with one cement, and the development of the influencing parameters is closely related to the utilized cement. Hence, the findings at this point cannot be generalized. More tests with different cements are required to clarify whether the similarities found for the reference concrete and the concrete with shrinkage-reducing admixture indicate some general rule.

### 6.4.9 Restraint stress versus autogenous shrinkage

An interesting aspect of investigations of the autogenous shrinkage cracking propensity is the relation between free shrinkage and stress under restraint conditions. The following figure shows the restraint stress as function of the free autogenous shrinkage (Figure 6–29). Irrespective of the concrete composition almost linear relations existed up to a strain of approximately 0.1 mm/m. These linear relations remained valid up to strains of 0.2, 0.3 and 0.4 mm/m for concretes 1A SAP, 1A and 1A BFS, respectively. With concretes 1A SAP and 1A BFS one can roughly speak of a linear relation throughout the whole test period. This can be attributed to the more steady development of autogenous shrinkage and mechanical properties. In contrast to that, with concretes 1A and 1A SRA the stress first increased and then decreased disproportionally. The rapid decrease occurred when shrinkage was proceeding at very low rates (cf. Figure 4–21). In relation to the free autogenous shrinkage the stress development was most pronounced with concrete 1A SAP, i.e. smaller strains led to higher stresses. This was due to the fact that the superabsorbent polymers did postpone the beginning of autogenous shrinkage whereas the mechanical properties developed as rapidly as in concrete 1A.



**Figure 6–29** Restraint stress vs. autogenous shrinkage from 0 to 24 h

# 6.4.10 Cracking propensity versus autogenous shrinkage

One of the long-term goals of research on autogenous shrinkage is to predict the cracking propensity of a concrete from its free autogenous shrinkage. The following figure shows the relation of these two properties for the ultra-high strength concretes investigated in this study (Figure 6–30). The figure is built on the data shown in Figure 4–20 and Figure 6–22.

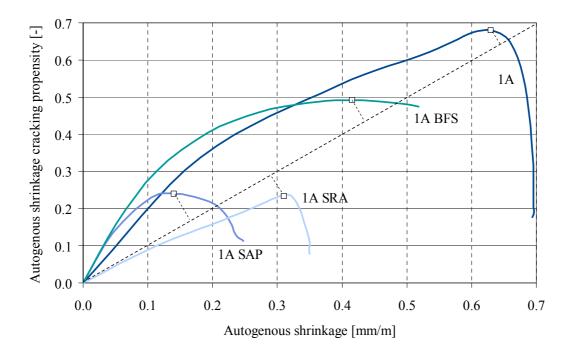


Figure 6–30 Autogenous shrinkage cracking propensity as function of autogenous shrinkage (solid curves), maximum cracking propensity in comparison to the function y = x (single values and dotted line)

Concretes with higher shrinkage led to higher maxima of cracking propensity, except for concrete 1A SRA. Remarkably, the maxima were relatively close to the linear function y = x. Hence, as a rule of thumb, at least for concrete compositions with the utilized cement, the autogenous shrinkage cracking propensity can be assumed to be in the range of the free autogenous shrinkage (expressed in mm/m). This rule needs to be verified for other concrete compositions.

Furthermore, the cracking propensity of concrete 1A (and similarly that of concrete 1A SRA) decreased rapidly after the peak. Notably, in the 2 h following the point of maximum cracking propensity of concrete 1A at 0.63 mm/m, i.e. between approximately 12 and 14 h, the actual stress increased from 1.45 to 1.87 MPa (cf. Figure 6–8). Also the autogenous shrinkage increased by another 0.064 mm/m, i.e. by additional 10 %. However, the rates of autogenous shrinkage and modulus of elasticity already decreased during that period, whereas the rate of the splitting tensile strength increased (cf. Figure 4–20 and Figure 3–9). Hence, from this period the resistance increased more rapidly than the stress, and therefore the autogenous shrinkage cracking propensity decreased at an increasing rate.

The high rate at which the autogenous shrinkage cracking propensity of concrete 1A decreased indicates that only large and rapid temperature decreases may increase the stress-strength ratio to values above the maximum ratio observed at the age of around 12 h. Thus, the superposition of thermal and autogenous shrinkage strains in this case appears not as serious a problem as sometimes suspected for high strength concrete in general, for instance in [IPACS].

However, this may be different for concrete 1A BFS. Here the stress-strength ratio decreases much more slowly, so that rapidly decreasing concrete temperatures due to decreasing ambient temperature may lead to stress-strength ratios above the one observed in the quasi-isothermal ring tests. Notably, the coefficient of thermal expansion was much higher for concrete 1A BFS as well (cf. Figure 3–12).

### **6.4.11** Further considerations on creep

As was outlined in the previous two sections, the prediction of restraint stress and cracking propensity from free autogenous shrinkage and mechanical properties is an essential task of research. Due to the high stress relaxation capacity at very early age, the restraint stress cannot be computed as per Hooke's law from autogenous shrinkage and modulus of elasticity. Some term is required to account for the influence of creep and relaxation. Commonly such terms are determined by creep and relaxation tests. This approach is not feasible with restrained ring tests (5.5). Instead, the ratio of actual to theoretical-elastic is used to quantify the influence of time-dependent deformations (cf. 6.3.2.2).

Here, for three out of four investigated ultra-high strength concretes it was found that roughly 75 % of the theoretical-elastic stress were eliminated by creep and relaxation during the first 24 h (6.4.5.3). This was in fair agreement with results of an earlier study cited above. Furthermore, in the previous section it was shown that the maximum cracking propensity for the concretes investigated could be estimated with reasonable precision by means of a linear function from the autogenous shrinkage. This surprisingly simple correlation in part can be attributed to the fact that the overall influence of creep was similar for all concretes.

Nevertheless, the global consideration of elastic and actual stresses at 24 h gives a rather rough description of the influence of creep and relaxation, not suitable for predicting the precise development of the actual stress from the elastic stress. In particular, it is impossible to estimate the stress development in case additional stresses occur, for instance due to drying shrinkage.

To examine whether a more detailed analysis is feasible, the creep factor was calculated from the stress increments that occurred within 1 h. **Figure 6–31** shows the '1 h-creep factor' for concrete 1A between the age of 8 h and 16 h (left) and between the age of 8 h and 24 h (right). The left part of **Figure 6–31** reveals that the creep factor until 14 h was roughly 0.5 to 0.75. Hence, 50 to 75 % of the theoretical-elastic stress increment were 'relaxed' within each hour. After 14 h the creep factor exceeded 1, meaning that the actual stress decreased while the theoretical-elastic stress still increased.

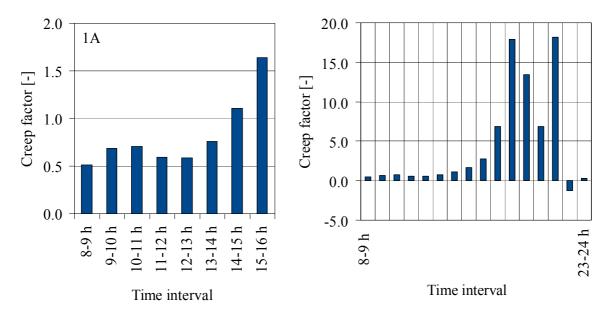


Figure 6–31 Creep factor for intervals of 1 h, concrete 1A, left: 8 h to 16 h, right: 8 h to 24 h

As can be seen in the right part of Figure 6–31, at approximately 18 h the creep factor started to vary enormously. During the subsequent period the theoretical-elastic stress did hardly change anymore (cf. Figure 6–12, left). The division by very small numbers obviously led to large errors. No consistent interpretation is possible.

For the interval from 22 h to 23 h the creep factor resulted negative because theoretical-elastic and actual stress decreased by 0.013 MPa and 0.03 MPa, respectively. The decrease of 0.03 MPa (actual stress) corresponded to a decrease of the steel ring strain of 0.22  $\mu$ m/m, whereas the standard deviation in the repeatability tests was 0.8 to 1.5  $\mu$ m/m during that period, cf. **Figure 6–5**. The decrease of 0.013 MPa (theoretical-elastic stress) corresponded to a free shrinkage deformation of approximately 0.06  $\mu$ m, which was one tenth of the resolution of the utilized laser. Hence, the deformations in the time-intervals observed were too small to be reasonably measured with the utilized test setups.

The following figure shows the 1 h-creep factor for concrete 1A SAP and 1A BFS (**Figure 6–32**). In both cases the development was more consistent than with concrete 1A. With concrete 1A SAP, the creep factor increased until it reached values between 1 and 1.2, indicating that the actual stress slightly decreased while the elastic stress continued to increase. With concrete 1A BFS the creep factor increased continuously. Values remained below 1, indicating that the actual stress kept increasing as well. Notably, with both concretes the elastic stress increased noticeably throughout the whole test period (cf. **Figure 6–12**, right). This appears to be a precondition for sound results.

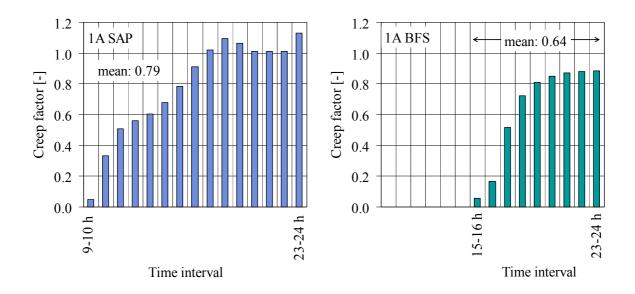


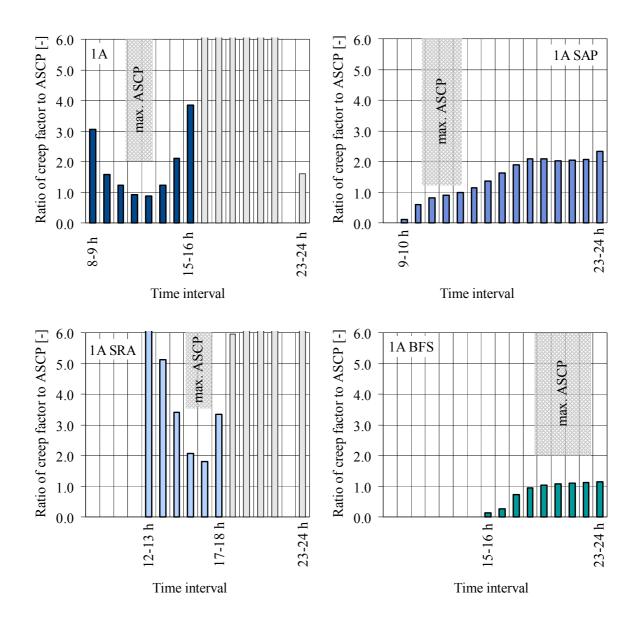
Figure 6-32 Creep factor for intervals of 1 h, left: concrete 1A SAP, right: concrete 1A BFS

In addition, the mean of the 1 h-creep factor values within the period of restraint stress was calculated. The mean was higher with concrete 1A SAP (0.79 compared to 0.64 for concrete 1A BFS), although the 24 h-creep factor was higher with concrete 1A BFS (0.78 compared to 0.64 for concrete 1A SAP, **Figure 6–13**). This again reveals the limited significance of the 24 h-creep factor.

The 1 h-creep factor is more specific in that regard. It allows computing the actual stress development from the theoretical-elastic stress, at least in the case of concretes 1A SAP and 1A BFS. However, the conventional description of creep by means of compliance functions serves a different and more general purpose. Such functions are to describe creep for *arbitrary* stress histories, given they are caused by service loads, i.e. stress levels below the proportionality limit acting from a later age (cf. 5.5). Since the 1 h-creep factor in this investigation was the result of non-constant deformations and stresses, it would not be suitable for a general formulation of creep even if the proportionality limit was not exceeded. Furthermore, compliance functions generally consider the age at loading, the magnitude of the load and basic mechanical properties, but not the specific concrete composition (cf. 5.5). Here, deformations and stresses essentially depended on the concrete composition.

To possibly find common patterns among the concretes investigated and to obtain a more general characterization of creep, one can relate the calculated creep factor to the calculated stress

level. The ratio of the 1 h-creep factor to the respective stress level, i.e. to the autogenous shrinkage cracking propensity (ASCP), is shown in **Figure 6–33**. (The calculation was based on mean stress levels; for instance, the stress level for the time interval from 12 h to 13 h was the mean of the cracking propensity at 12 h and 13 h.) In addition, the period of highest cracking propensity is indicated. With concretes 1A and 1A SRA, the inconsistent data within the period of very small changes of theoretical-elastic stress are marked light grey.



**Figure 6–33** Ratio of 1 h-creep factor to autogenous shrinkage cracking propensity (ASCP), concrete 1A (top left), concrete 1A SAP (top right), concrete 1A SRA (bottom left) and concrete 1A BFS (bottom right)

Two different patters can be observed. With concrete 1A and 1A SRA, the ratio of creep factor to cracking propensity initially was higher and strongly decreased during the first hours after solidification. The ratio was lowest in the period of maximum cracking propensity. Subsequently the ratio strongly increased again. This increase may be attributed to the high stress

levels in the preceding period. A similar behavior of concretes 1A and 1A SRA had already been observed for the restraint stress as a function of the autogenous shrinkage (Figure 6–29).

With concretes 1A SAP and 1A BFS, the pattern was different. The ratio gradually increased and then remained in the range of the maximum values. Compared to concretes 1A and 1A SRA, the absolute values were significantly lower, especially with concrete 1A BFS. In the case of concrete 1A SAP this probably was due to the attenuation of the autogenous shrinkage (in comparison to concrete 1A). With concrete 1A BFS the reason may be seen in the fact that hydration proceeded at much lower rates and mechanical properties developed more slowly.

Although being the result of a relatively simplistic evaluation, the clearly different patterns indicate that the creep and relaxation behavior of ultra-high strength concrete at very early age may be difficult to generalize. Optional constituents like shrinkage reducing admixtures and supplementary cementitious materials not only reduce the autogenous shrinkage, they also considerably alter the very early hydration. On the other hand, the different behavior does not seem to be a mere function of the degree of hydration. The creep pattern of concrete 1A and concrete 1A SAP was completely different, although it can be concluded from the very similar development of the mechanical properties that also the degree of hydration developed similarly.

The description of creep due to service loads by means of conventional creep and relaxation tests allows for generalizing the behavior of different concretes. Although models may not in all cases provide very precise predictions of long-term deformations, it is often not required to individually test the viscoelastic behavior of a specific concrete composition. Knowledge of the mechanical properties usually is sufficient. This does not seem to be the case with ultrahigh strength concrete at very early age.

It appears that the conventional way of describing the influence of time-dependent deformations is less appropriate for the problem of very early age cracking due to restrained autogenous shrinkage. Especially in view of the enormous experimental difficulties with very high tensile stress levels at very early age, an approach that does not aim at a complete and general description of creep and relaxation but focuses on the residual stress capacity appears more promising. It possibly can provide the data that is needed to use ultra-high strength concrete safely, i.e. without risking cracks that impair the durability, tightness or aesthetics of structures.

The main questions within such a residual stress capacity approach would be: how much additional stress can a concrete sustain, and how rapidly can this stress be applied without causing cracks. This study already executed some of the steps required to systematically investigate the relevant phenomena in accordance with this approach. The modified ring tests that made use of harsh drying actually were to test the residual stress capacity (6.4.6.4). To gain information on the tensile strength, it was attempted to provoke failure. By varying not only the age at additional loading but also the relative humidity, the effects of very high stress levels can be investigated more systematically.

The restrained ring test is sufficiently simple and inexpensive to allow for a sufficient number of tests, and possibly multiple tests of a number of similar concrete compositions reveal generally valid relations. The different time-dependent relaxation of stresses can be described by

the creep factor and the ratio of creep factor to cracking propensity. In case the additional load is triggered by drying, the free drying shrinkage needs to be measured to compute the theoretical-elastic stress. As an alternative the specific heating of the rings can be used; the required equations and a schematic illustration of the test procedure were given herein (6.3.2.3, 6.3.3).

The residual stress capacity approach offers the chance for a straightforward phenomenological investigation of the influence of creep and relaxation on cracking at early and very early age. It can be complemented by micro-analytical tests of the development of the pore structure and of other properties characterizing hydration. In addition, fracture mechanics may be a useful tool to understand and describe the development of micro-cracks and failure at that early stage of hydration.

### 6.5 Summary

The methodological analysis of alternative test methods proved the restrained ring test to be more suitable than large temperature-stress testing machines for investigating the autogenous shrinkage cracking propensity of fine-grained ultra-high strength concrete. The restrained ring test was used to determine tensile stresses due to restrained autogenous shrinkage. Steel ring strains in restrained ring tests started to increase at 8 h (concrete 1A), 9 h (1A SAP), 12 h (1A SRA) and 15 h (1A BFS). Restraint stresses were highest with the reference concrete 1A, reaching up to 1.87 MPa. The concrete compositions with reduced autogenous shrinkage led to lower stresses.

The standard deviation of the restrained ring test in a specific repeatability test series was roughly 1  $\mu$ m/m. If assessed on the basis of the coefficient of variation, repeatability during the first 24 h was rather mediocre. However, the initially very high coefficient of variation mostly was a consequence of the period under investigation. The steel ring strains were captured from the very beginning and developed slowly at the beginning. Generally, even with a very low standard deviation the coefficient of variation is high as long as strains are very low.

In any case a minimum of three to four tests is recommended. Three strain gauges per ring appear to be sufficient. They should be glued to the ring with utmost care. A Wheatstone bridge is not mandatory. Furthermore, the temperature of the steel ring should be continuously measured, and, if the cross section is large, also the temperature of the concrete. This not only allows for eliminating disturbing temperature artifacts in the recorded resistance data, it also can show to which degree isothermal conditions were provided for. The higher the degree of restraint is, the lower are the strains and the more care is required to achieve precise results. The steel ring strains should be close to nil prior to solidification. The onset of steel ring compression can be specified approximately to half an hour, at best to a quarter of an hour.

An analytical solution for the thermal stress component in non-isothermal restrained ring tests was added to the existing evaluation methods. Specific temperature changes and additional drying were suggested as means for investigating the influence of relaxation and creep with restrained ring tests. Furthermore, the possible effect of stress redistribution, generally neglected in the past, was quantified and considered in the stress calculations.

The early splitting tensile strength development was investigated. It was found that an age-dependent correction of the splitting tensile strength is required to determine the resistance to cracking at early and very early age. The applicability of the correction factor, calculated on the basis of a recently suggested formula for the conversion of splitting to uniaxial tensile strength, was confirmed by restrained ring tests with harsh drying to the ambient air, leading to cracking and thereby indicating the maximum stress capacity at different ages.

The maximum autogenous shrinkage cracking propensity was observed for concrete 1A. It amounted to approximately 0.68. Values were lower for concrete 1A BFS (0.49) and much lower for concretes 1A SRA (0.24) and 1A SAP (0.24). No cracks were observed on any of the rings. For all concretes the maximum cracking propensity was roughly equivalent to the free autogenous shrinkage expressed in mm/m.

In relation to the autogenous shrinkage the restraint stress development was most pronounced with concrete 1A SAP containing superabsorbent polymers. This was explained by the fact that the mechanical properties developed as fast as in the reference concrete whereas the autogenous shrinkage was delayed.

Evidenced for concretes 1A and 1A SRA, the maximum autogenous shrinkage cracking propensity occurred in the transition period from rapid to slow autogenous shrinkage and prior to the maximum rate of heat release, i.e. at very early age. Furthermore, the cracking propensity of concrete 1A after having peaked decreased much faster than that of concrete 1A BFS. This may signify that concrete 1A is less vulnerable to additional tensile stresses due to decreasing ambient temperatures.

The creep factor, when evaluated from theoretical-elastic and actual stress on an hourly basis, was determinable only for periods with substantial changes of stress. This underlines the relevance of the suggested residual stress capacity approach which makes use of additional thermal or drying shrinkage stresses to investigate and quantify the influence of creep and relaxation. Creep factors for a longer period of e.g. 24 h do neither give a sufficiently precise picture of very-early age viscoelasticity nor do they allow for the separation of thermal and shrinkage stresses according to the analytical solution that was suggested for non-isothermal restrained ring tests.

# 7 Summary, conclusions and outlook

# 7.1 Summary and conclusions

Restrained autogenous shrinkage can lead to considerable tensile stresses in concrete with low water/cement-ratio, especially in ultra-high strength concrete that was focused upon herein. To avoid cracking, it is essential to comprehensively investigate free and restrained autogenous shrinkage. However, the nature of these phenomena, in particular their very early onset, brings about some formidable challenges in terms of methodology.

Hydration is a complex process, in Portland cement paste and even more so in concretes with numerous constituents influencing hydration. It is very difficult and laborious to completely capture the very early development of hydration and of micro-structural properties. Therefore micro-analytical approaches, aiming at the complete understanding and modeling of underlying mechanisms, will require much time and large resources.

This study was to contribute to a more pragmatic and immediate solution. The strategy behind the chosen phenomenological approach was, on the one hand, to assess the autogenous shrinkage cracking propensity for the comparably simple case of constant temperature, and, on the other hand, to select, improve and develop methods suitable for investigating those parameters whose influence is most difficult to capture up to now, namely temperature and viscoelasticity. The basic objective of this strategy is to provide the methods for measuring all relevant quantities as precisely and efficiently as possible under all relevant conditions. The long-term goal is the empirical modeling of the autogenous shrinkage cracking propensity from as few and as simple input parameters as feasible and reasonable.

As indicated, the present knowledge about the influence of temperature on autogenous shrinkage and related cracking is insufficient. The corrugated tube method for measuring the autogenous shrinkage, recently standardized in the U.S., is precise and reliable, as was confirmed by the own tests. However, it is limited to isothermal conditions. Besides, with sticky or stiff mortars the casting is hampered by the small inlet diameter of the tubes. Therefore the 'shrinkage cone method for measuring the autogenous shrinkage' was developed. The investigation of repeatability and reproducibility proved that the method is precise, reliable and easy to use. The very good agreement with results obtained with the corrugated tube method evidenced the accuracy. Since the method is also suitable for tests under non-isothermal conditions, it will facilitate the required clarification of the role of temperature.

Tensile stresses due to autogenous shrinkage were investigated by the restrained ring test. The method proved to be very efficient and sufficiently precise. Most tests were carried out at a constant temperature of 20 °C and evaluated by means of an existing stress solution. In addition, an analytical solution for the thermal stress component due to different thermal expansion of steel and concrete ring at changing temperatures was derived. This will facilitate the future evaluation of restrained ring tests under non-isothermal conditions. Furthermore, the potential stress redistribution in restrained ring tests at early and very early age was pointed out.

The development of the stress-strength ratio was used to determine the autogenous shrinkage cracking propensity of four different ultra-high strength concretes under quasi-isothermal conditions. The applicability of the stress-strength criterion was analyzed and confirmed. Re-

sults of splitting tension tests at very early age needed to be multiplied by a factor of 2 to 3 to correspond to the maximum stress at failure observed when rings dried out at very low relative humidity from respective ages. These values as well as the factor's age-dependent disproportional decrease were in good agreement with a recently proposed function for the development of the ratio of uniaxial to splitting tensile strength.

The influence of tensile creep and relaxation on restraint stresses is known to disproportionally increase with increasing stress level and decreasing age at loading. However, there is hardly any quantitative knowledge as to this influence at very early age. An ample experimental investigation of tensile creep at very early age with very high stress levels and of its influence on the cracking propensity would be extremely demanding and has not been attempted, yet. Common stress rigs like the particularly large temperature-stress testing machines are hardly suitable for comprehensive parameter studies; actually, most investigations were found to have included too few tests to allow for any statistically backed conclusion. This appears particularly critical in view of the two round-robin tests with temperature-stress testing machines on free and restrained autogenous shrinkage which failed to prove a sufficient reproducibility of results.

Therefore it was analyzed whether the restrained ring test could be an alternative also for this task. In its elementary form it is a passive test method, i.e. the strains cannot be controlled and the stress history is always conditional on the autogenous shrinkage and the development of the mechanical properties. The influence of time-dependent deformations can be investigated by comparing theoretical-elastic and actual stress, however, not always in a satisfactory manner. A more versatile investigation is feasible with an additional stress component which can be created at arbitrary points in time and with different rates of stress increase by either drying of the concrete to the ambient air or by a specific variation of temperature. If the additional free concrete strains are known, the creep factor and the ratio of creep factor to stress level may be suitable to describe the influence of time-dependent deformations.

The main conclusions regarding the methodology can be summarized as follows:

- The shrinkage cone method is suitable for testing the autogenous shrinkage of hardening cement paste and fine-grained concrete. This includes sticky and stiff materials. Tests under non-isothermal conditions are possible as well.
- The restrained ring test is suitable for determining the autogenous shrinkage cracking propensity of concrete. It is more appropriate for determining the cracking propensity of fine-grained ultra-high strength concrete than temperature-stress testing machines.
- The stress-strength failure criterion is applicable at very early age. Splitting tension tests can be used to determine the strength, however, results require a strong age-dependent correction.
- An additional stress component in restrained ring tests may be used to investigate the
  residual stress capacity and the influence of stress relaxation on cracking. The additional stress can be produced by temperature changes or by drying.

#### 7.2 Outlook

Splitting tension tests can be used to determine a tensile strength at very early age. However, the appropriate execution of the test and the conversion to uniaxial tensile strength values in that period are still associated with significant uncertainties. The influence of the rate of stress increase on strength needs to be clarified. The coherent results achieved in this study by comparing the splitting tensile strength to the ultimate stress in specific restrained ring tests should be confirmed by further such tests.

Furthermore, it is required to develop suitable empirical models for the tensile strength development at very early age. It was shown that the assumption of a linear relation between tensile strength and degree of hydration is rather unsuitable at very early age. The suitability of sigmoid functions for modeling the splitting tensile strength and the modulus of elasticity should be examined.

The modulus of elasticity is required to evaluate the viscoelastic behavior. The modulus in tension would be the most appropriate quantity in this regard. It needs to be clarified whether the dynamic modulus of elasticity tested with the resonance frequency method provides proper results at very early age, as was assumed herein. From a practical point of view, automatic methods would be preferable.

The restrained ring test has not often been used to investigate the autogenous shrinkage cracking propensity, yet. Generally, the methodological as well as the absolute results of this study call for verification by other researchers. In particular, a round robin test ought to be carried out to further investigate the precision of the test method. In addition, similar investigations with concrete containing large aggregates and fibers should be undertaken in the future.

In the cases that were investigated the period of maximum autogenous shrinkage cracking propensity coincided with the transition of fast to slow autogenous shrinkage. If this turned out to be a general rule for certain types of ultra-high strength concrete, it could significantly facilitate future investigations and reduce the experimental effort. Furthermore, the maximum autogenous shrinkage cracking propensity of the concretes investigated was roughly equivalent to the free autogenous shrinkage expressed in mm/m. If this was confirmed for other concrete compositions, it would be a very helpful rule of thumb and possibly the basis for future modeling attempts.

The autogenous shrinkage cracking propensity was tested at a virtually constant temperature of 20 °C. Tests at different temperature levels and also under non-isothermal conditions must follow to eventually assess the cracking propensity under realistic on-site conditions. The developed shrinkage cone method and the derived stress solution for the effect of temperature changes in restrained ring tests facilitate such tests. However, much more work is to be done in that regard. Especially, an efficient and precise method for determining the development of the coefficient of thermal expansion is required.

Finally, a new approach for the experimental investigation and mathematical description of the influence of creep and relaxation on cracking was suggested. Its application is relatively straightforward but not trivial. It needs to be confirmed that the proposed residual stress capacity approach is appropriate and efficient.

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## 9 Annex

**Table 9–1** Characteristic parameters of cements (A, B) and ground-granulated blast furnace slag (GGBFS)

				Cen	nents			GCDFG
Parameter	Unit		A	1)		В	3)	GGBFS 4)
		1	2	3 <sup>2)</sup>	4	1	2	

	Chemical analysis – loss-on-ignition-free																															
Loss on ignition	950 °C, Infrared spectroscopy		1.19	1.22	1.17	0.85	1,08	0,65	0.1																							
SiO <sub>2</sub>			21.2	21.43	22.7	21.96	22.28	22.38	36.39																							
Al <sub>2</sub> O <sub>3</sub>			3.31	3.31	3.36	3.2	4.1	3.62	10.95																							
TiO <sub>2</sub>			0.23	0.22	n. d.	0.2	0.24	0.20	0.84																							
Fe <sub>2</sub> O <sub>3</sub>	X-ray fluorescence	M%	5.21	4.92	5.5	5.53	1.39	1.15	0.28																							
MgO	analysis		0.73	0.74	0.68	0.65	0.92	0.96	11.72																							
CaO				65.15	64.8	67.4	64.97	65.65	67.09	37.64																						
$SO_3$				2.01	2.19	2.2	2.25	3.5	3.55	0.03																						
Na <sub>2</sub> O-equivalent			0.5	0.54	0.44	0.44	0.52	0.58	0.84																							
C <sub>3</sub> S	X-ray		71.02	68.37	65.76	62.87	61.03	69.14	-																							
C <sub>2</sub> S	fluorescence		8.12	10.75	15.47	15.78	18.55	12.11	-																							
C <sub>3</sub> A	analysis,			_	_	0	0.46	0	0	8.59	7.65	-																				
C <sub>4</sub> AF	phase calculation						_	_	_		-	-									16.04	15.18	16	15.3	4.29	3.51	-					
$C_2F$	acc. to Bogue																							0.06	0	0.43	0.9	0	0	-		
Lime standard	X-ray												96.12	95.02	92.69	92.7	93.47	96.60	-													
Silica modulus	fluorescence		2.5	2.6	2.56	2.51	4.06	4.69	-																							
Alumina modulus	analysis																									0.64	0.67	0.61	0.58	2.94	3.14	-
$C_3S$	V 4:66 ati		60.26	56.98	n. d.	n. d.	62.41	n. d.	-																							
$C_2S$	X-ray diffraction, refinement acc. to Rietveld							17.39	22.32	n. d.	n. d.	16.66	n. d.	-																		
C <sub>3</sub> A			1.98	2.04	n. d.	n. d.	8.4	n. d.	-																							
C <sub>4</sub> AF			14.26	13.04	n. d.	n. d.	2.7	n. d.	-																							

Granulometric parameters										
Density kg/dm³ 3.18 n. d. n. d. 3,12 n. d. 2.88										
Blaine Fineness	cm²/g	4650	n. d.	4950	4690	4710	n. d.	3890		

Physical parameters according to EN 197-1											
Water demand M% 28.5 29 28.5 29 30,5 n. d											
Initial setting	min	140	170	140	170	155	n. d.	-			
Final setting	min	195	225	190	210	195	n. d.	-			
Compressive strength (2 d)	MPa	30.6	35.1	36.3	32.5	39,8	n. d.	-			
Compressive strength (28 d)	MPa	61.8	64.1	63.8	61.4	64,3	n. d.	-			
Heat of hydration	J/g	320	n. d.	n. d.	n. d.	357	n. d.	1			

<sup>1)</sup> A: CEM I 52,5 R-HS/NA (HS: high sulfate resistance; NA: low alkali equivalent acc. to DIN 1164-10), batch 1 to 4

<sup>2)</sup> Data provided by cement manufacturer

<sup>3)</sup> B: CEM I 52,5 R, batch 1 and 2

<sup>4)</sup> Ground-granulated blast furnace slag

n. d.: not determined

 Table 9–2
 Compressive strength tests

		Age	D	imensio	ns	Mass	Area	Volume	Density	max. Force	Compressive strength	Mean value
Concrete	e	[d]	1 [mm]	w [mm]	h [mm]	[g]	[cm <sup>2</sup> ]	[cm³]	[kg/dm³]	[kN]	[N/mm <sup>2</sup> ]	[N/mm²]
	1		40.9	40.3	40.2	152	16.48	66.26	2.29	166.0	100.7	
	2		41.0	40.7	40.6	158	16.69	67.75	2.33	174.4	104.5	
	3	2	40.6	40.8	40.4	152	16.56	66.92	2.27	170.0	102.6	101.9
	4		40.7	40.3	40.3	154	16.40	66.10	2.33	163.4	99.6	
	1		40.5	40.7	40.3	154	16.48	66.43	2.32	235.2	142.7	
	2		40.3	40.7	40.4	155	16.40	66.26	2.34	223.1	136.0	
	3	7	40.6	40.8	40.5	154	16.56	67.09	2.30	215.2	129.9	137.1
	4		40.5	40.8	40.5	156	16.52	66.92	2.33	231.1	139.9	
1A	1		40.9	40.9	40.5	154	16.73	67.75	2.27	218.7	130.7	
	2	7	40.5	40.6	40.6	153	16.44	66.76	2.29	218.7	133.0	129.6
	3	,	40.7	40.7	40.5	152	16.56	67.09	2.27	207.1	125.0	129.0
	1		40.5	41.1	40.2	153	16.65	66.91	2.29	282.4	169.7	
	2	28	40.7	40.4	40.1	153	16.44	65.94	2.32	253.1	153.9	165.3
	3		40.8	40.4	40.2	153	16.48	66.26	2.31	283.8	172.2	
	1		41.1	40.5	40.5	152	16.65	67.41	2.25	317.2	190.6	
	2	57	40.5	40.4	40.6	153	16.36	66.43	2.30	290.0	177.2	185.3
	3		40.5	41.4	40.6	153	16.77	68.07	2.25	315.6	188.2	
	1		40.2	40.6	40.3	150	16.32	65.77	2.28	162.0	99.3	
	2	_	40.4	40.5	40.3	153	16.36	65.94	2.32	165.9	101.4	100.0
	3	2	40.4	40.3	40.2	151	16.28	65.45	2.31	164.9	101.3	100.8
	4		41.0	40.4	40.3	153	16.56	66.75	2.29	168.0	101.4	
	1		40.4	40.6	40.5	153	16.40	66.43	2.30	222.1	135.4	
	2		40.4	40.5	40.3	152	16.36	65.94	2.31	216.8	132.5	
	3	7	40.4	40.5	40.2	152	16.36	65.78	2.31	208.2	127.2	129.4
	4		40.5	40.5	40.5	153	16.40	66.43	2.30	205.6	125.3	
1 4 C 4 D	5		40.4	40.7	40.4	152	16.44	66.43	2.29	208.0	126.5	
1A SAP	1		40.7	39.8	40.6	152	16.20	65.77	2.31	232.1	143.3	
	2	7	40.7	40.2	40.3	162	16.36	65.94	2.46	209.2	127.9	132.5
	3		41.4	40.4	40.5	159	16.73	67.74	2.35	211.2	126.3	
	1		40.1	40.9	40.4	152	16.40	66.26	2.29	269.6	164.4	
	2	28	40.5	40.3	40.6	157	16.32	66.27	2.37	276.1	169.2	169.8
	3		40.6	40.4	40.2	158	16.40	65.94	2.40	288.6	175.9	
	1		40.3	40.6	40.3	161	16.36	65.94	2.44	295.5	180.6	
	2	56	40.3	40.2	40.3	158	16.20	65.29	2.42	289.3	178.6	179.0
	3		40.2	40.5	40.5	159	16.28	65.94	2.41	289.7	177.9	

Concret	e	Age	Di	imensio	ns	Mass	Area	Volume	Density	max. Force	Compressive	Mean value
		[d]	l [mm]	w [mm]	h [mm]	[g]	[cm <sup>2</sup> ]	[cm³]	[kg/dm³]	[kN]	strength [N/mm²]	[N/mm²]
	1		40.9	40.9	40.1	154	16.73	67.08	2.30	138.1	82.6	
	2	2	40.9	40.7	40.2	153	16.65	66.92	2.29	134.4	80.7	92.2
	3	2	40.3	40.9	40.2	152	16.48	66.26	2.29	136.2	82.6	82.2
	4		41.0	40.8	40.3	154	16.73	67.41	2.28	138.5	82.8	
	1		41.0	40.5	40.2	153	16.61	66.75	2.29	202.2	121.8	
	2		40.7	40.5	40.3	153	16.48	66.43	2.30	198.5	120.4	
	3	7	40.7	40.5	40.2	152	16.48	66.26	2.29	200.1	121.4	120.9
	4		40.7	40.2	40.2	152	16.36	65.77	2.31	201.9	123.4	
4 . 675 .	5		40.7	40.8	40.8	152	16.61	67.75	2.24	194.9	117.4	
1A SRA	6		41.3	41.1	40.3	158	16.97	68.41	2.31	230.9	136.0	
	7	7	41.1	40.4	40.2	155	16.60	66.75	2.32	213.9	128.8	131.7
	8		40.2	40.8	40.5	152	16.40	66.43	2.29	213.4	130.1	
	1		41.0	40.7	40.2	157	16.69	67.08	2.34	287.6	172.3	
	2	28	41.0	40.2	40.4	155	16.48	66.59	2.33	286.1	173.6	171.5
	3		40.8	40.3	40.2	153	16.44	66.10	2.31	277.2	168.6	
	1		40.7	40.5	40.5	155	16.48	66.76	2.32	298.7	181.2	
	2	56	40.2	40.5	40.3	151	16.28	65.61	2.30	320.5	196.9	189.6
	3		39.8	40.8	40.5	149	16.24	65.77	2.27	309.7	190.7	
	1		40.4	40.4	40.3	152	16.32	65.78	2.31	40.4	24.8	
	2	_	40.5	40.4	40.4	153	16.36	66.10	2.31	41.0	25.1	25.0
	3	2	40.4	40.7	40.2	154	16.44	66.10	2.33	40.0	24.3	25.0
	4		40.7	40.3	40.7	152	16.40	66.76	2.28	42.7	26.0	
	1		40.5	40.4	40.5	154	16.36	66.27	2.32	150.3	91.9	
	2	7	40.4	40.5	40.3	153	16.36	65.94	2.32	157.4	96.2	93.5
	3		40.3	40.6	40.5	153	16.36	66.27	2.31	151.3	92.5	
14 PEG	1		40.4	40.3	40.4		16.28	65.78		125.3	77.0	
1A BFS	2	7	41.0	40.7	40.5		16.69	67.58		153.3	91.9	93.6
	3		40.7	40.7	40.5		16.56	67.09		158.0	95.4	
	1		40.6	40.9	40.2	153	16.61	66.75	2.29	242.0	145.7	
	2	28	40.3	40.7	40.6	152	16.40	66.59	2.28	237.6	144.9	144.0
	3		40.4	40.7	40.4	152	16.44	66.43	2.29	232.4	141.3	
	1		40.5	40.8	40.4	152	16.52	66.76	2.28	258.5	156.4	
	2	62	40.3	40.7	40.7	151	16.40	66.76	2.26	267.2	162.9	157.3
	3		41.2	40.7	41.0	155	16.77	68.75	2.25	255.8	152.5	







Figure 9–1 Splitting tension tests: filled mold (top), specimen preparation (centre) and testing (bottom)

**Table 9–3** Splitting tensile strength tests. Rate of stress increase: 3.0 MPa/min

Concret		Age	Diameter				Volume	Density	max. Force	Splitting tensile	Mean value
		[h]	[mm]	[mm]	[g]	[cm <sup>2</sup> ]	[cm³]	[kg/dm³]	[kN]	strength [N/mm²]	[N/mm²]
	1		37.0	45.5	112	10.75	48.92	2.29	0.739	0.28	
	2	9	36.8	44.9	108	10.64	47.76	2.26	0.673	0.26	0.28
	3		37.0	50.9	124	10.75	54.73	2.27	0.864	0.29	
	1		37.0	50.5	125	10.75	54.30	2.30	1.494	0.51	
	2	10	36.8	50.7	124	10.64	53.93	2.30	1.54	0.53	0.52
	3		36.8	51.2	126	10.64	54.46	2.31	1.542	0.52	
	1		36.6	50.0	126	10.52	52.60	2.40	4.77	1.66	
	2	13	36.9	50.0	125	10.69	53.47	2.34	4.63	1.60	1.63
	3		36.9	50.0	127	10.69	53.47	2.38	6.83	2.36	
	1		37.0	49.2	121	10.75	52.90	2.29	12.21	4.27	
1A	2	16.5	36.7	50.2	123	10.58	53.10	2.32	12.18	4.21	4.10
	3		36.6	50.6	122	10.52	53.24	2.29	11.14	3.83	
	1		37.0	51.3	126	10.75	55.16	2.28	18.62	6.25	
	2	19.5	36.7	50.5	125	10.58	53.42	2.34	16.84	5.78	6.16
	3		37.0	49.9	125	10.75	53.65	2.33	18.73	6.46	
	1		36.7	50.7	123	10.58	53.63	2.29	20.24	6.92	
	2	22.5	36.7	51.3	127	10.58	54.27	2.34	20.47	6.92	6.63
	3		37.0	50.4	125	10.75	54.19	2.31	17.72	6.05	
	1		36.9	49.8	123	10.69	53.26	2.31	25.72	8.91	
	2	40	36.9	49.9	124	10.69	53.36	2.32	29.84	10.32	9.82
	3		37.0	49.8	125	10.75	53.55	2.33	29.58	10.22	
	1		36.6	41.1	104	10.52	43.24	2.41	6.11	2.59	
	2	15.5	36.7	50.8	127	10.58	53.74	2.36	10.66	3.64	3.46
	3		37.0	50.4	129	10.75	54.19	2.38	9.61	3.28	
	1		37.0	50.2	127	10.75	53.98	2.35	14.84	5.09	
	2	17.5	36.6	48.8	120	10.52	51.34	2.34	16.17	5.76	5.82
	3		36.9	50.6	126	10.69	54.11	2.33	19.36	6.60	
	1		36.6	50.3	125	10.52	52.92	2.36	20.06	6.94	
1A SAP	2	20.5	36.8	50.4	126	10.64	53.61	2.35	18.15	6.23	6.49
	3		37.0	49.6	123	10.75	53.33	2.31	18.12	6.29	
	1		37.0	50.4	125	10.75	54.19	2.31	18.57	6.34	
	2	23.5	36.6	51.2	127	10.52	53.87	2.36	20.49	6.96	7.02
	3		36.6	50.5	124	10.52	53.13	2.33	22.57	7.77	
	1		37.2	50.4	124	10.87	54.78	2.26	24.34	8.26	
	2	41	36.8	49.8	124	10.64	52.97	2.34	23.73	8.24	8.92
	3		37.0	50.2	126	10.75	53.98	2.33	29.92	10.26	

Concrete	e	Age	Diameter	Height	Mass	Area	Volume	Density	max. Force	Splitting tensile	Mean value
		[h]	[mm]	[mm]	[g]	[cm <sup>2</sup> ]	[cm³]	[kg/dm³]	[kN]	strength [N/mm²]	[N/mm²]
	1		37.1	50.3	126	10.81	54.38	2.32	4.342	1.48	
	2	16	37.3	49.9	125	10.93	54.53	2.29	4.943	1.69	1.73
	3		37.5	50.0	127	11.04	55.22	2.30	5.91	2.01	
	1		36.7	50.7	124	10.58	53.63	2.31	7.39	2.53	
	2	18.5h	36.9	50.4	123	10.69	53.90	2.28	11.42	3.91	3.71
	3		37.0	50.3	124	10.75	54.08	2.29	10.29	3.52	
	1		36.7	50.0	124	10.58	52.89	2.34	12.68	4.40	
	2	20.5	36.8	50.1	123	10.64	53.29	2.31	10.37	3.58	4.19
	3		37.0	50.0	126	10.75	53.76	2.34	13.36	4.60	
	1		36.6	50.4	123	10.52	53.03	2.32	12.22	4.22	
1A SRA	2	21.5h	37.0	49.7	123	10.75	53.44	2.30	13.21	4.57	4.35
	3		36.8	50.3	123	10.64	53.50	2.30	12.37	4.25	
	1		37.2	51.0	125	10.87	55.43	2.26	17.56	5.89	
	2	24	36.8	49.0	121	10.64	52.12	2.32	16.86	5.95	6.01
	3		37.0	50.5	126	10.75	54.30	2.32	18.18	6.19	
	1		36.8	50.4	122	10.64	53.61	2.28	24.4	8.38	
	2		36.8	50.4	123	10.64	53.61	2.29	22.34	7.67	
	3	40.51	37.0	50.0	124	10.75	53.76	2.31	22.92	7.89	7.92
	4	40.5h	37.2	50.8	124	10.87	55.21	2.25	22.54	7.59	7.82
	5		36.8	50.8	125	10.64	54.03	2.31	24.88	8.47	
	6		37.0	49.4	122	10.75	53.12	2.30	19.82	6.90	
	1		37.0	50.8	124	10.75	54.62	2.27	0.876	0.30	
	2	18	37.0	50.1	121	10.75	53.87	2.25	0.653	0.22	0.25
	3		37.0	50.4	122	10.75	54.19	2.25	0.643	0.22	
	1		36.6	50.6	125	10.52	53.24	2.35	1.265	0.43	
	2	20.5	37.0	49.5	124	10.75	53.22	2.33	1.485	0.52	0.50
	3		36.8	50.8	126	10.64	54.03	2.33	1.613	0.55	
	1		36.6	50.6	123	10.52	53.24	2.31	1.736	0.60	
	2	22.5	37.0	49.9	125	10.75	53.65	2.33	2.115	0.73	0.67
	3		36.8	50.8	125	10.64	54.03	2.31	2.006	0.68	
	1		36.8	50.2	125	10.64	53.39	2.34	2.646	0.91	
	2	25.5	36.6	50.6	123	10.52	53.24	2.31	2.728	0.94	0.93
	3		37.0	49.3	123	10.75	53.01	2.32	2.702	0.94	
1A BFS	1		37.0	50.7	124	10.75	54.51	2.27	5.086	1.73	
IA DIS	2	39.5	36.8	50.0	125	10.64	53.18	2.35	6.667	2.31	2.12
	3		37.0	51.7	130	10.75	55.59	2.34	7.001	2.33	
	1		36.6	50.4	124	10.52	53.03	2.34	7.066	2.44	
	2		36.6	50.3	124	10.52	52.92	2.34	5.773	2.00	
	3	47	36.8	50.0	124	10.64	53.18	2.33	7.23	2.50	2.42
	4		36.6	51.3	128	10.52	53.97	2.37	7.004	2.37	
	5		36.6	50.5	127	10.52	53.13	2.39	8.069	2.78	
	1		36.6	50.7	126	10.52	53.34	2.36	9.89	3.39	
	2	65	36.8	50.5	124	10.64	53.71	2.31	10.55	3.61	3.63
	3		37.2	50.4	126	10.87	54.78	2.30	11.47	3.89	
	1		36.8	49.5	122	10.64	52.65	2.32	24.99	8.73	_
	2	163	36.8	50.0	125	10.64	53.18	2.35	18.58	6.43	7.54
	3		37.0	49.0	123	10.75	52.69	2.33	21.21	7.45	

 Table 9–4
 Splitting tensile strength tests. Rate of stress increase: 0.2 MPa/min

Concre	te	Age				Area	Volume	Density	max. Force	Splitting tensile	Mean value
		-						,		strength	
		[h]	[mm]	[mm]	[g]	[cm <sup>2</sup> ]	[cm <sup>3</sup> ]	[kg/dm³]	[kN]	[N/mm <sup>2</sup> ]	[N/mm <sup>2</sup> ]
	1	8	37.2	50		10.87	54.34		0.36	0.12	0.12
	2		37.4	50.2		10.99	55.15		0.355	0.12	****
	1		36.8	50.0	126	10.64	53.18	2.37	0.601	0.21	
	2	9	37.0	50.0	125	10.75	53.76	2.33	0.798	0.27	0.26
	3		37.2	49.6	125	10.87	53.91	2.32	0.831	0.29	
	7		37.0	49.7	124	10.75	53.44	2.32	1.929	0.67	
	8	10	36.8	51.5	126	10.64	54.78	2.30	1.853	0.62	0.65
	9		36.8	51.0	126	10.64	54.24	2.32	1.989	0.67	
	10		37.0	50.2	124	10.75	53.98	2.30	3.609	1.24	
	11	11	37.0	51.2	127	10.75	55.05	2.31	3.36	1.13	1.19
	12		37.0	50.4	125	10.75	54.19	2.31	3.559	1.21	
1A	13		37.0	50.0	122	10.75	53.76	2.27	4.917	1.69	
	14	12	36.8	50.8	125	10.64	54.03	2.31	5.671	1.93	1.84
	15		37.0	51.4	128	10.75	55.27	2.32	5.688	1.90	
	1		37.0	51.0	130	10.75	54.84	2.37	15.68	5.29	
	2	17	36.8	50.4	125	10.64	53.61	2.33	17.25	5.92	5.66
	3		36.8	50.3	125.0	10.64	53.50	2.34	16.74	5.76	
	4		37.0	50.3	126	10.75	54.08	2.33	19.02	6.51	
	5	20	36.8	50.5	127	10.64	53.71	2.36	18.87	6.46	6.46
	6		37.0	50.0	127.0	10.75	53.76	2.36	18.62	6.41	
	7		37.0	49.7	126	10.75	53.44	2.36	17.6	6.09	
	8	24	36.8	50.2	126	10.64	53.39	2.36	19.7	6.79	6.62
	9		36.8	50.4	126	10.64	53.61	2.35	20.29	6.96	
	4		37.0	50.7	124	10.75	54.51	2.27	1.092	0.37	
	5	18	37.0	51.4	124	10.75	55.27	2.24	1.121	0.38	0.37
	6		37.0	50.3	125	10.75	54.08	2.31	1.049	0.36	
	7		37.0	51.4	125	10.75	55.27	2.26	1.658	0.56	
1A BFS	8	20	37.0	50.0	122	10.75	53.76	2.27	1.972	0.68	0.62
	9		37.0	50.0	123	10.75	53.76	2.29	1.79	0.62	1
	10		36.8	50.5	122	10.64	53.71	2.27	2.25	0.77	
	11	23	37.0	50.0	122	10.75	53.76	2.27	2.642	0.91	0.90
	12		36.8	50.7	123	10.64	53.93	2.28	3.028	1.03	1

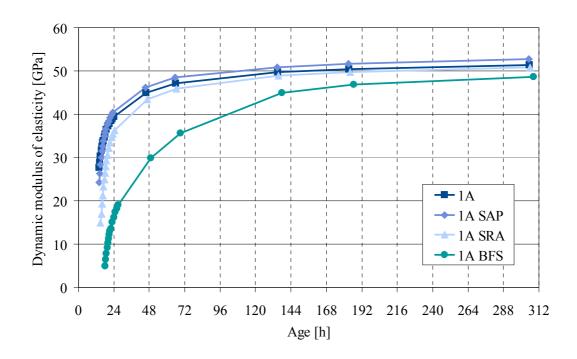


Figure 9–2 Dynamic modulus of elasticity until the age of approximately 300 h

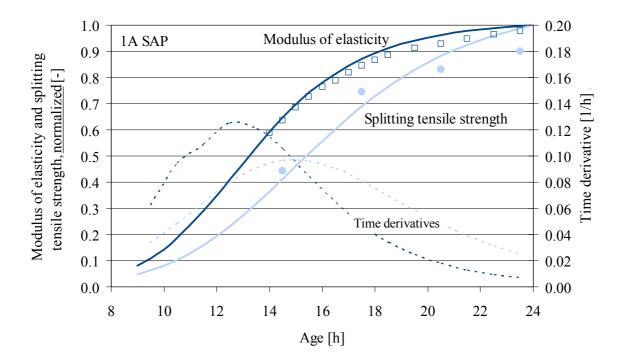


Figure 9-3 Modulus of elasticity and splitting tensile strength normalized to the 24 h-value, concrete 1A SAP

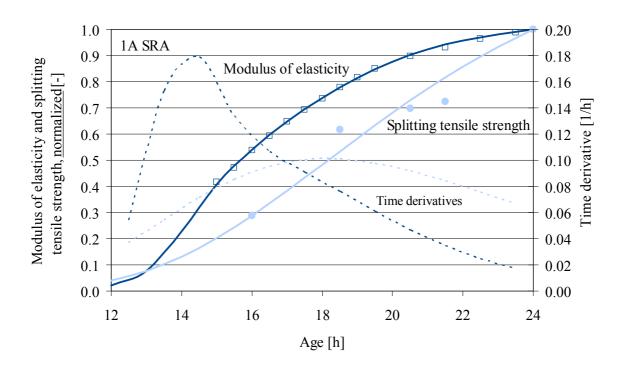


Figure 9-4 Modulus of elasticity and splitting tensile strength normalized to the 24 h-value, concrete 1A SRA

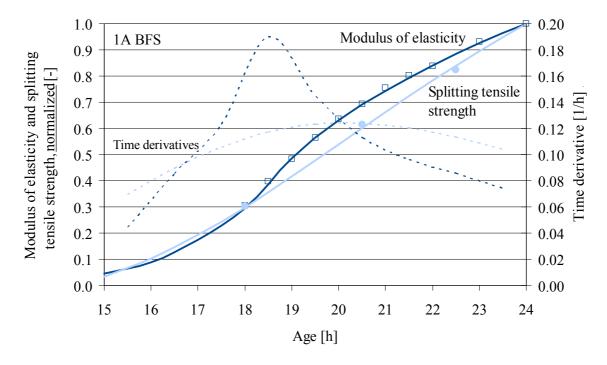
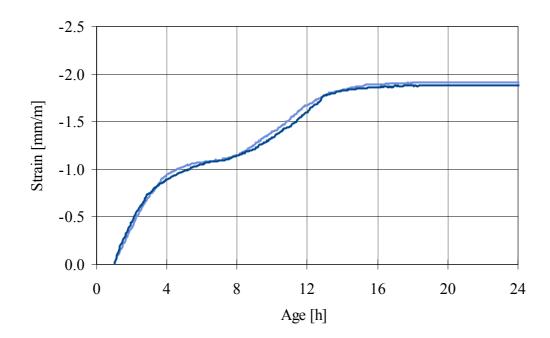
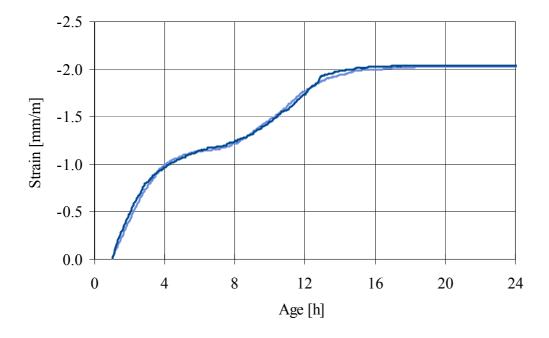


Figure 9–5 Modulus of elasticity and splitting tensile strength normalized to the 24 h-value, concrete 1A BFS



**Figure 9–6** Strain under sealed conditions evaluated from 1 h, measured by the shrinkage cone method (Lab 2), 2 tests with 2 shrinkage cone apparatus, concrete 1A, **test 1** 



**Figure 9–7** Strain under sealed conditions evaluated from 1 h, measured by the shrinkage cone method (Lab 2), 2 tests with 2 shrinkage cone apparatus, concrete 1A, **test 2** 

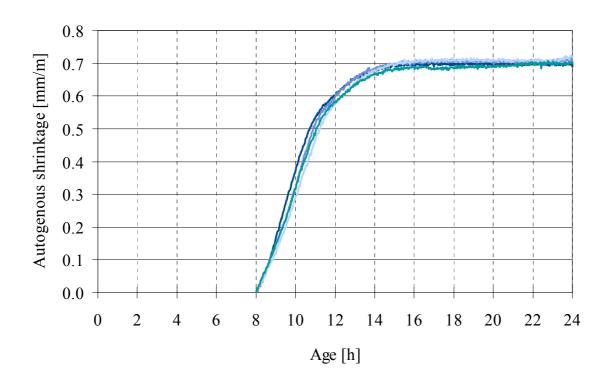


Figure 9–8 Autogenous shrinkage measured four times with the shrinkage cone method, concrete 1A

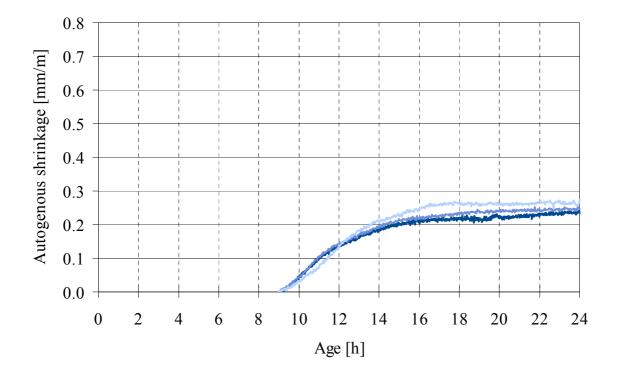


Figure 9–9 Autogenous shrinkage measured three times with the shrinkage cone method, concrete 1A SAP

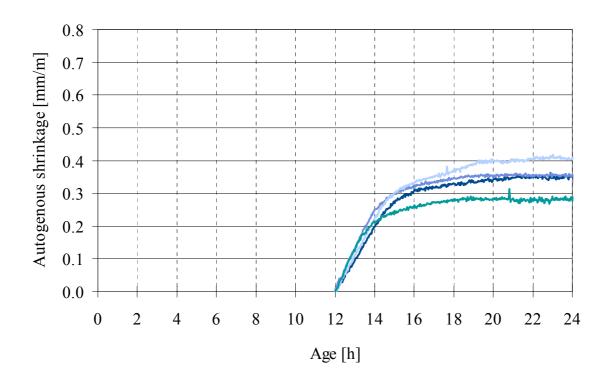


Figure 9–10 Autogenous shrinkage measured four times with the shrinkage cone method, concrete 1A SRA

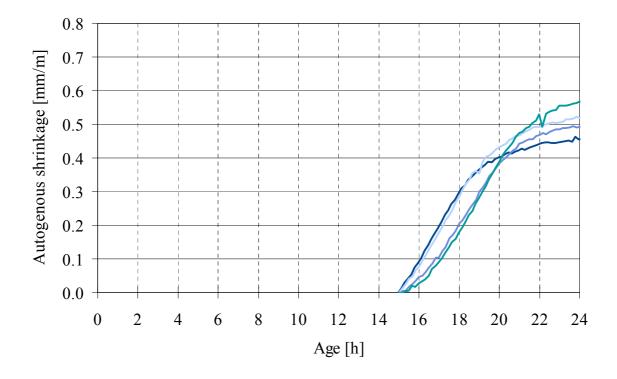


Figure 9–11 Autogenous shrinkage measured four times with the shrinkage cone method, concrete 1A BFS



**Figure 9–12** Movable measuring cabinet for restrained ring tests: 8 ring test setups, data acquisition system for strain gauges and resistance temperature detectors, computer for data storage and control of data acquisition system, uninterruptible power supply; left: front panel, centre: rear panel; right: air tight container for tests at low humidity with ventilation and heating unit (top: top view, bottom: front view)

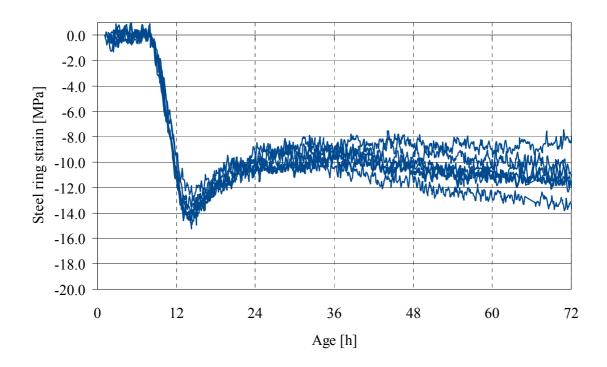


Figure 9–13 Restrained ring tests, main test series, six rings from three mixes, concrete 1A

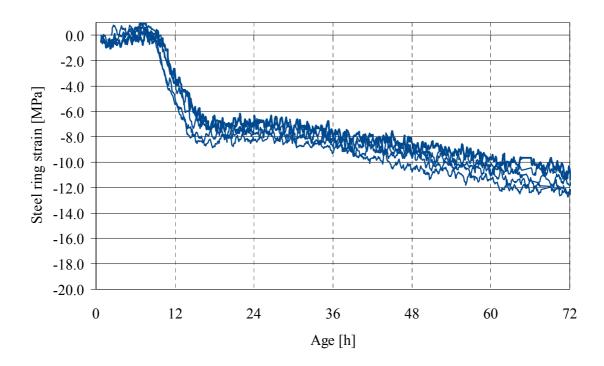


Figure 9–14 Restrained ring tests, main test series, six rings from three mixes, concrete 1A SAP

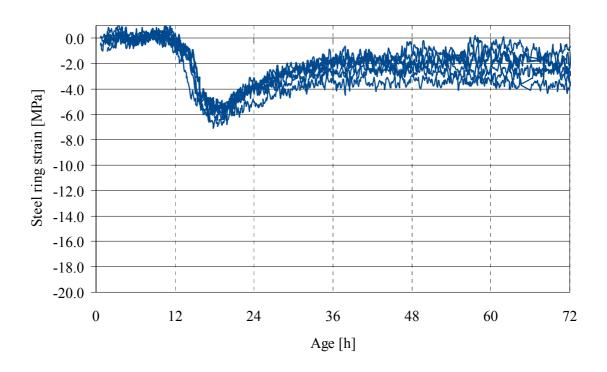


Figure 9–15 Restrained ring tests, main test series, six rings from three mixes, concrete 1A SRA

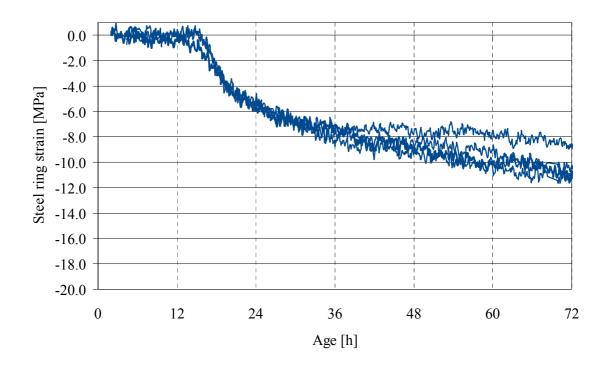


Figure 9–16 Restrained ring tests, main test series, four rings from two mixes, concrete 1A BFS

**Table 9–5** Fitting functions for stress analysis, concrete 1A

0		iess alialysis, (		
Dynamic modulus of elasticity [MPa]	Static modulus of elasticity [MPa]	Splitting tensile strength [MPa]	Autogenous shrinkage [mm/m]	Restraint stress [MPa]
582	483	0.11	0.000	0.000
1917	1591	0.28	0.182	0.231
4730	3926	0.52	0.391	0.594
9709	8058	0.83	0.536	1.015
16400	13612	1.20	0.610	1.448
22860	18974	1.63	0.652	1.759
27610	22916	2.14	0.675	1.872
30640	25431	2.80	0.687	1.843
32600	27058	3.64	0.693	1.761
33990	28212	4.56	0.695	1.675
35090	29125	5.38	0.696	1.604
36040	29913	5.96	0.696	1.548
36890	30619	6.31	0.696	1.506
37680	31274	6.50	0.696	1.471
38410	31880	6.65	0.696	1.441
39090	32445	6.78	0.696	1.411
39720	32968	6.90	0.694	1.380
	FedW   Attribute   FedW   September   FedW   Septem	[a]       [a]         [a]       [	Fe   W   W   W   W   W   W   W   W   W	Edwl in items         Image: Team of the component of the c

 Table 9–6
 Fitting functions for stress analysis, concrete 1A SAP

Age [h]	Dynamic modulus of elasticity [MPa]	Static modulus of elasticity [MPa]	Splitting tensile strength [MPa]	Autogenous shrinkage [mm/m]	Restraint stress [MPa]
9	3279	2722	0.37	0.000	0.000
10	5783	4800	0.63	0.046	0.155
11	9560	7935	1.01	0.105	0.388
12	13920	11554	1.52	0.144	0.556
13	18950	15729	2.14	0.172	0.707
14	23800	19754	2.85	0.194	0.843
15	28020	23257	3.60	0.210	0.938
16	31400	26062	4.35	0.221	0.986
17	33980	28203	5.06	0.227	0.999
18	35880	29780	5.69	0.231	0.997
19	37260	30926	6.23	0.233	0.991
20	38260	31756	6.69	0.236	0.987
21	38980	32353	7.07	0.239	0.986
22	39500	32785	7.37	0.242	0.985
23	39880	33100	7.61	0.245	0.984
24	40160	33333	7.81	0.247	0.978

 Table 9–7
 Fitting functions for stress analysis, concrete 1A SRA

Age [h]	Dynamic modulus of elasticity [MPa]	Static modulus of elasticity [MPa]	Splitting tensile strength [MPa]	Autogenous shrinkage [mm/m]	Restraint stress [MPa]
12	755	627	0.23	0.000	0.000
13	2707	2247	0.44	0.111	0.086
14	8179	6789	0.76	0.219	0.228
15	14600	12118	1.17	0.283	0.413
16	19400	16102	1.67	0.310	0.597
17	23220	19273	2.22	0.324	0.723
18	26480	21978	2.80	0.334	0.769
19	29210	24244	3.39	0.341	0.754
20	31390	26054	3.95	0.346	0.708
21	33070	27448	4.48	0.349	0.654
22	34310	28477	4.97	0.350	0.603
23	35200	29216	5.41	0.349	0.562
24	35840	29747	5.79	0.350	0.532

 Table 9–8
 Fitting functions for stress analysis, concrete 1A BFS

Age [h]	Dynamic modulus of elasticity [MPa]	Static modulus of elasticity [MPa]	Splitting tensile strength [MPa]	Autogenous shrinkage [mm/m]	Restraint stress [MPa]
15	462	600	0.03	0.000	0.000
16	814	1200	0.08	0.052	0.042
17	2074	2350	0.16	0.113	0.129
18	4831	4010	0.24	0.190	0.241
19	7925	6578	0.34	0.271	0.352
20	10280	8532	0.44	0.343	0.448
21	12120	10060	0.54	0.405	0.526
22	13690	11363	0.64	0.455	0.588
23	15090	12525	0.73	0.492	0.636
24	16300	13529	0.81	0.518	0.670